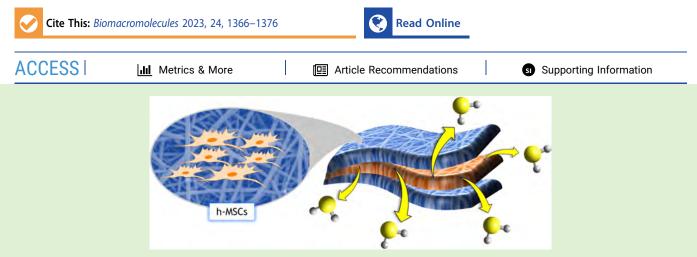


Controlled Release of H₂S from Biomimetic Silk Fibroin–PLGA Multilayer Electrospun Scaffolds

Anna Liguori,[∇] Elisabetta Petri,[∇] Chiara Gualandi, Luisa S. Dolci, Valentina Marassi, Mauro Petretta, Andrea Zattoni, Barbara Roda, Brunella Grigolo, Eleonora Olivotto, Francesco Grassi, and Maria Letizia Focarete^{*}



ABSTRACT: The possibility of incorporating H₂S slow-release donors inside biomimetic scaffolds can pave the way to new approaches in the field of tissue regeneration and anti-inflammatory treatment. In the present work, GYY4137, an easy-to-handle commercially available Lawesson's reagent derivative, has been successfully incorporated inside biomimetic silk fibroin-based electrospun scaffolds. Due to the instability of GYY4137 in the solvent needed to prepare silk fibroin solutions (formic acid), the electrospinning of the donor together with the silk fibroin turned out to be impossible. Therefore, a multilayer structure was realized, consisting of a PLGA mat containing GYY4137 sandwiched between two silk fibroin nanofibrous layers. Before their use in the multilayer scaffold, the silk fibroin mats were treated in ethanol to induce crystalline phase formation, which conferred waterresistance and biomimetic properties. The morphological, thermal, and chemical properties of the obtained scaffolds were thoroughly characterized by SEM, TGA, DSC, FTIR, and WAXD. Multilayer devices showing two different concentrations of the H₂S donor, i.e., 2 and 5% w/w with respect to the weight of PLGA, were analyzed to study their H₂S release and biological properties, and the results were compared with those of the sample not containing GYY4137. The H₂S release analysis was carried out according to an "ad-hoc" designed procedure based on a validated high-performance liquid chromatography method. The proposed analytical approach demonstrated the slow-release kinetics of H₂S from the multilayer scaffolds and its tunability by acting on the donor's concentration inside the PLGA nanofibers. Finally, the devices were tested in biological assays using bone marrowderived mesenchymal stromal cells showing the capacity to support cell spreading throughout the scaffold and prevent cytotoxicity effects in serum starvation conditions. The resulting devices can be exploited for applications in the tissue engineering field since they combine the advantages of controlled H₂S release kinetics and the biomimetic properties of silk fibroin nanofibers.

■ INTRODUCTION

Hydrogen sulfide (H_2S) has been considered a toxic gas with a noxious odor for many years. Although the presence of H_2S inside the mammalian tissues was already known, only in 1996 the endogenous production and signaling of this compound were elucidated, leading to its introduction into the family of gasotransmitters.¹ H_2S has been demonstrated to play a relevant role in regulating inflammatory processes, in the homeostasis of different tissues and organs,² as a vasodilator substance,^{3,4} and in the regulation of angiogenesis and osteogenesis.^{5,6} In particular, H_2S has been demonstrated to drive the capacity of self-renewal and multilineage differ-

entiation into osteoblasts, chondrocytes, myocytes, and adipocytes of mesenchymal stromal cells (MSCs).²

Considering the therapeutic efficacy of H_2S , several approaches have been investigated for its exogenous delivery, spanning from direct delivery methods (i.e., inhalation of gaseous H_2S and introduction of sulfide salts, such as NaHS)

Received:November 20, 2022Revised:January 26, 2023Published:February 7, 2023



Downloaded via UNIV BOLOGNA on May 26, 2023 at 07:05:21 (UTC). See https://pubs.acs.org/sharingguidelines for options on how to legitimately share published articles. to the use of H₂S donors. Compared to the direct sources, the H₂S donors enable a more controllable and prolonged release over time. Indeed, they need to be subjected to a chemical reaction under the action of a specific stimulus (i.e., water, light, pH, etc.) to release H₂S. The control of the release kinetics using a reliable method is extremely important to modulate the stimulus correctly and avoid undesired toxic effects. Commonly employed methods to assess the release kinetics of H₂S consist of (i) the use of selective electrochemical probes;^{7,8} (ii) the formation of methylene blue from the reaction of sulfide species in an acidic aqueous solution of N,N-dimethylphenylenediamine and iron (III) chloride $(FeCl_3)$;⁹ and (iii) the employment of selective fluorescent probes.^{10,11} In the past few years, there has been considerable interest in methods based on monobromobimane (MBB) for the derivatization and quantification of sulfide in solutions.¹²⁻¹⁵ However, complications related to the quantification of H₂S can occur during laboratory procedures, mainly due to the quick oxidation of sulfide when exposed to air.⁹ Therefore, to avoid any undesired oxidation, strategies based on the use of chelating agents or the preparation of the solutions in the absence of oxygen are required.

