ACSNANO

www.acsnano.org

Spotting Local Environments in Self-Assembled Monolayer-Protected Gold Nanoparticles

4 Cristian Gabellini, Maria Sologan, Elena Pellizzoni, Domenico Marson, Mario Daka, Paola Franchi, 5 Luca Bignardi, Stefano Franchi, Zbyšek Posel, Alessandro Baraldi, Paolo Pengo, Marco Lucarini,* 6 Lucia Pasquato,* and Paola Posocco*



11 therapy, just to name a few. Much of their potential stems from 12 the unique control of organic environments around inorganic 13 sites within a single O–I nanomaterial, which allows for new 14 properties that were inaccessible using purely organic or 15 inorganic materials. Structural and mechanistic characterization 16 plays a key role in understanding and rationally designing such 17 hybrid nanoconstructs. Here, we introduce a general method-18 ology to identify and classify local (supra)molecular environ-



19 ments in an archetypal class of O–I nanomaterials, i.e., self-assembled monolayer-protected gold nanoparticles (SAM-AuNPs). 20 By using an atomistic machine-learning guided workflow based on the Smooth Overlap of Atomic Positions (SOAP) 21 descriptor, we analyze a collection of chemically different SAM-AuNPs and detect and compare local environments in a way 22 that is agnostic and automated, i.e., with no need of *a priori* information and minimal user intervention. In addition, the 23 computational results coupled with experimental electron spin resonance measurements prove that is possible to have more 24 than one local environment inside SAMs, such as the thickness of the organic shell and solvation primary factors in the 25 determining number and nature of multiple coexisting environments. These indications are extended to complex mixed 26 hydrophilic—hydrophobic SAMs. This work demonstrates that it is possible to spot and compare local molecular environments 27 in SAM-AuNPs exploiting atomistic machine-learning approaches, establishes ground rules to control them, and holds the 28 potential for the rational design of O–I nanomaterials instructed from data.

29 KEYWORDS: mixed monolayers, fluorinated nanoparticles, ESR, multiscale modeling, machine learning, SOAP, nanoconfinement

here is an intense interest in the rational design of 30 organic-inorganic (O-I) hybrid nanomaterials.¹ 31 Installation of organic molecules and specifically 32 33 thiol-containing ligands on a nanosized gold core is a primary 34 example of such O-I nanoplatforms. Thanks to reproducible 35 synthetic approaches that enable fine control over size, shape, 36 surface chirality, and dispersion, the easiness to passivate the 37 gold surface by the formation of a self-assembled monolayer 38 (SAM) and to further introduce a variety of functional groups 39 has enabled significant steps forward in the last years, granting 40 access to a plethora of SAM-enabled gold nanoparticles 41 (AuNPs) with functional properties.² Indeed, the self-42 organization of ligands endows SAM-AuNPs with unique 43 molecular recognition and sensing characteristics, which arise

ACS Publications

© XXXX The Authors. Published by American Chemical Society from the collective and cooperative behavior of the organic 44 layer.^{3,4} The nanoconfinement imposed to surface-bound 45 molecules dramatically influences their chemical and physical 46 properties, as well as conformation.^{5–8} For instance, Kay 47 studied the nanoparticle-confined hydrazone exchange.⁹ With 48 the help of molecular dynamics calculations, the work 49 demonstrated that at nanoscale SAM structure and conforma- 50

Received: August 24, 2022 Accepted: November 29, 2022

f1



Figure 1. Exemplification of the concept of local (supra)molecular environment (highlighted in blue) in SAM-AuNPs and its exploitation. (a) If ligands contain a catalytic group and the surrounding molecules adopt specific cooperative conformation and order, 3D binding sites similar to those in enzymes may arise with enhanced catalytic properties. (b) The end group on the surface switches on/off the access to a catalytic center and grants selective diffusion to the organic layer, causing different local structural features and reagent concentration. (c) Heteroligand monolayers of two immiscible ligands lead to surface anisotropy with implications for surface related biological processes and sensing of biomolecules, biomarkers, and drugs.

51 tional dynamics affects the transport properties and local 52 concentration of reagent water involved in the exchange, the 53 accessibility to the reaction sites, and ultimately the overall 54 reaction kinetics. Grzybowski and collaborators conceived a 55 mixed SAM-AuNP, in which longer ligands end in "gating 56 units" able to control both the access and orientation of the 57 incoming substrates with respect to the catalytic centers 58 tethered at the end of shorter ligands. Gating and substrate and 59 site selectivities derived from the molecular details of the on-60 particle molecular environment needed to be carefully designed.¹⁰ Mimicking the catalytic activity of proteins or 61 62 their interaction with biological matter exploiting SAM-AuNPs 63 has also been the object of growing exploration.¹¹⁻¹⁴ The 64 integration of bio-orthogonal catalytic systems such as 65 transition-metal catalysts into nanoparticle scaffolds allowed 66 the creation of synthetic catalytic nanosystems (nanozymes) 67 able to replicate the complex behavior of natural enzymes in 68 biological media.^{15,16} Hydrophobicity of surface motifs and 69 monolayer compaction regulate the kinetic behavior of the 70 nanozyme, together with temperature or pH.^{17,18}

The examples cited above point out the beauty and 71 72 complexity of surface confined environments in SAMs. They 73 all rely on the local structure, dynamics, and solvation of the 74 monolayer at atomic and nanoscale, although to a different 75 extent. With a broad term, they exploit the features of local (supra)molecular environments in SAMs (Figure 1). For 76 77 instance (Figure 1b), molecular structure, accessibility, surface 78 morphology, and local reagent concentration change when the 79 gate is open or closed. Thus, in this context, we can think of 80 local (supra)molecular environments as regions of the 81 monolayer with unique distinct fingerprints. The term 82 encompasses multiple interconnected effects, such as atom 83 density, ligand dynamics and conformation, monolayer 84 structure, and ligand-ligand and ligand-solvent interactions 85 as well as local solvation or substrate concentration (if any). As 86 such, they are hard to anticipate and only few of them can be 87 directly assessed with experiments by using techniques such as ⁸⁸ NMR,^{19–21} SANS,²² MALDI-TOF,²³ and ESR;^{24,25} yet, these 89 techniques suffer of some limitations, as the monolayer needs 90 to be designed *ad hoc* for the specific technique.

Thus, we wondered if a general way to identify specific local settings in SAMs could exist. Molecular dynamics (MD) and or coarse-grained simulations have been instrumental in retrieving information difficult to infer from experiments and in sexplaining the behavior of SAM-AuNPs at molecular and nanoscale with good reliability.^{26–29} Over recent years, the 96 increasingly large amounts of data produced by these 97 calculations have also been used by algorithms to extrapolate 98 molecular patterns and predict (meta)stable configurations or 99 structural motifs in complex matter.^{30–32} 100

Here, in a proof-of-concept study, we introduce a two-step 101 computational workflow able to detect first and then compare 102 local (supra)molecular environments in SAM-AuNPs with no 103 need of predefined information and minimal user intervention. 104 It combines atomistic all-atom MD (AA-MD) calculations and 105 the Smooth Overlap of Atomic Positions (SOAP) descriptors 106 for machine-learning guided analysis. The retrieved local 107 environments are then described and rationalized by MD 108 calculations and supported by experiments of electron spin 109 resonance (ESR), a spectroscopic technique highly sensitive to 110 polarity changes in the local background perceived by a radical 111 probe,²⁵ that are carried out at different temperatures. 112

A set of AuNPs (roughly 4.0 nm in size), which support 113 homo- and hetero (mixed)-SAMs composed of thiolates 114 ending in positive (ligands 1 and 2) or negative (ligands 3 115 and 4) or zwitterionic (ligand 5) charged end groups and short 116 fluorinated ligands (ligand 6) (Figure 2), is tested. We sought 117 f2 to augment the complexity of the monolayer by including 118 fluorine containing mixed SAMs, which are particularly 119 relevant for driving surface phase separation,^{21,26} controlling 120 hydrophobicity or superphydrophobicity of surfaces,³³ or 121 tuning the molecule–NP interaction.³⁴

Hereafter, we adopt the following notation: NP1 indicates a 123 SAM of ligand 1 on AuNP while NP1/6, a SAM of ligand 1 124 and 6 on AuNP. 125

The paper is organized as follows: first, NP structure and 126 properties from AA-MD simulations in solvent (water) are 127 discussed; second, the computational approach for the 128 identification and comparison of local motifs in different 129 SAMs is illustrated and the outcomes considered; third, the 130 results are interpreted in light of ESR investigation. 131

Overall, this work not only demonstrates that it is possible to 132 spot local (supra)molecular environments in SAM-AuNPs by 133 exploiting atomistic data-driven approaches but also is a step 134 toward the design of functional nanoparticles with a programmable response. 136

RESULTS AND DISCUSSION

MD-Derived SAM-AuNP Characterization. The specific 138 structure of the monolayer is imparted by the self-organization 139

137



Figure 2. Structure of the thiolates 1-6 for the AuNP coating. Radical probe 7 for ESR investigation. Ligand 6 is used in mixed monolayers with 1-5. Thiolates differ in nature and charge of the terminal group (1 and 2, a positively charged quaternary ammonium ion; 3 and 4, a negatively charged sulfonate ion; 5, a zwitterionic group, composed by a trimethylammonium and a phosphate group) as well as in length of the alkyl chain (C₁₂ in 1, 3, 5; C₁₆ in 2, 4).