Several synthetic H₂S donors have been investigated over the years for therapeutic applications. Some of them have also been incorporated in hydrogels^{16,17} and polymeric scaffolds^{18–22} to achieve a long-lasting release of H_2S , allowing local delivery at sites of tissue injury. Mauretti et al. proposed a poly(ethylene glycol)-fibrinogen hydrogel (PFHy) loaded with air or perfluorohexane-filled bovine serum albumin microbubbles coated with a TST enzyme able to catalyze H₂S production.¹⁶ Wu et al. reported the development of a hyaluronic acid hydrogel doped with JK1, a hydrolysistriggered pH-controllable donor, for dermal wound healing.¹ This compound was also electrospun in a poly(caprolactone) (PCL) solution to obtain nanofibrous scaffolds showing a pHdependent H₂S releasing behavior for applications in wound dressing.¹⁸ Electrospun scaffolds of PCL and poly(L-lactic acid) for the release of N-(benzoylthio)benzamide derivatives and garlic-derived H₂S donors, respectively, were also realized.¹⁹ In all of these studies, the methylene blue method was employed to determine the in vitro H₂S release kinetics. Recently, Zhang et al. proposed the development of large porous microspheres (LPMs) containing a H₂S-releasing aspirin derivative (ACS14), a novel synthetic H₂S donor belonging to the family of dithiolthiones, for the treatment of pulmonary arterial hypertension. The H₂S release kinetics was investigated in vitro by measuring the release of ACS14 and converting the moles of ACS14 into moles of H₂S. The device was tested for the H2S release in vivo in rats' lung tissue homogenates and plasma.²⁰

Among H_2S donors, the hydrolysis-triggered ones, i.e., Lawesson's reagent derivatives and dithiolthiones, have been widely reported. GYY4137 (GYY) belongs to the class of Lawesson's reagent derivatives, and it is the most commonly studied synthetic H_2S donor, mainly due to its commercial availability and ease of handling. Furthermore, this compound is generally regarded as a slow-releasing H_2S donor, showing release kinetics significantly lower than NaHS^{23,24} and a hydrolysis pathway occurring through a two-step process, as carefully described by Alexander et al.²⁵ Focusing on the incorporation of GYY inside scaffolds, Patil et al. developed a nonaqueous *in situ* gelling sustained-release delivery system obtained by dissolving GYY in the poly(lactide-*co*-glycolide) (PLGA) solution prepared in a mixture of benzyl alcohol and benzyl benzoate.²¹ The use of a polymer soluble in nonaqueous solvents was necessary due to aqueous GYY instabilities to prevent compound's hydrolysis. The resulting hydrogel, envisioned for the reduction of intraocular pressure in glaucoma pathogenesis, ensured a sustained release of H₂S for 72 h, evaluated with the ethylene blue method, and did not show any significant toxicity. Raggio et al. developed silk fibroin sponges loaded with GYY by solvent casting and particulate leaching.²² GYY was incorporated using dimethyl sulfoxide as a vehicle. The H₂S release kinetics was investigated employing an electrochemical method, using a sulfide gas amperometric microsensor. The results demonstrated that the scaffold did not induce cytotoxicity in any tested cells.

In the context of scaffolds for tissue engineering, the use of silk fibroin (SF) has been widely proposed for several applications. SF shows superior biocompatibility, controllable biodegradation, and attractive mechanical properties thanks to a balance of modulus, breaking strength, and elongation, which contributes to its toughness and ductility.²⁶ Furthermore, SF scaffolds have been demonstrated to stimulate human osteoblast-like cell attachment, growth, and proliferation;²⁷ to be suitable for tissue regeneration, including ligament, tendon, cartilage, bone, liver, skin, trachea, cornea, nerve, eardrum, and bladder;^{28,29} to be endowed with osteoinductive properties when loaded with recombinant human bone morphogenic protein-2 (rhBMP2).³⁰ Among the native silk proteins, the silkworm silk, primarily that of the domesticated Bombyx mori (B. mori), has been recognized as a high-quality textile fiber and suture. The SF, obtained from B. mori silk fibers through specific extraction protocols,³¹ has also been widely employed to fabricate electrospun scaffolds for tissue engineering. Although some studies documented the possibility to electrospin SF from water solution,^{4,32} the majority of the works proposed the use of formic acid as a solvent³³⁻³⁶ since it enables the rapid solubilization of the SF obtained from the extraction process without inducing any degradation during the experimental period.³³ In the context of bone tissue engineering, the use of electrospinning to produce scaffolds capable of stimulating osteogenesis finds a clinical application in guided bone regeneration (GBR), a procedure by which scaffolds are used at once to exclude non-osteogenic tissues from interfering with bone regeneration and to actively promote osteogenesis.³⁷ Combining the osteogenic properties of H₂S-releasing materials with biocompatible and clinically manageable scaffolds appears to be an attractive perspective to improve GBR-based bone regeneration.

Among the synthetic biodegradable and biocompatible polymers, PLGA is widely investigated and approved by the Food and Drug Administration (FDA) for therapeutic device development, spanning from sutures to tissue regeneration. PLGA is largely employed to obtain electrospun scaffolds for biomedical applications.³⁸ The suitability of PLGA nanofibrous scaffolds as a drug delivery vehicle has also been documented, and it exploits the possibility of tailoring the nanofibers' morphology by acting on process parameters to optimize the incorporation of drugs and their release from the nanofibers.^{39,40} The combination of PLGA and SF for obtaining electrospun scaffolds has allowed the development of invaluable materials for biomedical applications. In the context of wound treatment, the potentiality of PLGA/SF electrospun mats, produced with the technique of dual-source electrospinning, was investigated both in vitro and in vivo, confirming

the most prominent wound healing effect of the bicomponent polymeric scaffolds compared to both the single components.⁴¹ The usefulness of this polymeric combination has also been investigated for bone tissue regeneration: PLGA/SF nanofiber scaffolds containing recombinant human bone morphogenetic protein-2 and dexamethasone were obtained via coaxial electrospinning. The devices were employed for in vitro bone formation with rat bone marrow mesenchymal stem cells and turned out to enable the sustained release of the two molecules, promoting cell adhesion and proliferation.⁴²

In the present work, we have developed a biomimetic scaffold characterized by a nanofibrous and microporous morphology, able to release H_2S in a controlled manner. The proposed scaffold has a multilayer architecture consisting of two external layers of electrospun silk to endow the scaffold with superior biomimetic properties and an internal layer of PLGA nanofibers incorporating GYY to achieve slow-release kinetics of H_2S . The release of H_2S was determined by applying an optimized analytical method based on the derivatization of MBB and high-performance liquid chromatography with fluorescence detection (HPLC-FLD). This method has shown high sensitivity and a limit of detection of 0.5 μ M for the quantification of H_2S species in serum samples,⁴³ and it was successfully implemented for the kinetic study of H_2S release.

MATERIALS AND METHODS

Materials. Natural silk fibroin extracted from Bombyx Mori cocoon, purchased by Chul Thai Silk Co., Phetchabun, Thailand, was used. Poly(D,L lactide-co-glycolide) (PLGA), lactide/glycolide = 75:25 molar ratio (average molecular weight by GPC Mw = 50,900 g mol^{-1} , polydispersity index, PDI = 1.55), was supplied by Evonik. H₂S donor GYY (MW = $376.47 \text{ g mol}^{-1}$), supplied as the morpholinium salt of (4-methoxyphenyl) morpholino-phosphinodithioic acid, was purchased from Cayman Chemical (Ann Arbor, MI). Formic acid (FA), dichloromethane (DCM), N,N-dimethylformamide (DMF), and ethanol (EtOH) were purchased from Sigma-Aldrich. Anhydrous sodium sulfide (Na2S, Sigma-Aldrich), monobromobimane (MBB, Sigma-Aldrich), tris(hydroxymethyl)aminomethane (Sigma-Aldrich), trifluoroacetic acid (TFA, Sigma-Aldrich), 5-sulfosalicylic acid dihydrate (SSA, Sigma-Aldrich), potassium dihydrogen phosphate (KH₂PO₄, Fluka Chemie GmbH), diethylenetriaminepentaacetic acid (DTPA, Sigma-Aldrich), and acetonitrile (Sigma-Aldrich) were used. Purified deionized water used throughout the study was obtained from a classic purification Millipore Milli-Q system (ELGA LC134, 0.2-micron filter, 18.5 m Ω cm⁻¹, Milford, MA). Unless otherwise specified, all chemicals and solvents were used without further purifications.

Preparation of Electrospun Solutions. *Silk Fibroin Solution.* Native *B. mori* cocoons were treated according to a previously reported protocol.²² Briefly, cocoons were degummed through two consecutive treatments in a 0.02 M Na₂CO₃ aqueous solution at 100 °C for 20 min each time. The obtained fibers were rinsed six times with warm ultrapure water and dried at 37 °C overnight. The degummed SF (Deg-SF) was dissolved at a concentration of 20% w/v in 9.3 M LiBr solution at 65 °C for 4 h. Afterward, the solution was dialyzed in a Slide-A-Lyzer Dialysis Cassette with 3500 MWCO (Thermo Fisher Scientific, Waltham, MA) against ultrapure water for two days, with regular water changes. The resulting silk fibroin solution was dried at 100 °C for 6 h, and a film was obtained. The electrospinning solution was finally obtained by dissolving the film in 98% v/v FA for 1 h at a concentration of 18% w/v.

PLGA/GYY solutions. PLGA/GYY solutions were prepared starting from a PLGA solution with a concentration of 33% w/v in DCM/ DMF 25:75 v/v and GYY concentration of either 2% w/w or 5% w/w with respect to the weight of PLGA. The following procedure was employed: (i) GYY was dissolved in DMF, (ii) PLGA was slowly added to the solution, and (iii) after PLGA solubilization, DCM was added. The resulting solutions were kept for 20 min under stirring before electrospinning. PLGA solution, not containing GYY, was also prepared.

Preparation of Electrospun Mats and Multilayer Scaffolds. Electrospun mats were prepared using an in-house electrospinning apparatus composed of a high-voltage power supply (Spellman, SL 50 P 10/CE/230), a syringe pump (KD Scientific 200 series, Massachusetts), a glass syringe, a stainless-steel blunt-ended needle connected with the power supply electrode, and a grounded steel plate collector ($6.5 \times 6.5 \text{ cm}^2$). The entire system was located inside a glovebox (Iteco Eng., Ravenna, Italy, $100 \times 75 \times 100 \text{ cm}^3$) equipped with a temperature and humidity control system. The polymer solution was dispensed through a Teflon tube to the needle orthogonally positioned with respect to the steel collector. The electrospinning of the silk fibroin solution was performed at a temperature and relative humidity of 25 °C and 30%, respectively, with a solution flow rate of 3 mL h^{-1} , an applied voltage of 22 kV, and a gap of 20 cm between the needle outlet and the collector. Nanometric fibers with a random arrangement were collected, and the resulting silk fibroin mat was labeled SF-mat. The post-treatment of SF-mat was carried out by immersing it in absolute ethanol for 15, 30, and 60 min. Samples showing a thickness of around 135 μm and named SF-mat15, SF-mat30, and SF-mat60, respectively, were obtained.