140 of the individual thiolates on the surface of the gold core. We 141 have very recently demonstrated by calculations²⁸ that the 142 surface morphology depends on size and hydrogen bonding 143 capability of the ligand end group, while other features, such as 144 the alkyl chain length or the core size, affect the final ligand 145 organization less. In particular, a large space-filling group like 146 trimethylammonium or zwitterionic ones give rise to spatially 147 uniform arrangements due to the steric hindrance of bulky 148 terminal moieties; small end groups like sulfonate allow association of the chains in bundles, which instead leads to 149 anisotropic shells (Figures 3 and S1). The combination of 150 f3 more than one kind of ligand in the shell has long been used in 151 the nanoparticle community to tune nanoparticle solubility, 152 wettability, interfacial properties, hydrophobic interactions for 153 self-assembling nanoparticles, respond to the surrounding 154 (bio)environment, and induce nanoscale surface morpholo-155 gies.^{4,35-39}

Indeed, when a mixture of dissimilar and/or immiscible 157 molecules are employed to coat AuNPs, nanoscale domains 158 may spontaneously form in the shell via ligand surface 159 rearrangement.⁴⁰ Fluorinated ligands are known to be highly 160 lipophobic, and we have already tested their ability to trigger 161 phase separation in 3D SAMs when used in combination with 162 hydrogenated thiolates even at low molar fraction.^{21,26} Here, 163 we have carried out auxiliary mesoscale simulations (to cope 164 with the slow evolution of the phase separation at the 165 nanoscale)⁴¹ coupled with AA-MD calculations in water to 166 predict the pattern of organic shells containing fluorinated 167 thiolates, namely, ligand 6. For details on molecular models 168 and simulations, see the Experimental Section and Supporting 169 Information (SI) Section S3. Gold size, ligand density, and 170 monolayer composition have been assigned to closely match 171 those obtained experimentally (see SI Section S2). 172

The calculations confirm that, irrespective to the chemical 173 nature and the chain length of the primary ligand (i.e., 1-5), 174 ligands **6** separate in small domains (Figures 3 and S1). For 175 AuNPs bearing sulfonates and zwitterionic moieties as surface 176 groups (namely, NP3/6, NP4/6, and NP5/6) these domains 177 appear as elongated patches with an average width of 1.6–1.9 178 nm and length of 2.7–3.6 nm. Stripe-like patterns are indeed 179 present on NP1/6 and NP2/6, where the bulkier headgroups 180 favor the formation of domain interfaces more (Figure S2).⁴² 181



Figure 3. Representative molecular structures of homoligand NP1, NP3, and NP5 AuNP and its heteroligand NP1/6, NP3/6, and NP5/6 counterpart from molecular dynamics simulations in explicit solvent (water). For clarity, water and counterions are not shown. Color representation of atoms: C, gray; O, red; S, yellow; P, orange; N, blue; F, green; H, white.





Figure 4. (a) Normalized water distribution at increasing distance from the gold surface for NP1/6, NP3/6, and NP5/6. The graphs plot the distribution of the atom (oxygen of water or carbon of thiolates) closest to gold surface (centered on the gold core and placed at increasing distances from its surface) shown as a two-dimensional projection of the sphere surface (*x*-axis, the azimuthal angle φ ; *y*-axis, the cosine of the polar angle θ). A value of 1 indicates that an oxygen atom of a water molecule is always the closest; if it is equal to 0, it indicates that a carbon/fluorine atom of a chain is always the closest. Simplifying, red to salmon areas represent poorly hydrated zones, while blue areas stand for highly hydrated parts of the monolayer (at a certain distance from the gold surface). At distances lower than those considered, the microenvironment is almost hydrophobic, while at higher distances, it is fully hydrated and no major difference between the monolayers could then be detected. Maps for NP2/6 and NP4/6 can be found in the SI (Figures S5 and S6). (b) Examples of possible different hydration states within SAMs.

The phase separation does not alter the propensity of the 182 183 most abundant ligand to associate in bundles. Thus, NP3/6 184 and NP4/6 have a spatially heterogeneous ligand distribution 185 as is also observed for their homoligand counterpart NP3 and 186 NP4 (Figures 3 and S1). The clustering (or bundling) of 187 ligands can be quantified by means of Voronoi diagrams, which 188 allow local density estimation through nearest neighbor 189 analysis⁴³ (Figures S3 and S4). In both monolayers, the 190 presence of high-density regions where the chains form 191 bundles is evident and these roughly correspond in number 192 to those identified using a different clustering algorithm (e.g., 193 HBDSCAN, see SI Table S1), supporting the presence of time-194 persistent aggregation of ligands. Regular and more uniform 195 patterns instead characterize the Voronoi diagram of NP1, 196 NP3, and NP5 and their heteroligand partners NP1/6, NP3/ 197 6, and NP5/6 (Figures S3 and S4) consistently with an 198 isotropic distribution of the ligands around the gold core. The 199 results also highlight that long ligands (i.e., HS-C₁₆-FG) on 200 nanoparticled induce more heterogeneous ligand distributions, 201 which appeared clearly from the visual inspection of the 202 diagrams (e.g., compare Figures S3a and S4a or Figures S3c 203 and S4c). Yet, a simple measure is provided by the area 204 dispersion index (ADI), which describes the spread of the

tessellation cell areas (see SI Section S3.3 for how ADI is 205 calculated) (Table S1). For NP1, ADI is equal to 2.24 and 206 increases to 2.52 for NP2, indicating a broader distribution of 207 the area available for each ligand; the increased local 208 heterogeneity in long chains has also been seen by others,⁴⁴ 209 and it is promoted by higher interchain van der Waals 210 interactions and higher free chain volume due to the increased 211 radiality. For anisotropic shells like NP3 and NP4, this 212 phenomenon is less evident from ADI analysis (ADI is equal to 213 2.96 and 3.12, respectively) but still detectable in the diagrams. 214 Adding a second ligand in the monolayer does not affect the 215 overall monolayer structure yet impacts the ligand local order. 216 Indeed, for almost all the heteroligand monolayers, the ADI 217 decreases compared to the homoligand AuNP, thus indicating 218 a more uniform distribution of the space available for each 219 chain, likely because of the bulky fluorinated alkyl thiolates. 220 Further structural analysis of the monolayer is available in 221 Tables S1 and S2. 222

Revealing monolayer structure and molecular order is the 223 first necessary step to gather information about nanoparticle 224 hydration and solvation-related properties.⁴⁵ Previous exper- 225 imental and computational efforts^{46–48} have highlighted that 226 ligand ordering is more correlated than other conventionally 227



Figure 5. Conceptual diagram of the workflow used for the detection and comparison of local molecular environments within self-assembled monolayers (SAMs) using the Smooth Overlap of Atomic Positions (SOAP)-based structural analysis. Molecular dynamics calculations of the SAM-AuNP and reporter 7 are conducted in explicit solvent. The SOAP descriptor vector is constructed taking the reporter atoms (here the nitrogen atom) as the center of the structural environment up to a given cutoff radius r_1 (*medium*-range description) and employed for the identification of molecular fingerprints assigned by an unsupervised clustering algorithm (step 1). The *short*-range SOAP descriptor is built considering only solvent molecules up to a range of r_2 ($<r_1$), and a linear kernel between SOAP vectors is used to measure the similarity between the environments (step 2) and interpreted by correlating the location of the data with the MD evidence. For more details on each step, see Figure S8 and Section S3 in the SI.

228 considered chemical properties (such as the solvent-accessible 229 surface area (SASA)) with the interfacial hydrophobicity of 230 SAMs. Most of the studies have been conducted on planar 231 SAMs and, when extended to curved surfaces, were focused 232 only on the description of the SAM–water interface. Here, we 233 expand the investigation of nanoparticle hydration to the 234 whole interior of the monolayer and we also consider the 236 hydrophobicities. To do that, we relate the normalized water 237 content and the spatial distribution of water molecules within 238 the monolayer at increasing distances from the core and we 239 project it onto bidimensional planes. This provides an 240 immediate view of the average degree of solvation of the 241 monolayer and the topological distribution of the solvent 242 within the monolayer (Figure 4).