Nonwoven PLGA/GYY mats were prepared from the two electrospinning solutions described above, containing different concentrations of GYY (2 and 5% w/w), and were labeled PLGA-2GYY and PLGA-5GYY, respectively. The electrospinning process was carried out at a temperature of 25 °C and a relative humidity of 40%, using an applied voltage of 20–22 kV, a solution flow rate of 6–8 mL h⁻¹, and a gap between the needle and the collector of 20 cm. Electrospun scaffolds were kept under vacuum at room temperature (RT) for 30 min to remove residual solvents. Mats of plain PLGA, not containing GYY, were also produced as a reference and labeled PLGA. PLGA, PLGA-2GYY, and PLGA-5GYY showed a thickness of around 200, 190, and 120 μ m, respectively.

Multilayer electrospun scaffolds were obtained by assembling a PLGA-based fibrous mat, sandwiched between two layers of SFmat15, directly on a Scaffdex support (CellCrown24NX inserts). Multilayer samples containing GYY concentrations of 2 and 5% w/w were labeled ML2% and ML5%, respectively, whereas the reference multilayer sample without GYY is named ML0. Before biological experiments, the assembled scaffolds were sterilized using γ -rays (25 kGy).

Chromatographic Method to Detect the Released H₂S. The ad-hoc developed procedure for the H2S release study from the assembled scaffolds is schematically illustrated in Figure 4. The H₂Sreleasing tests were performed on multilayer scaffolds incubated in aqueous 0.1 M phosphate buffer (PB) (pH adjusted to 7.4 with 37% HCl at RT) in a thermostatic shaking bath at 37 °C. The H₂S release was measured using the MBB method coupled with HPLC-FLD.⁴³ The release of hydrogen sulfide in the incubation medium was monitored for 7 days at the following sampling time: 0, 2, 5, 19, 23, 28, 47, 72, 96, and 168 h. Three replicates for each sample were studied, and two aliquots for sampling time were examined to have statistical data. Quantitative determinations were carried out by peak area measurements at the emission wavelength of the SDB derivatization product after interpolation in a calibration curve prepared with Na2S standard solutions in PB (Supporting Information).

Characterization methods. The morphological analysis of the electrospun mats was carried out using a Scanning Electron Microscope (SEM, Leica Cambridge Stereoscan 360) at an accelerating voltage of 20 kV. Prior to SEM analysis, the samples were sputter-coated with gold. The distribution of fiber diameters was determined through the measurement of about 300 fibers employing ImageJ software, and the results were given as the average diameter \pm standard deviation. The Student's unpaired *t*-test was used to test the statistical significance of the difference between the mean values (p <

pubs.acs.org/Biomac

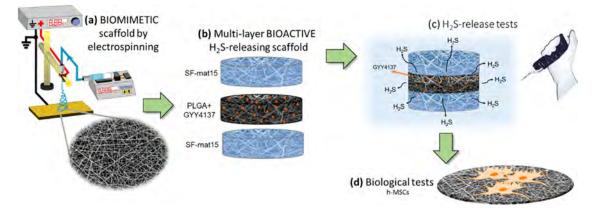


Figure 1. Schematic representation of the steps of the work: (a) production of fibrous biomimetic scaffolds of silk fibroin and PLGA by electrospinning, (b) preparation of multilayer bioactive scaffold releasing H_2S , composed of an inner layer of PLGA loaded with GYY sandwiched between silk fibroin mats, (c) H_2S release tests *in vitro*, and (d) h-MSC culture onto scaffolds.

0.05). Differential scanning calorimetry (DSC) measurements were carried out on SF-mat and SF-mat15 using Q2000 DSC (TA instruments, Delaware) in a N_2 atmosphere from -90 to 250 °C, with a heating scan of 20 °C min⁻¹. The glass transition temperature (Tg) was taken at half-height of the glass transition heat capacity step in the second heating scan performed after quenching. Thermogravimetric analysis (TGA) was conducted using a TA Instrument TGA Q500 analyzer in a N2 atmosphere by applying a temperature ramp of 10 °C min⁻¹ from RT to 700 °C. Fourier transform infrared spectroscopy (ATR-FTIR) was carried out using a Spectrum Two (PerkinElmer) apparatus in attenuated total reflection mode. Each spectrum was collected in the wavenumber range 4000–400 cm⁻¹, with a resolution of 4 cm⁻¹ and 64 signal accumulations. Wide-angle X-ray diffraction (WAXD) analysis was performed directly on silk fibroin electrospun mats and SF-mat after the ethanol treatment deposited on quartz glass. The diffractogram patterns were recorded in the 5–60° 2θ range and a step rate of 0.03, with an X'Celerator detector at 40 and 40 kA, using a PANalytical X'Pert apparatus with a copper target and nickel filter.