When one observes the water density maps reported in 244 Figure 4, it appears that isotropic monolayers allow a uniform 245 diffusion of the solvent within the organic layer; the water 246 content decreases progressively when moving toward the 247 nanoparticle center, and there is a concentration gradient with 248 respect to bulk solution (Figure 4b). On the contrary, the 249 presence of bundles generates alternation of highly hydrated 250 zones between the bundles (at a level comparable to that of 251 bulk solvent) and dehydrated areas, where solvent penetration 252 is hindered by the strong self-association among bundled alkyl 253 chains.

An additional element affecting the hydration is the existence of the extended ligand/water hydrogen bond existence of the nanoparticle surface, which reduces the internal diffusion of the solvent and makes zwitterionic site nanoparticles less hydrated than other isotropic systems, like for example NP1.

Fluorine-rich ligands **6** are considerablely shorter than all the the other thiolates; thus, when they segregate in domains, they ease nable the local diffusion of the solvent closer to gold, resulting as in a higher content of water with respect to homoligand AuNPs at the same distance from the gold surface (see also 264 Figures S5–S7). 265

Automated Detection of Local (Supra)Molecular 266 Environments in SAM. The calculations just described are 267 the entry points of an automated workflow able to identify first 268 and then compare local (supra)molecular environments within 269 any SAMs. It is based on the combination of AA-MD 270 calculations of SAM-AuNP carried out in explicit solvent, an 271 agnostic machine-learning structural analysis employing the 272 SOAP⁴⁹ formalism to describe the 3D atomic environment 273 that surrounds a reporter molecule (here, the radical probe 7; 274 see Figure 2) interacting with the monolayer, and an 275 unsupervised probability-based method for clustering the 276 data (Figure 5). In the SOAP framework, the local atomic 277 f5 environment of an atom (defined as a SOAP center) is 278 represented by the sum of element-specific smooth Gaussian 279 densities centered on the positions of neighborhood atoms 280 within a spatial cutoff, and it is associated with a vector, 281 commonly known as "SOAP power spectrum" or "SOAP 282 fingerprint" (see SI Section S3 for the SOAP formal 283 derivation). SOAP vectors provide a high-dimensional, 284 agnostic representation of molecular environments. SOAP 285 descriptors have been successfully applied in exploring the 286 conformational landscape of single molecules,⁵⁰ recognizing 287 local structural motifs⁵¹ and describing formation/dynamics of 288 soft supramolecular fibers,⁵² returning a rich structural/ 289 dynamical characterization of complex molecular systems. 290 Such an analysis in our systems allows us to unveil different 291 states of the molecular reporter 7 based on differences in the 292 local environment (microenvironment) that surrounds it 293 during the AA-MD simulation time, accounting for overall 294 atomic composition, molecular conformation, local order, 295 persistency in the interactions, and degree of solvation. 296

The workflow consists of two main steps both starting from 297 an (equilibrated) AA-MD trajectory of a specific SAM-AuNP/ 298 7 complex. The first one is the *classification* of the local states 299



Figure 6. First two principal components (PCA1 and PCA2) obtained from dimensionality reduction of the medium-range SOAP feature space of the probe 7 in thicker homoligand NP4 (a) and NP2 (b). Dots are colored according to the clusterization obtained by the GMM analysis. For each cluster, the inset shows the molecular environment centered on the probe 7, extracted from the corresponding MD frames. Color legend: probe, same color of the cluster; ligands 4 and 2 in gray; solvent not shown for clarity. (c, d) Example of the molecular view of the local environments NP4₁ and NP4₂ including all atoms within the cutoff r_1 . The reporter is colored according to the cluster assigned as a sphere; water is shown in the same color of the probe but as a transparent surface, and the ligands belonging to the environment are highlighted as white spheres. The remaining ligands are left as a background gray surface. (e, f) Free energy surface (FES) (kcal/mol) calculated from the state's probability distribution in (a) and (b), respectively. Dots identify the minima on the FES and are colored based on the microstate (cluster) they refer to.

³⁰⁰ of the probe (step 1) (Figure 5 and flowchart in Figure S8). To ³⁰¹ identify the relevant microenvironments visited by 7, the ³⁰² SOAP descriptors are calculated to be centered on the nitrogen ³⁰³ atom of the probe. The SOAP data set includes all atoms ³⁰⁴ within a cutoff radius r_1 (9 Å), which is taken as a compromise ³⁰⁵ between the ability to capture relevant local structural ³⁰⁶ correlations and necessity to minimize the computational ³⁰⁷ requirements for SOAP manipulation and storage (see SI ³⁰⁸ Section S3, Figure S23). We refer to that as "medium-range ³⁰⁹ SOAP vector".

Linear principal component analysis (PCA) is then applied to reduce the high dimensionality of the SOAP features space (14354 dimensional on average) without losing important features. ~94% of the total variance (e.g., global information) if is retained keeping the first 10 principal components (Figure S24). Then, a probabilistic model based on Gaussian mixtures (GMMs) is exploited as an unsupervised clustering scheme. This allows one to partition and classify all the environments perceived by the probe into groups (i.e., clusters) and distinguish them without any prior information on the number of clusters (for a description of the clustering algorithm, see SI S21 Section S3). The outcomes of PCA are visualized by projecting the 10 PCs in 2D onto the first two principal components, PCA1 and PCA2, to provide simple and intuitive maps.

Through the SOAP-GMM analysis, two distinct states (i.e., microenvironments) are identified for NP4 (Figure 6a): the probe 7 lays at the ligand bundle–water interface close to the gold core (1, orange) or parallel to ligand chain (2, blue). 327 From now on, each local environment is reported in subscript: 328 for example, **NP4**₁ indicates the local environment (1) in **NP4**. 329

As an example, Figure 6c,d shows a molecular view of the 330 ligands and water molecules forming the local environments 331 **NP4**₁ and **NP4**₂. From the MD trajectory, we also calculate the 332 free energy surface (FES) of the reporter 7 in the system as the 333 probability distribution of states in the PCA reduced SOAP 334 feature space (P) by using the standard statistical relation FES 335 = $-K_bT \log(P)$ and find that the states correspond to two local 336 minima equally visited by 7 (Figure 6e). The classification is 337 fully consistent with our previous findings,²⁸ where two distinct 338 positions of 7 were also identified by classical analysis of the 339 MD trajectory in **NP4**, one more deeper in the organic layer 340 and one more exposed to the exterior. 341

Two structural states are also detected in NP2 (Figure 6b), 342 meaning that thicker monolayers are able to host a small 343 molecule in structurally distinguishable *loci*. Yet, inspection of 344 the clusterization maps suggests that the difference between 345 the two states is sharper in bundled shells. In fact, in NP4, the 346 clusters are well distinct and clearly separated; in NP2, the 347 transition is smoother, although measurable by SOAP-GMM. 348 We attribute this to the diverse ligand arrangement in NP2 and 349 NP4. Chain packing allows accommodation of a small 350 molecule like 7 by simple binding at the ligand bundle 351 interface at increasing depth from the outer surface, and the 352 search of an optimal interaction position for the probe is 353



Figure 7. First two principal components (PCA1 and PCA2) obtained from dimensionality reduction of the medium-range SOAP feature space of the probe 7 in heteroligand bundled NP3/6 and NP4/6 (a) and isotropic NP1/6, NP2/6, and NP5/6 (c) monolayers. Dots are colored according to the clusterization obtained by the GMM analysis. For each cluster, the inset shows the molecular environment centered on the probe 7, as extracted from the corresponding MD frames. Color legend: probe, same color of the cluster; ligands 1–5 colored in gray; ligand 6 colored in dark gray; solvent not shown for clarity. (b) Free energy surface (FES) (kcal/mol) calculated from the state's probability distribution for NP3/6 and NP4/6 (b) and NP1/6, NP2/6, and NP5/6 (d). Dots identified the minima on the FES and are colored based on the microstate (cluster) they refer to. The arrows indicate the transition probabilities between the states from the minimum.



Figure 8. Similarity matrix for all local (most visited) environments generated by calculating the pairwise SOAP kernels K_{SOAP} between all the reduced short-range SOAP feature vectors. Dark blue color indicates high similarity between the environments.

354 facilitated by the freedom to explore the conformational space 355 at that interface; this would lead to well-defined binding sites 356 for **NP4**.

On the other hand, in isotropic monolayers as NP2, the accommodation of a guest requires diffusion within the ligands, hampering the access to the whole depth of the monolayer and thus leveling out differences between interaction positions.