Biological Tests. *Cells.* Bone resident human MSCs (h-MSCs) were isolated, from the tibial plateau of 3 patients undergoing total knee replacement, after obtaining their informed consent, according to the procedure already established by the laboratory.⁴⁴ Briefly, bone fragments were mechanically fragmented into small pieces to generate a cell suspension that was subjected to a Ficoll-density gradient isolation protocol, as previously reported.⁴⁴ Cells were grown and expanded in α -MEM medium supplemented with 15% FBS and 1% penicillin/streptomycin until passage 2.

Live and Dead Staining. Cell viability of h-MSC seeded on the scaffolds was evaluated after 72 h in culture by the LIVE/DEAD[®] Viability/Cytotoxicity Assay Kit (Thermo Fisher Scientific) based on the simultaneous determination of live (green) and dead (red) cells with two specific probes calcein-AM and ethidium homodimer (EthD-1), respectively. The scaffolds were washed with phosphate-buffered saline (PBS) and then incubated with ethidium homodimer 1 (4 μ M) and calcein-AM (2 μ M) for 30 min at 37 °C at 5% CO₂. After two washing steps with PBS, the scaffolds were evaluated by an Eclipse 90i microscope equipped with Nikon Imaging Software elements (Nikon, Japan). For each sample, Z-stacking images at 10× magnification were captured. Z-stack images combine multiple images (28 consecutive layers) taken at different focal distances every 2.8 μ m to provide a composite image for a total depth of 74.8 μ m, as shown in Figure 7.

Cytotoxicity Assay. Quantification of cytotoxicity was performed using a colorimetric assay based on the measurement of lactate dehydrogenase (LDH) released in the supernatants by damaged cells, according to the manufacturer's instructions (Cytotoxicity Detection Kit, Roche). h-MSCs were seeded onto the scaffolds at a concentration of 2×10^4 cells/cm² in α -MEM 15% FBS; after 24 h, the medium was replaced with α -MEM 15% FBS depleted of phenol-red for 72 h. At the end of incubation, 100 ml of supernatant was assayed for LDH release. Colorimetric detection of LDH was performed at 492–620 nm on a TECAN spectrophotometer, and cytotoxicity was calculated with reference to the control (unstimulated samples) and positive control (Triton X-100 treated samples) according to the formula

$$[(A_{sample} - A_{unstimulated sample})/(CTRL + -A_{unstimulated sample})] \times 100$$

where A is the absorbance. Samples containing h-MSCs grown on plastic were used as a reference value. Each sample was assayed in triplicate.

Apoptosis (TUNEL Assay). Quantification of cell death by apoptosis was performed by measuring the levels of cytoplasmic histone-associated DNA fragments (oligonucleosomes) through an ELISA assay, following the manufacturer's instructions (Cell Death Detection Elisa Plus, Roche). Environmental stress was induced by culturing cells in the condition of serum starvation (5% FBS) for up to 72 h. h-MSCs were seeded onto the scaffolds, as stated above. After 24 h, the cells were starved by replacing the medium with α -MEM 5% FBS. A positive control (labeled as CTRL+ in Figure 7) is provided by the manufacturer, and it is constituted by a lyophilized DNAhistone complex; moreover, cells grown on the scaffold with α -MEM 15% FBS were used as a "negative" control for the occurrence of apoptosis. After 72 h in culture, cells were lysed, and apoptosis induced by reduced serum conditions was assessed in each sample by measuring the enrichment of nucleosomes in the cytoplasm. Colorimetric detection was finally performed at 405 nm on a TECAN spectrophotometer (Infinite M200). Each sample was assayed in triplicate.

RESULTS AND DISCUSSION

This paper describes a new approach for developing a biomimetic SF electrospun scaffold loaded with the H₂S-donor GYY molecule, intended to be used as a functional scaffold in tissue engineering applications (Figure 1). Among the native silk proteins, silkworm silk, mainly that of the domesticated *B. mori*, has been recognized as a high-quality textile fiber and suture. Furthermore, the immersion of the SF electrospun mat in ethanol or methanol solution has been widely demonstrated to induce a fast regeneration of the SF crystalline phase, which was lost during the extraction process.³³

To overcome the instability of GYY in formic acid and in general in aqueous solutions,^{24,25,45} a multilayer electrospun scaffold composed of two external layers of electrospun SF and an inner layer of PLGA containing GYY was realized (Figure

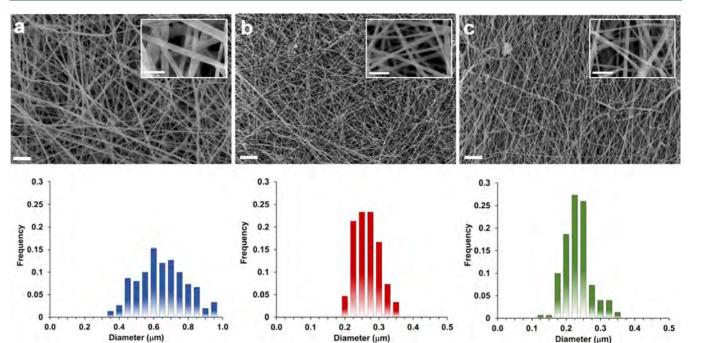


Figure 2. SEM images and corresponding fiber diameter distribution of PLGA and PLGA/GYY electrospun mats: (a) PLGA, (b) PLGA-2GYY, (c) PLGA-5GYY (scale bar = 10 μ m). The insets show the SEM images at higher magnification (scale bar = 2 μ m).

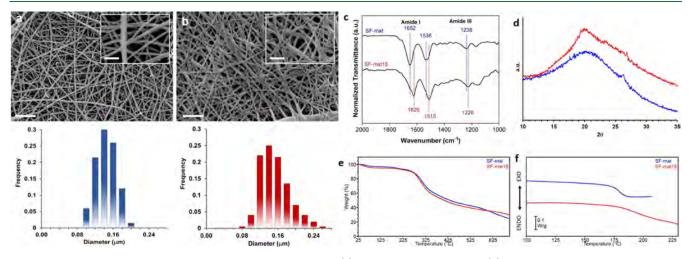


Figure 3. SEM images and corresponding fiber diameter distribution of (a) the SF electrospun mat and (b) SF-mat15 electrospun mat. Scale bar = $2 \mu m$. Insets: scale bar = $1 \mu m$. (c) ATR-IR spectra, (d) WAXD, (e) DSC curves (heating scan after quenching), and (f) TGA curves of SF and SFmat15 mats.

1a,b). PLGA was selected in light of its biocompatibility and the possibility of being electrospun in organic solvents able to induce the solubilization of the H₂S donor without causing its undesired hydrolysis. To investigate the effect of GYY concentration on the H₂S release kinetics and on cells' viability, two concentrations of the H₂S release molecule (2% and 5% w/w with respect to the weight of PLGA) were introduced into the PLGA nanofibers.

The morphological analysis of PLGA, PLGA-2GYY, and PLGA-5GYY (Figure 2) documented the possibility of electrospinning organic solutions of PLGA containing the H₂S donor molecule. For both tested GYY concentrations, regular and bead-free nanofibers were obtained, whose diameter was influenced by the presence of GYY. Notably, the presence of the H₂S releasing agent in the electrospinning solution enabled to obtain a reduction of the nanofibers' mean

diameter (277 \pm 35 nm for PLGA-2GYY; 246 \pm 40 nm for PLGA-5GYY) with respect to PLGA plain fibers (673 \pm 140 nm), ascribable to an increase of the conductivity of the solution.

Diameter (µm)

The influence of GYY on the thermal properties of the PLGA electrospun mats was investigated by DSC and TGA (Figure S1). As expected, the plain polymer is amorphous, with a glass transition around 50 °C and a marked enthalpic relaxation peak related to the polymer's physical aging. The scaffolds show similar DSC curves, suggesting that the presence of GYY at the concentrations studied in this work does not affect PLGA thermal transitions. Similarly, TGA analysis shows that all PLGA-based samples degrade in a single step with a temperature of maximum weight loss rate (T_{max}) of 300 °C (Figure S1), in line with literature findings.⁴⁶ In contrast, GYY shows a more complex degradation behavior,

1370

Article

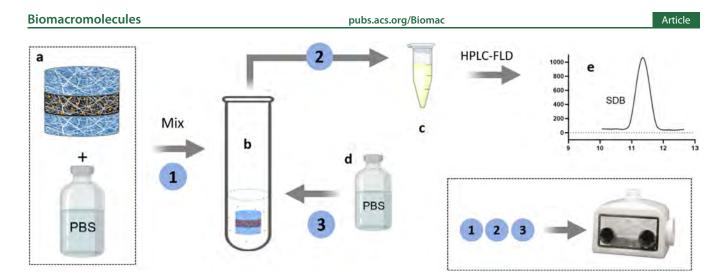


Figure 4. Schematic representation of the H_2S quantification method: (a) 4 mL of PB, sonicated for 30 min before use to remove dissolved gasses and/or entrained gas bubbles, and the multilayer sample was placed into a 20 mL plastic tube first; (b) afterward, the tube was wrapped with an aluminum film and incubated in the thermostatic shaking bath at 37 °C; (c) derivatization reaction of releasing H_2S with the fluorescent alkylating agent MBB; (d) after each withdrawal, the PB solution was refilled with the same amount of PB (30 μ L); (e) quantification of hydrogen sulfide: derivatization of H_2S with MBB, forming a sulfide-dibimane (SDB) product via S-alkylation; the resultant fluorescent SDB is analyzed by HPLC-FLD; the H_2S concentration was then obtained after SDB area interpolation in a calibration curve (Supporting information). Steps 1, 2, and 3 were carried out in a hypoxic chamber (1% O₂) at RT.

with multiple steps and the main step at a T_{max} of 215 °C. GYY molecule does not affect the polymer degradation mechanism, but it seems to slightly increase PLGA stability: the polymer T_{max} goes from 300 °C for plain PLGA to 312 °C for PLGA-2GYY and to 326 °C for PLGA-5GYY. From the analysis of the vibrational absorbance peaks in the FTIR/ATR spectra, no signals due to the GYY molecule were observed, in addition to the expected signals of PLGA, due to the low amount of this component inside the PLGA fibers (Figure S1). In line with previously reported studies,^{33,34,47} the electro-