When the SOAP-GMM classification is applied to nano-362 particles having a shorter hydrophobic portion like NP1 and 363 NP3, one single microenvironment is identified for 7 (see 364 Figure S9). Although NP5 could be assimilated to NP1 and 365 NP3, the classification returns a different picture (Figure S9); 366 in fact, it unveils the presence of two clusters, namely, two 367 states explored by the probe. Nevertheless, the FES indicates 368 that one of them is much more visited than the other and sets 369 itself as a local minimum. The more complex behavior of the 370 zwitterionic NP5 reflects the uniqueness of this monolayer in 371 agreement with the evidence from the AA-MD calculations.

Mixed shells containing hydrophobic patches enrich the moligand to their respective homoligand and isotropic monolayers (Figure 7).

NP3/6 and NP4/6, which have ligand clusters, show three 376 (metastable) local environments for the probe: the first on the 377 378 fluorinated chains $(NP3/6_3 \text{ and } NP4/6_3)$ being the least 379 visited, the second down at the interface between alkyl and 380 fluorinated domains (NP3/ 6_2 and NP4/ 6_1), and the third with ³⁸¹ the probe parallel to the bundles (NP3/ 6_1 and NP4/ 6_2). The 382 three states are distinct and clearly separated in the SOAP 383 feature space. On the contrary, in isotropic monolayers like 384 NP1/6 and NP2/6, there are only two states possible, which 385 are not so well divided in the SOAP space as in NP3/6 and 386 NP4/6, thus highlighting the importance of the monolayer 387 arrangement in shaping local environments. An exception is 388 the zwitterionic NP5/6 for which the interfacial NP5/ 6_1 is 389 highly favorable and well distinguished from NP5/ 6_2 . From

the FES inspection, still in heteroligand monolayers, two states $_{390}$ are the most probable for C₁₆ long chains (i.e., NP2/6 and $_{391}$ NP4/6), and these reduce to one for shorter ligands (NP1/6, $_{392}$ NP3/6, and NP5/6).

Comparison of Local Environments in Different 394 **SAMs.** Once the most probable interaction site(s) is identified 395 for each system, we want to compare them (step 2 of the 396 workflow, Figure 5). This means assessing how much those 397 local environments are similar; they belong to either the same 398 monolayer or to different nanoparticles. To ensure that the 399 SOAP analysis is meaningful and fully comparable across 400 different systems, one needs to choose a representation of the 401 structural space that takes into account common features 402 between systems. For that reason, we select the (roughly) first 403 hydration layer of the reporter 7. This choice also allows us to 404 limit the computational costs, since now all the systems have to 405 be analyzed together. The nitrogen atom of the probe 7 is still 406 assigned as the SOAP center, and the cutoff radius r_2 is now set 407 to 4.5 Å (Figure S23), including only solvent molecules. We 408 refer to that as "short-range SOAP". Accordingly, taking the 409 MD snapshots where the probe is in the most favorable 410 state(s) based on the assignment of the medium-range SOAP- 411 GMM clusterization, we construct the corresponding short- 412 range SOAP fingerprint for each molecular environment and 413 each nanoparticle. Then, we perform a dimensionality 414 reduction via linear PCA to obtain a low-dimensional 415 representation and consider only the first 10 components 416 (Figure S24). 417

Measuring structural similarity requires the definition of a 418 metric that is capable of identifying identical molecular 419 fingerprints. There are different ways of combining atom- 420 centered representations to obtain a structure-level compar- 421 ison;⁵³ in the SOAP space, one natural choice is to define a 422 linear kernel of the density representation in the form of the 423 dot product of the SOAP power spectra of the two molecular 424 environments $K(i,j)_{SOAP}$ (see SI Section S3). SOAP-based 425

⁴²⁶ structural similarity kernels can be interpreted as a measure of ⁴²⁷ how much two (smoothed) atomic distributions (*i*,*j*) are ⁴²⁸ superimposed on each other (i.e., how much similar the local ⁴²⁹ environments are in the SOAP space). The data are displayed ⁴³⁰ in the form of a similarity matrix by converting the similarity ⁴³¹ value $K(i,j)_{\text{SOAP}}$ to an Euclidean distance metric d_{SOAP} , ranging ⁴³² from 0 to 2 (Figure 8).

The color level is proportional to the value of the similarity 434 between environments: dark blue corresponds to the highest 435 value of similarity computed and light blue indicates larger 436 SOAP distances and an increased structural difference between 437 the environments. From Figure 8, it appears evident that there 438 are two distinct classes of environments according to the 439 similarity metric: the first one (type-1) is that corresponding to 440 the lower right quadrant of the matrix and the second (type-2), 441 to the upper left quadrant.

It is worth noting that the classes are not equally populated. Type-2 includes only a few molecular environments, of which anoparticles), while the type-1 class is broader and comprises environments from long $(-S-C_{16}-FG)$ to short $(-S-C_{12}$ the environments from long $(-S-C_{16}-FG)$ to short $(-S-C_{12}$ the that thick $(\geq C_{16})$ monolayers have the ability to form local environments with structural features well distinguishable from those existing in thin monolayers. In addition, the results show that it is possible to capture and discriminate multiple environments applying a pure data-driven evaluation without a priori assumptions.

To gain more insights and in an attempt to rationalize these 454 455 outcomes, we then link each state to the corresponding 456 molecular structure retrieved from the MD snapshots as 457 assigned by the medium-range SOAP-GMM to that environ-458 ment; in this way, we find out that nanoparticles with a single 459 interaction site (namely, $-S-C_{12}-FG$) are classified as type-1; 460 in systems with two main interaction sites, one is of type-1 and 461 the other is of type-2. Type-2 sites correspond to local 462 environments where the probe is placed closer to the gold core 463 and the overall hydration is limited (as an example, see Figure 464 6c for NP4₁ or Figure 7a for NP4/ 6_1). We assess that by 465 simply calculating the radial distribution function (RDF) of the 466 nitrogen atom of the reporter (i.e., the probability distribution 467 as a function of distance from the metal center) from the 468 corresponding MD frames and matching the peak of the RDF 469 with the solvation map to that distance. Type-1 environments 470 instead share a higher solvation, and the probe is more exposed 471 to the external environment (as an example, see Figure 6d for 472 NP4₂ or Figure 7a for NP4/ 6_2). The chemistry of the thiolates 473 end group has no major influence on the features of the 474 interaction site, which is not completely surprising since the 475 probe is mainly interacting with the alkyl part of the ligands 476 (Figures 6, 7, and S9).

ESR Analysis of SAM-AuNPs. Experimentally, monolayer task features can be investigated by molecular probes, which are are able both to enter inside the monolayer and to possess spectral task features that depend on the molecular environment of the surroundings. Functionalized benzyl *tert*-bytulnitroxides (BTBN) possess such characteristics and have been largely task employed to characterized different types of water-soluble task SAM-protected AuNPs.^{54–57} In the present study, probe 7 task containing a pentyl chain at the *para* position of the aromatic task ring and a hydroxymethyl group in place of the methyl in the task tert-butyl substituent is employed for ESR investigation. This task hydrophobic probe has been chosen because of its good

affinity for the nanoparticle organic monolayer when dissolved 489 in water. Experimental values of hyperfine splitting constants 490 (hfsc's) of heteroligand nanoparticles are collected in Table 1 491 t1

Table 1. Spectroscopic Parameters for Radical Probe 7 at	
Different Temperatures (Black at 300 K and Light Blue a	ιt
340 K)	

NP	$T(\mathbf{K})$	$a_{\mathrm{N}}(G)$	$a_{ m 2H}(G)$
-	300	16.25	10.14
_	340	16.22	9.80
NP1 ^a	300	15.20	8.50
NP1/6	300	15.23	8.55
NP1 ^a	340	15.35	8.46
NP1/6	340	15.32	8.44
NDo ah	300	14.50	8.45
NF2 ^{4,0}	300	15.18	8.58
NIDa /ch	300	14.59	8.50
$\mathbf{NF}2/0^{\circ}$	300	15.10	8.65
NP2 a	340	15.15	8.50
NP2/6	340	15.13	8.60
NP3 a	300	15.15	8.40
NP3/6	300	15.10	8.50
NP3 a	340	15.40	8.48
NP3/6	340	15.33	8.40
ND4 a	300	14.40^{b}	8.38^{b}
INF 4 ^{**}	300	15.23	8.30
ND4/c	300	14.30^{b}	8.50 ^b
INI 47 O	300	15.35	8.35
NP4 ^a	340	15.32	8.40
NP4/6	340	15.28	8.36
NP5 ^a	300	15.25	8.35
NP5/6	300	15.30	8.55
NP5/6	340	15.22	8.50

^{*a*}Data from ref 28. ^{*b*}The a_N values given in bold refer to the probe in the most hydrophobic location.

together with those previously²⁸ measured in the presence of 492 homoligand nanoparticles and in a temperature range between 493 300 and 340 K (for details on mixed-monolayer nanoparticle 494 synthesis, XPS characterization, and ESR measurements, see 495 the Experimental Section and SI Sections S2, S4, and S5; 496 otherwise, refer to our previous work²⁸). 497

With NP1/6, NP3/6, and NP5/6 and their homoligand 498 partner, spectra are characterized by two different resolved sets 499 of signals (as an example, see Figure 9) at 300 K. The one with 500 f9 larger hyperfine coupling constants is due to the probe located 501 in water, while the second one, has nitrogen hfsc's ($a_{\rm N}$, 502 reported in Table 1) significantly smaller than that measured 503 for 7 in solution, resulting from the probe positioned in the 504 monolayer. Analysis of the spectra suggests the presence of a 505 single interaction site, in line with the SOAP-GMM analysis for 506 nanoparticles having short chain shells.

In all thick monolayers (namely, NP2- and NP4-type 508 systems), the ESR analysis shows the presence of two distinct 509 environments, where the probe bound to the monolayer 510



Figure 9. ESR spectra of the radical probe 7 recorded in water in the presence of NP3/6 (a) and NP4/6 (b) at 300 K. In red are reported the corresponding theoretical simulations obtained by employing the spectroscopic parameters reported in Table 1.

s11 experiences different background polarities (Table 1). The first s12 one has experimental values of a_N in the range of 14.30–14.60 s13 G, significantly smaller than that for the probe in water (16.26 s14 G), indicating an extremely low polarity. This is consistent s15 with type-2 settings, where 7 lies close to the gold core and the s16 overall hydration is limited, thus corresponding to NP2₂, s17 NP4₁, NP2/6₁, and NP4/6₁ states. In addition, such low s18 values of a_N are seen only for long chain shells, in agreement s19 with the SOAP-GMM classification (see Figure 8).

520 The second interaction location has much higher spectro-521 scopic parameters (15.10–15.35 G), closer to that of the probe 522 free in solution, which are indeed associated with an increased 523 environment polarity perceived by the probe. The existence of 524 a second interaction site in NP2- and NP4-type systems 525 parallels well the computational prediction, supporting the 526 possibility to have distinguishable local environments within 527 the same (thick) monolayer. Interestingly, the spectroscopic 528 parameters are comparable to those for NP1-, NP3-, and NP5-529 type monolayers. This shows that the radical samples very 530 similar environments in all these systems. SOAP similarity 531 analysis returns a classification which is in line with this 532 interpretation: in fact, NP21 NP42, NP2/62, and NP4/62 533 environments are assigned to the same category (type-1 534 class) as NP1(/6), NP3(/6), and NP5(/6) environments 535 (Figure 8).

By increasing the temperature, a new set of signals, 537 characterized by spectroscopic parameters very similar to 538 those previously measured in type-1 *loci* appears in the 539 spectrum, as was also seen by repeating the SOAP-GMM 540 analysis including the most stable states at 340 K (Figures S10 541 and S11). Hence, when the temperature is increased, the probe 542 experiences local environments with higher polarity, solvation, 543 and exposure to the surroundings that makes type-1 sites the 544 most favorable interaction locations for the systems under 545 investigation.

Quite unexpectedly, based on our current understand-547 ing,^{26,57} the spectroscopic parameters of the probe in mixed-548 monolayer NPs do not differ significantly from those measured 549 in the corresponding homoligand shell. Previous evidence on 550 the monolayers made by mixtures of hydrocarbon/perfluor-551 ocarbon chains terminating with a short poly(oxoethylene) 552 moiety indeed suggested that the probe should preferentially reside in fluorinated domains. Here, instead, MD calculations 553 clearly show that 7 never enters or fully interacts with the 554 fluorinated patches and is located preferably at the fluorine 555 domain interface (and thus explains the similarity in the 556 nitrogen hspcs). In addition, MD calculations and Voronoi 557 diagrams display that short F-alkyl chains are densely packed in 558 the NPs considered here, physically and energetically 559 preventing them to host the radical probe. 560

CONCLUSIONS

In summary, ligands self-assembling on the surface of gold 562 nanoparticles can create local (supra)molecular environments 563 with unique fingerprints that allow them to be precisely 564 detected and exploited. We have presented a computational 565 approach, which enables automated identification and 566 comparison of such environments driven from the data (i.e., 567 from atomistic MD trajectories) and without feeding input 568 parameters. The computational workflow is built on 569 unsupervised clustering of the Smooth Overlap of Atomic 570 Position (SOAP) atomic descriptors and a simple SOAP 571 metric to classify the environments. In this proof-of-concept 572 study, we have considered a collection of chemically different 573 SAM-AuNPs, bearing cationic, anionic, and zwitterionic 574 surface groups and having different monolayer thicknesses. 575 The set includes homo- and heteroligand monolayers; the 576 second ones present alternating hydrophilic/hydrophobic 577 surface patterns that stem from the nanoscale separation of 578 two immiscible ligands. By the SOAP analysis and in 579 conjunction with ESR measures, we have successfully 580 demonstrated that multiple structural and chemical micro- 581 environments can exist together within the SAM-AuNPs 582 investigated. In particular, they differ for accessibility, local 583 solvation, and hydrophobicity, which are imparted by specific 584 ligand length, nature of the ligand end group, and monolayer 585 3D structure. 586

The results of our investigation allow us to draw some 587 general conclusions: (i) anisotropic monolayers may facilitate 588 the establishment of settings having well-defined and easily 589 distinguishable local (supra)molecular motifs; (ii) in the 590 absence of chemical groups designed to recreate specifically 591 intended binding or catalytic sites, thick monolayers naturally 592 lead to multiple, coexisting environments, which are shaped by 593 confined solvent, organization, and conformational mobility of 594 the ligands; (iii) surface patterns in heteroligand shells give rise 595 to a multiplicity of states, which could be potentially targeted 596 under appropriate thermodynamic or kinetic pathways. 597

Overall, this work provides a promising general approach for 598 systematic and computationally efficient investigation of local 599 (supra)molecular environments in SAM-AuNPs, a widely used 600 class of O–I nanomaterials, and establishes a mechanistic 601 understanding of their intimate features with a full account of 602 nanoscale effects. The next steps will be the extension to more 603 complex functional nanoparticles and the design guided by 604 machine-learning algorithms of local motifs with predefined 605 properties. 606

EXPERIMENTAL SECTION

Nanoparticle Synthesis and Characterization. Detailed 608 synthetic procedures and characterization for mixed monolayers 609 nanoparticles can be found in the Supporting Information; otherwise, 610 the reader may refer to our previous work.²⁸ All commercial reagents 611 were purchased from Aldrich and VWR and used without purification 612 unless otherwise mentioned. Solvents were purchased from Aldrich 613

607

561

614 and VWR and deuterated solvents, from Cambridge Isotope 615 Laboratories and Aldrich. Dry solvents were obtained from Aldrich. 616 Chlorinated solvents were kept over K₂CO₃ for at least 24 h prior to 617 use. All other solvents were reagent grade and used as received. 618 Reactions were monitored by TLC on Merck silica gel plates (0.25 619 mm) and visualized by UV light, I2, or KMnO4-H2SO4 solution. 620 Chromatography was performed on Merck silica gel 60F-254 (230-621 400 mesh), and the solvents employed were of analytical grade. NMR 622 spectra were recorded on a Varian 500 spectrometer (operating at 500 623 MHz for proton and at 125 MHz for ¹³C) or on a Varian 400 MHz $_{624}$ (operating at 400 for proton, at 376.16 MHz for $^{19}\textrm{F}\textsc{,}$ and at 100.5 625 MHz for carbon). ¹H NMR chemical shifts were referenced to the 626 residual protons in the deuterated solvent. ¹⁹F NMR spectra were 627 referenced to CFCl3 chemical shift, and ¹³C NMR chemical shifts 628 were referenced to the solvent chemical shift. Chemical shifts (δ) are 629 reported in ppm, and the multiplicity of each signal is designated by 630 the conventional abbreviations: s, singlet; d, doublet; t, triplet; q, 631 quartet; m, multiplet; br, broad; dd, doublet of doublets. Coupling 632 constants (J) are quoted in Hz. UV-visible spectra were recorded on 633 a Shimadzu UV-1800 spectrophotometer. TGA analyses were 634 performed on TGA Q500 V6.3 Build 189 using a heating rate of 635 10 °C min⁻¹ up to 1000 °C under N₂ flow. TEM images were 636 obtained with a Jeol 3010 high resolution electron microscope (1.7 637 nm point-to-point) operating at 300 keV using a Gatan slow-scan 638 CCD camera (mod. 794). TEM samples of protected gold 639 nanoparticles were prepared by placing a single drop of 0.5 mg 640 mL⁻¹ MeOH or H₂O/iPrOH solution onto a 200-mesh copper grid 641 coated with an amorphous carbon film. NP gold core diameters were 642 measured manually using a Gatan software Digital Micrograph on at 643 least 200 particles. Electrospray ionization (ESI) mass analyses were 644 performed on a PerkinElmer APII at 5600 eV and exact mass analyses, 645 on a Bruker Daltonics microTOF-Q operating at 3200 V capillary 646 potential. DLS measurements have been performed on a Malvern zeta 647 Sizer Nano using a concentration for the nanoparticles between 0.1 648 and 0.4 mg/mL in water, scattering angle of 173°, 25 °C, and 649 disposable cuvettes.