spinning of SF in formic acid enabled to obtain regular nanofibers, as evident in Figure 3a, with a mean diameter of around 145 \pm 23 nm. To stabilize the structure of SF by inducing the transition of conformation from random coil to crystallizable β -sheet, the SF-mats were subjected to ethanol treatment. In fact, the amorphous structure of SF-mats causes the material to be water soluble and can limit its applications in aqueous biological environments. The immersion of the SFmat in ethanol solutions has been demonstrated to induce a fast regeneration of the SF crystalline phase, which was lost during the extraction process.⁴⁸ As shown in Figure 3b, the treatment performed by immersing the samples for 15 min in absolute ethanol turned out not to significantly affect the fibers' morphology and mean diameter (163 \pm 32 nm), differently from longer immersion times (i.e., 30 and 60 min) for which a partial swelling of the fibers and partial loss of mats' porosity were registered (Figure S2). ATR-IR spectra were collected to investigate the vibrational mode of SF-mat and SF-mat15 amides, which are correlated to the organization of the secondary structure of the protein. Figure 3c documents a significant shift of the SF's amide I and amide III absorption peaks to lower wavenumbers after the treatment in ethanol, clearly highlighting the occurrence of the transition from α helix/random coil to β -sheet conformation.⁴⁸⁻⁵⁰ WAXD analysis (Figure 3d) was in line with previously reported studies^{33,49} and highlighted the presence in the SF-mat of silk I, which accounts for the amorphous structure of the fibroin, as documented by the broad peak centered at $2\theta = 22^{\circ}$. After the ethanol treatment, a sharper peak at 19°, followed by a slightly

detectable peak at 24°, documented the induction of the β -sheet crystalline phase,^{49,51,52} although the amorphous phase still resulted being the most prevalent one. DSC analysis was also performed to investigate further the ethanol treatment's effect on the SF macromolecule conformation (Figure 3e). Due to the limited thermal stability of the SF-mats, demonstrated by the TGA analysis carried out on SF-mat and SF-mat15 (Figure 3f), the DSC heating scans were carried out only up to around 220-250 °C to highlight the endothermic step associated with the glass transition. The calorimetric curves, reported in Figure 3e, document the shift of the SF T_g toward higher temperature after the ethanol treatment, accounting for an increase in the mat's rigidity and confirming an increase of the more stable β -sheet conformation in SF-mat15. Taken altogether, the results of the solid-state characterization of SF electrospun mats confirmed the hypothesis that the ethanol post-treatment induced the formation of the β -sheet conformation in the SF secondary structure, in addition to the α -helix/random coil phase that continued to be the most prevalent phase.

The electrospun SF and PLGA/GYY mats were assembled into a multilayer architecture (Figure 1) and sterilized before being subjected to H_2S release studies. As is well known, the unstable nature of free hydrogen sulfide in solution makes measurement and analysis usually difficult. In this work, the MBB method with HPLC-FLD, suitable for sensitive, quantitative measurement of hydrogen sulfide, was employed according to the ad-hoc developed procedure schematically illustrated in Figure 4.

The HPLC-FLD procedure (see Material and Methods) was first applied to GYY in PB solutions (see the Supporting Information) to check the instrumental setup and verify the efficacy of GYY molecule as a H_2S donor in the solution (Figure S4a,b). Results show an increase of H_2S release in 5h, and then values remain almost stable. A slight decrease, probably ascribable to the degradation of H_2S in the medium, can be observed at time points higher than 20 h. Subsequently, we tested three replicates of multilayer scaffolds without GYY (ML0). No signal was obtained after the derivatization step

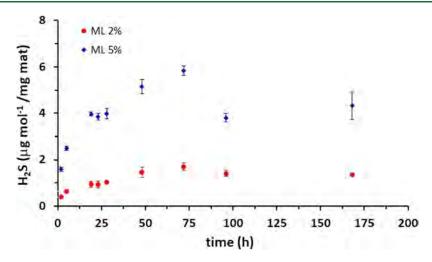


Figure 5. H_2 S-release from sterilized multilayer samples from 0 to 168 h. Three replicates for each kind of sample were assembled, and two derivatizations for each assembled sample were tested and analyzed by HPLC-FLD. Each point is the result of the measurement of six different concentrations. The values plotted represent mean \pm SD.

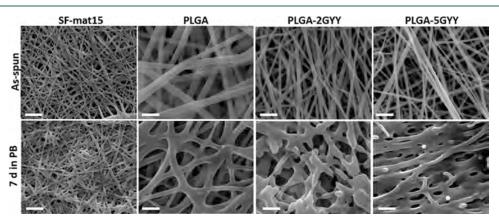


Figure 6. SEM images of SF-mat15, PLGA, PLGA-2GYY, and PLGA-5GYY as produced (upper row) and after 7 days of H₂S-releasing tests (lower row). Scale bar = 5 μ m.

and HPLC-FLD analysis for all sampling time points (data not shown). Conversely, signals were observed for multilayer samples ML2% and ML5%, assembled with different amounts of GYY in the PLGA layer (0.08 mg for ML2% and 0.25 mg for ML5%), highlighting controlled kinetics for H₂S release (Figures 5 and S5). Results show that the sulfide concentration presents an increasing trend and reaches the highest value in 72 h for both ML2% and ML5%. On the other hand, a slow decrease in H₂S concentration was detected after 72 h of immersion, as already shown in previous studies investigating the H₂S release from donors embedded in the polymeric matrix.^{18,44} Furthermore, in tune with previous work,⁴⁴ both the increase and decrease of the sulfide concentration at shorter and longer releasing times, respectively, turned out to strongly depend on the concentration of GYY loaded in the scaffold.