Molecular Modeling Methods. A coarse-grained (CG) simu-650 651 lation approach based on dissipative particle dynamics (DPD) was 652 first adopted to retrieve the phase separation of ligands on a gold 653 surface in mixed SAMs, namely, nanoparticles NP1-5/6. This choice 654 was necessary since the self-organization of chains requires long times 655 that cannot be accessed simply by atomistic calculations. Once 656 obtained, the CG nanoparticle model was mapped back onto the 657 corresponding all-atom (AA) nanoparticle structure. Homoligand 658 SAMs were modeled purely at atomic level. The full computational 659 procedure for constructing the CG and AA SAM-functionalized NPs 660 follows our previous works^{21,22,27,29,38} and is described in detail in the 661 Supporting Information. AA nanoparticle models in explicit water 662 were then extracted from equilibrated MD trajectories and used for 663 subsequent MD and SOAP-GMM analysis. CG calculations were 664 carried out in a Culgi simulation package (v.12.0, Culgi B.V., Leiden, 665 The Netherlands) and AA simulations, in an AMBER 18 modeling 666 suite

MD and SOAP-GMM Analysis. MD analysis was generated with 667 668 a combination of an AMBER analysis tool, in-house developed 669 Python codes, and Python package *scipy*.⁵⁸ SOAP descriptors were 670 derived by using the *Dscribe*⁵⁹ Python package. For GMM 671 clusterization and environment classification, we adopted the scikit-672 learn⁶⁰ Python package. The parameter setting is given in the 673 Supporting Information.

Electron Spin Resonance (ESR) Measurements. ESR spectra 674 675 were collected using a Bruker ELEXYS spectrometer equipped with 676 an NMR gaussmeter for field calibration. The sample temperature was 677 controlled with a standard variable temperature accessory and 678 monitored before and after each run using a copper-constantan 679 thermocouple. The instrument settings were as follows: microwave 680 power 5.0 mW, modulation amplitude 0.05 mT, modulation 681 frequency 100 kHz, and scan time 180 s. Digitized EPR spectra 682 were transferred to a personal computer for analysis using digital

simulations carried out with a program developed in our laboratory 683 and based on a Monte Carlo procedure. 684

Synchrotron-Based X-ray Photoelectron Spectroscopy 685 (XPS) Measurements. Synchrotron-based X-ray photoelectron 686 spectroscopy (XPS) experiments were carried out at the Material 687 Science beamline of the Elettra synchrotron radiation facility in 688 Trieste, Italy. The NPs were dispersed in aqueous solution and then 689 drop-casted on a n-doped Si wafer, capped with a layer of native oxide 690 (thickness of the oxide \sim 4 nm). After drying the samples for 24 h in a 691 protected environment at atmospheric pressure, they were inserted in 692 the experimental UHV chamber of the beamline and promptly 693 measured. The base pressure during the experiment was ca. $\hat{2} \times 10^{-10}$ 694 mbar. XPS spectra were acquired by means of a Specs Phoibos 150 695 mm mean-radius electron energy analyzer, equipped with a 1D-delay 696 line detector built in-house. The overall energy resolution of the 697 experiment was ca. 200 meV. The photoelectrons were collected at a 698 normal emission angle, and for each sample measured, the same 699 acquisition conditions (pass energy, entrance slit, lens mode of the 700 spectrometer) were used. The measured signal was normalized to the 701 incoming photon current and to the number of sweeps. The 702 decomposition of the core-level spectra was carried out by using 703 Doniach-Sunjic profiles⁶¹ convoluted with a Gaussian (to take into 704 account the experimental resolution, the thermal effects, and the 705 inhomogeneous broadening) on a linear background, thus obtaining 706 the line shape parameters, the photoemission intensity (i.e., the area 707 delimited by the peak), and the core electron binding energy (BE) for 708 each spectral component. 709

Supporting Information

710 711

718

743

The Supporting Information is available free of charge at 712 https://pubs.acs.org/doi/10.1021/acsnano.2c08467. 713

Additional computational and experimental results, 714 synthesis and characterization of mixed-monolayers 715 nanoparticles, computational methods, ESR background, 716 and details on XPS analysis (PDF) 717

AUTHOR INFORMATION

Corresponding Authors	719
Paola Posocco – Department of Engineering and Architecture,	720
University of Trieste, 34127 Trieste, Italy; [©] orcid.org/	721
0000-0001-8129-1572; Email: paola.posocco@dia.units.it	722
Lucia Pasquato – Department of Chemical and	723
Pharmaceutical Sciences and INSTM Trieste Research Unit,	724
University of Trieste, 34127 Trieste, Italy; <a>[6] orcid.org/	725
0000-0003-1842-9609; Email: l.pasquato@units.it	726
Marco Lucarini – Department of Chemistry "G. Ciamician",	727
University of Bologna, I-40126 Bologna, Italy; 💿 orcid.org/	728
0000-0002-8978-4707; Email: marco.lucarini@unibo.it	729
Authors	730
Cristian Gabellini – Department of Engineering and	731
Architecture, University of Trieste, 34127 Trieste, Italy	732
Maria Sologan – Department of Chemical and	733
Pharmaceutical Sciences and INSTM Trieste Research Unit,	734
University of Trieste, 34127 Trieste, Italy	735
Elena Pellizzoni – Department of Chemical and	736
Pharmaceutical Sciences and INSTM Trieste Research Unit,	737
University of Trieste, 34127 Trieste, Italy	738
Domenico Marson – Department of Engineering and	739
Architecture, University of Trieste, 34127 Trieste, Italy	740
Mario Daka – Department of Chemical and Pharmaceutical	741
Sciences and INSTM Trieste Research Unit, University of	742

Trieste, 34127 Trieste, Italy

- 744 Paola Franchi Department of Chemistry "G. Ciamician",
- 745 University of Bologna, I-40126 Bologna, Italy
- 746 Luca Bignardi Department of Physics, University of Trieste,
- 747 34127 Trieste, Italy; Occid.org/0000-0002-9846-9100
- 748 Stefano Franchi Elettra Sincrotrone Trieste, 34149 Trieste,
 749 Italy;

 orcid.org/0000-0002-5009-9147
- 750 Zbyšek Posel Department of Informatics, Jan Evangelista
- 751 Purkyně University, 400 96 Ústí nad Labem, Czech
- 752 *Republic;* ^(b) orcid.org/0000-0003-4271-5349
- 753 Alessandro Baraldi Department of Physics, University of
- 754 Trieste, 34127 Trieste, Italy
- 755 Paolo Pengo Department of Chemical and Pharmaceutical

Sciences and INSTM Trieste Research Unit, University of
 Trieste, 34127 Trieste, Italy

758 Complete contact information is available at:

759 https://pubs.acs.org/10.1021/acsnano.2c08467

760 Notes

761 The authors declare no competing financial interest.

762 ACKNOWLEDGMENTS

763 This work received support from the Italian Ministry of 764 University Research through the projects "Structure and 765 function at the nanoparticle biointerface" (RBSI14PBC6 to 766 P. Posocco), PRIN2017 NiFTy (2017MYBTXC to L.P.), 767 PRIN2017 "BacHounds: Supramolecular nanostructures for 768 bacteria detection" (2017E44A9P to M.L.), and "Nemo" 769 (20173L7W8K to P.F.). P. Pengo and C.G. are particularly 770 grateful to the University of Trieste for scholarship support and 771 acknowledge the CINECA award under the ISCRA initiative 772 for the availability of high performance computing resources. 773 CERIC-ERIC consortium is acknowledged for the access to 774 the Material Science beamline at the Elettra synchrotron 775 radiation facility (proposal number 20192081). The staff of the 776 Material Science beamline is kindly acknowledged for technical 777 support. Z.P. acknowledges the assistance provided by the 778 Technology Agency of the Czech Republic, under the project 779 Metamorph, project No. TO01000329.

780 REFERENCES

- (1) Goodman, E. D.; Zhou, C.; Cargnello, M. Design of organic/
 inorganic hybrid catalysts for energy and environmental applications.
 ACS Cent. Sci. 2020, 6, 1916–1937.
- 784 (2) Prins, L. J. Emergence of complex chemistry on an organic 785 monolayer. *Acc. Chem. Res.* 2015, 48, 1920–1928.
- (3) Sun, X.; Riccardi, L.; De Biasi, F.; Rastrelli, F.; De Vivo, M.; et al. 787 Molecular-dynamics-simulation-directed rational design of nano-788 receptors with targeted affinity. *Angew. Chem., Int. Ed.* **2019**, *58*, 789 7702–7707.
- 790 (4) Zeiri, O. Metallic-nanoparticle-based sensing: Utilization of 791 mixed-ligand monolayers. *ACS Sens.* **2020**, *5*, 3806–3820.
- 792 (5) Grommet, A. B.; Feller, M.; Klajn, R. Chemical reactivity under 793 nanoconfinement. *Nat. Nanotechnol.* **2020**, *15*, 256–271.
- 794 (6) Zhu, Q.; Murphy, C. J.; Baker, L. R. Opportunities for 795 electrocatalytic CO_2 reduction enabled by surface ligands. *J. Am.* 796 Chem. Soc. **2022**, 144, 2829–2840.
- (7) Chu, Z.; Han, Y.; Bian, T.; De, S.; Král, P.; et al. Supramolecular
 control of azobenzene switching on nanoparticles. *J. Am. Chem. Soc.* **2019**, *141*, 1949–1960.

800 (8) Szewczyk, M.; Sobczak, G.; Sashuk, V. Photoswitchable catalysis 801 by a small swinging molecule confined on the surface of a colloidal 802 particle. *ACS Catal.* **2018**, *8*, 2810–2814. (9) Mati, I. K.; Edwards, W.; Marson, D.; Howe, E. J.; Stinson, S.; 803 Kay, E. R.; et al. Probing multiscale factors affecting the reactivity of 804 nanoparticle-bound molecules. *ACS Nano* **2021**, *15*, 8295–8305. 805 (10) Kim, M.; Dygas, M.; Sobolev, Y. I.; Beker, W.; Zhuang, Q.; 806 Grzybowski, B. A.; et al. On-nanoparticle gating units render an 807

ordinary catalyst substrate- and site-selective. J. Am. Chem. Soc. 2021, 808 143, 1807–1815. 809 (11) Cha, M.; Emre, E. S. T.; Xiao, X.; Kim, J.-Y.; Bogdan, P.; et al. 810

Unifying structural descriptors for biological and bioinspired nano- 811 scale complexes. *Nat. Comput. Sci.* **2022**, *2*, 243–252. 812

(12) Siek, M.; Kandere-Grzybowska, K.; Grzybowski, B. A. Mixed- 813 charge, pH-responsive nanoparticles for selective interactions with 814 cells, organelles, and bacteria. *Acc. Mater. Res.* **2020**, *1*, 188–200. 815 (13) Riccardi, L.; Gabrielli, L.; Sun, X.; De Biasi, F.; Rastrelli, F.; 816

et al. Nanoparticle-based receptors mimic protein-ligand recognition. 817 Chem. 2017, 3, 92–109. 818

(14) Pecina, A.; Rosa-Gastaldo, D.; Riccardi, L.; Franco-Ulloa, S.; 819 Milan, E.; et al. On the metal-aided catalytic mechanism for 820 phosphodiester bond cleavage performed by nanozymes. *ACS Catal.* 821 **2021**, *11*, 8736–8748. 822

(15) Cao-Milán, R.; Gopalakrishnan, S.; He, L. D.; Huang, R.; 823 Wang, L.-S.; et al. Thermally gated bio-orthogonal nanozymes with 824 supramolecularly confined porphyrin catalysts for antimicrobial uses. 825 *Chem.* **2020**, *6*, 1113–1124. 826

(16) Zhang, X.; Huang, R.; Gopalakrishnan, S.; Cao-Milán, R.; 827 Rotello, V. M. Bioorthogonal nanozymes: Progress towards 828 therapeutic applications. *Trends Chem.* **2019**, *1*, 90–98. 829

(17) Cao-Milán, R.; He, L. D.; Shorkey, S.; Tonga, G. Y.; Wang, L.- 830 S.; et al. Modulating the catalytic activity of enzyme-like nanoparticles 831 through their surface functionalization. *Mol. Syst. Des. Eng.* **2017**, *2*, 832 624–628. 833

(18) Huang, R.; Luther, D. C.; Zhang, X.; Gupta, A.; Tufts, S. A.; 834 et al. Engineering the interface between inorganic nanoparticles and 835 biological systems through ligand design. *Nanomaterials* **2021**, *11*, 836 1001. 837

(19) Wu, M.; Vartanian, A. M.; Chong, G.; Pandiakumar, A. K.; 838 Hamers, R. J.; et al. Solution NMR analysis of ligand environment in 839 quaternary ammonium-terminated self-assembled monolayers on gold 840 nanoparticles: The effect of surface curvature and ligand structure. J. 841 Am. Chem. Soc. **2019**, 141, 4316–4327. 842

(20) Liu, X.; Yu, M.; Kim, H.; Mameli, M.; Stellacci, F. 843 Determination of monolayer-protected gold nanoparticle ligand– 844 shell morphology using NMR. *Nat. Commun.* **2012**, *3*, 1182. 845

(21) Sologan, M.; Marson, D.; Polizzi, S.; Pengo, P.; Boccardo, S.; 846 et al. Patchy and Janus nanoparticles by self-organization of mixtures 847 of fluorinated and hydrogenated alkanethiolates on the surface of a 848 gold core. *ACS Nano* **2016**, *10*, 9316–9325. 849

(22) Luo, Z.; Marson, D.; Ong, Q. K.; Loiudice, A.; Kohlbrecher, J.; 850 et al. Quantitative 3D determination of self-assembled structures on 851 nanoparticles using small angle neutron scattering. *Nat. Commun.* 852 **2018**, *9*, 1343. 853

(23) Luo, Z.; Zhao, Y.; Darwish, T.; Wang, Y.; Hou, J.; et al. Mass 854 spectrometry and Monte Carlo method mapping of nanoparticle 855 ligand shell morphology. *Nat. Commun.* **2018**, *9*, 4478. 856

(24) Lucarini, M.; Franchi, P.; Pedulli, G. F.; Pengo, P.; Scrimin, P.; 857 et al. EPR study of dialkyl nitroxides as probes to investigate the 858 exchange of solutes between the ligand shell of monolayers of 859 protected gold nanoparticles and aqueous solutions. *J. Am. Chem. Soc.* 860 **2004**, *126*, 9326–9329. 861

(25) Lucarini, M.; Pasquato, L. ESR spectroscopy as a tool to 862 investigate the properties of self-assembled monolayers protecting 863 gold nanoparticles. *Nanoscale* **2010**, *2*, 668–676. 864

(26) Posocco, P.; Gentilini, C.; Bidoggia, S.; Pace, A.; Franchi, P.; 865 et al. Self-organization of mixtures of fluorocarbon and hydrocarbon 866 amphiphilic thiolates on the surface of gold nanoparticles. *ACS Nano* 867 **2012**, *6*, 7243–7253. 868

(27) Marson, D.; Posel, Z.; Posocco, P. Molecular features for 869 probing small amphiphilic molecules with self-assembled monolayer- 870 protected nanoparticles. *Langmuir* **2020**, *36*, 5671–5679. 871

872 (28) Pellizzoni, E.; Şologan, M.; Daka, M.; Pengo, P.; Marson, D.; 873 et al. Thiolate end-group regulates ligand arrangement, hydration and 874 affinity for small compounds in monolayer-protected gold nano-875 particles. J. Colloid Interface Sci. **2022**, 607, 1373–1381.

876 (29) Marson, D.; Guida, F.; Şologan, M.; Boccardo, S.; Pengo, P.; 877 et al. Mixed fluorinated/hydrogenated self-assembled monolayer-878 protected gold nanoparticles: *In silico* and *in vitro* behavior. *Small* 879 **2019**, *15*, 1900323.

(30) Musil, F.; Grisafi, A.; Bartók, A. P.; Ortner, C.; Csányi, G.; et al.
Physics-inspired structural representations for molecules and materials. *Chem. Rev.* 2021, *121*, 9759–9815.

883 (31) Gasparotto, P.; Meißner, R. H.; Ceriotti, M. Recognizing local 884 and global structural motifs at the atomic scale. *J. Chem. Theory* 2018, 885 14, 486–498.

(32) Shyshov, O.; Haridas, S. V.; Pesce, L.; Qi, H.; Gardin, A.; et al.
supramolecular polymerization of fluorinated cyclohexanes. *Nat. Commun.* 2021, *12*, 3134.

(33) Ofir, Y.; Samanta, B.; Arumugam, P.; Rotello, V. M. Controlled
fluorination of FePt nanoparticles: Hydrophobic to superhydrophobic
surfaces. Adv. Mater. 2007, 19, 4075–4079.

892 (34) Marsh, Z. M.; Lantz, K. A.; Stefik, M. QCM detection of 893 molecule–nanoparticle interactions for ligand shells of varying 894 morphology. *Nanoscale* **2018**, *10*, 19107–19116.

895 (35) Elbert, K. C.; Jishkariani, D.; Wu, Y.; Lee, J. D.; Donnio, B.;
896 et al. Design, self-assembly, and switchable wettability in hydrophobic,
897 hydrophilic, and Janus dendritic ligand–gold nanoparticle hybrid
898 materials. *Chem. Mater.* 2017, 29, 8737–8746.

899 (36) Basham, C. M.; Premadasa, U. I.; Ma, Y.-Z.; Stellacci, F.; 900 Doughty, B.; et al. Nanoparticle-induced disorder at complex liquid— 901 liquid interfaces: Effects of curvature and compositional synergy on 902 functional surfaces. *ACS Nano* **2021**, *15*, 14285–14294.

903 (37) Pan, S.; Richardson, J. J.; Christofferson, A. J.; Besford, Q. A.; 904 Zheng, T.; et al. Fluorinated metal-organic coatings with selective 905 wettability. *J. Am. Chem. Soc.* **2021**, *143*, 9972–9981.

906 (38) Edwards, W.; Marro, N.; Turner, G.; Kay, E. R. Continuum 907 tuning of nanoparticle interfacial properties by dynamic covalent 908 exchange. *Chem. Sci.* **2018**, *9*, 125–133.

909 (39) Stewart, A.; Zheng, S.; McCourt, M. R.; Bell, S. E. J. 910 Controlling assembly of mixed thiol monolayers on silver nano-911 particles to tune their surface properties. *ACS Nano* **2012**, *6*, 3718– 912 3726.

913 (40) Pengo, P.; Şologan, M.; Pasquato, L.; Guida, F.; Pacor, S.; et al. 914 Gold nanoparticles with patterned surface monolayers for nano-915 medicine: Current perspectives. *Eur. Biophys. J.* **2017**, *46*, 749–771.

916 (41) Luo, Z.; Hou, J.; Menin, L.; Ong, Q. K.; Stellacci, F. Evolution 917 of the ligand shell morphology during ligand exchange reactions on 918 gold nanoparticles. *Angew. Chem., Int. Ed.* **2017**, *56*, 13521–13525.

919 (42) Singh, C.; Ghorai, P. K.; Horsch, M. A.; Jackson, A. M.; Larson, 920 R. G.; et al. Entropy-mediated patterning of surfactant-coated 921 nanoparticles and surfaces. *Phys. Rev. Lett.* **2007**, *99*, 226106.

922 (43) Bock, M.; Tyagi, A. K.; Kreft, J.-U.; Alt, W. Generalized 923 Voronoi tessellation as a model of two-dimensional cell tissue 924 dynamics. *Bull. Math. Biol.* **2010**, *72*, 1696–1731.

925 (44) Liang, D.; Dahal, U.; Wu, M.; Murphy, C. J.; Cui, Q. Ligand
926 length and surface curvature modulate nanoparticle surface hetero927 geneity and electrostatics. *J. Phys. Chem. C* 2020, *124*, 24513–24525.
928 (45) Kelkar, A. S.; Dallin, B. C.; Lehn, R. C. V. Identifying
929 nonadditive contributions to the hydrophobicity of chemically
930 heterogeneous surfaces via dual-loop active learning. *J. Chem. Phys.*931 2022, *156*, 024701.

932 (46) Chew, A. K.; Dallin, B. C.; Van Lehn, R. C. The interplay of 933 ligand properties and core size dictates the hydrophobicity of 934 monolayer-protected gold nanoparticles. *ACS Nano* **2021**, *15*, 935 4534–4545.

936 (47) Hoff, S. E.; Di Silvio, D.; Ziolo, R. F.; Moya, S. E.; Heinz, H. 937 Patterning of self-assembled monolayers of amphiphilic multisegment 938 ligands on nanoparticles and design parameters for protein 939 interactions. *ACS Nano* **2022**, *16*, 8766–8783. (48) Luo, Z.; Murello, A.; Wilkins, D. M.; Kovacik, F.; Kohlbrecher, 940 J.; et al. Determination and evaluation of the nonadditivity in wetting 941 of molecularly heterogeneous surfaces. *Proc. Natl. Acad. Sci. U. S. A.* 942 **2019**, *116*, 25516–25523. 943

(49) Bartók, A. P.; Kondor, R.; Csányi, G. On representing chemical 944 environments. *Phys. Rev. B* 2013, 87, 184115. 945

(50) Musil, F.; De, S.; Yang, J.; Campbell, J. E.; Day, G. M.; et al. 946 Machine learning for the structure–energy–property landscapes of 947 molecular crystals. *Chem. Sci.* **2018**, *9*, 1289–1300. 948

(51) Bartók, A. P.; De, S.; Poelking, C.; Bernstein, N.; Kermode, J. 949 R.; et al. Machine learning unifies the modeling of materials and 950 molecules. *Sci. Adv.* **2017**, *3*, e1701816. 951

(52) de Marco, A. L.; Bochicchio, D.; Gardin, A.; Doni, G.; Pavan, 952 G. M. Controlling exchange pathways in dynamic supramolecular 953 polymers by controlling defects. *ACS Nano* **2021**, *15*, 14229–14241. 954

(53) De, S.; Bartók, A. P.; Csányi, G.; Ceriotti, M. Comparing 955 molecules and solids across structural and alchemical space. *Phys.* 956 *Chem. Chem. Phys.* **2016**, *18*, 13754–13769. 957

(54) Ionita, P.; Caragheorgheopol, A.; Gilbert, B. C.; Chechik, V. 958 EPR study of a place-exchange reaction on Au nanoparticles: Two 959 branches of a disulfide molecule do not adsorb adjacent to each other. 960 J. Am. Chem. Soc. **2002**, 124, 9048–9049. 961

(55) Lucarini, M.; Franchi, P.; Pedulli, G. F.; Gentilini, C.; Polizzi, 962 S.; et al. Effect of core size on the partition of organic solutes in the 963 monolayer of water-soluble nanoparticles: An ESR investigation. *J.* 964 *Am. Chem. Soc.* **2005**, *127*, 16384–16385. 965

(56) Gentilini, C.; Evangelista, F.; Rudolf, P.; Franchi, P.; Lucarini, 966 M.; et al. Water-soluble gold nanoparticles protected by fluorinated 967 amphiphilic thiolates. *J. Am. Chem. Soc.* **2008**, *130*, 15678–15682. 968

(57) Gentilini, C.; Franchi, P.; Mileo, E.; Polizzi, S.; Lucarini, M.; 969 et al. Formation of patches on 3D sams driven by thiols with 970 immiscible chains observed by ESR spectroscopy. *Angew. Chem., Int.* 971 *Ed.* **2009**, *48*, 3060–3064. 972

(58) Virtanen, P.; Gommers, R.; Oliphant, T. E.; Haberland, M.; 973 Reddy, T.; et al. Scipy 1.0: Fundamental algorithms for scientific 974 computing in Python. *Nat. Methods* **2020**, *17*, 261–272. 975

(59) Himanen, L.; Jäger, M. O. J.; Morooka, E. V.; Federici Canova, 976 F.; Ranawat, Y. S.; et al. Dscribe: Library of descriptors for machine 977 learning in materials science. *Comput. Phys. Commun.* **2020**, 247, 978 106949. 979

(60) Pedregosa, F.; Varoquaux, G.; Gramfort, A.; Michel, V.; 980 Thirion, B.; et al. Scikit-learn: Machine learning in Python. J. Mach. 981 Learn. Res. 2011, 12, 2825–2830. 982

(61) Doniach, S.; Sunjic, M. Many-electron singularity in X-ray 983 photoemission and X-ray line spectra from metals. J. Phys. C Solid 984 State Phys. 1970, 3, 285–291. 985