The obtained results can be compared with those reported by Patil et al.²¹ in their work dealing with the H₂S release from a PLGA/GYY gelling delivery system thought for the treatment of glaucoma pathogenesis. The device was characterized by a GYY/PLGA weight ratio of 2% w/w, and it turned out to enable a sustained H₂S release in simulated tear fluid. Indeed, after 24 h, the cumulative release of H₂S from the formulation was around 5 μ g mL⁻¹. Regarding our scaffolds, the release of H₂S from ML2% and ML5% after 20 h of immersion in PB were 0.19 and 1.02 μ g mL⁻¹, respectively. The differences in the released H₂S concentration might be explained by considering the greater propensity of gel formulations to swell and release molecules in a liquid environment in comparison to the solid nanofibrous scaffold reported in our work. It is worth noting that with respect to the solid porous SF devices containing GYY proposed by Raggio et al.,²² both the ML scaffolds (2 and 5%) demonstrated comparable GYY encapsulation efficiency and amount of released H₂S after around 2 h of immersion in an aqueous solution. Moreover, while in Raggio et al. work,²² the H₂S release reached the plateau after 90 min of immersion for all of the tested GYY concentrations, the ML scaffolds described in the present work are characterized by a more controllable and sustained release over time, thus representing a valuable platform to modulate the H₂S release kinetics. When compared to similar scaffolds based on electrospun fibers, our system shows a comparable or more sustained H₂S release: Cacciotti et al.⁵⁶ described a fibrous mat based on electrospun PLA with H₂S releasing capacity, but in this system, the H₂S release peaked after 2 h of incubation in an aqueous medium. Feng et al.¹⁹ described a construct based on electrospun PCL fibers and showed that the diameter of the fibers significantly affects H_2S release; even in this system, the release of H_2S reached a plateau after nearly 24 h, and, in the following 70 h, it slowly

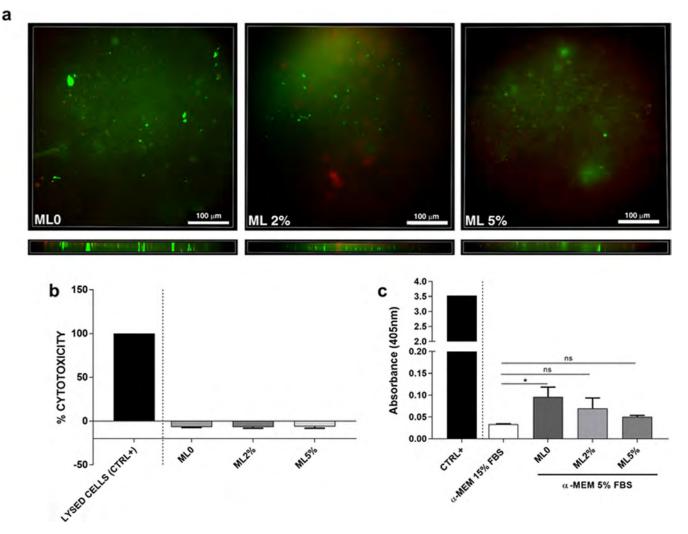


Figure 7. (a) Cell viability evaluated by Calcein-AM/Ethidium homodimer staining. Human MSCs were seeded on the scaffolds and stained after 72 h in culture. Panels show a top-view, single-layer picture, and a picture representing the transversal section of each scaffold for a total depth of 78.4 μ m across each scaffold. (b) Cytotoxicity assay: LDH released in the supernatants by h-MSC was measured after 72 h in culture. Data represent the % cytotoxicity using cells grown on plastic as a control (negative) sample and are expressed as mean \pm SD of N = 3 independent experiments. (c) Histograms representing the enrichment of cytoplasmic oligonucleosomes in h-MSC after 72 h of starvation. Data are expressed as mean \pm SD of N = 3 independent experiments. *p < 0.05 compared to control cells grown on α -MEM 15% FBS; CTRL+ represents a positive control provided by the manufacturer, constituted by lyophilized DNA-histone complex. ns = not significant.

declined. Importantly, consistent with our findings, in each of these works, a scaffold obtained with micro or nanostructured electrospun fibers appears to establish a pro-regenerative microenvironment by counteracting oxidative cell damage, favoring cell colonization and viability and supporting the neosynthesis of extracellular matrix components.

To better understand the results obtained from the H_2S release tests, SEM analyses were carried out on each layer of the three multilayer scaffolds after 7 days of immersion in PB, and the results were compared with those obtained on the layers not subjected to the release tests. Figure 6 reports the good preservation of SF-mat15 after the release test, documenting that the morphology of the external layers is not compromised by the H_2S release from the inner layer. Conversely, PLGA fibers significantly change their morphology after PB immersion, showing increased fiber diameter and "fusion" at contact points. This swelling effect is even more evident with the increase of GYY concentration, highlighting the increased capability of PLGA nanofibers to be penetrated by PB in the presence of the H_2S donor inside the fibers. The

absorption of the aqueous solution inside the fibers might drive the release of H_2S from the multilayer scaffolds.

To evaluate the effect of the electrospun mats with and without GYY on a relevant biological system, h-MSCs were cultured onto the scaffolds for up to 72 h, and cytotoxicity and cell viability were assayed both in standard conditions and in starvation, maintaining MSCs in medium containing 2% FBS. Cellular viability after 72 h in culture was first assessed through the vital dyes calcein and propidium iodide; Figure 7a shows representative pictures of h-MSCs at the end of culture in each of the tested multilayer scaffolds, namely, ML0, ML2%, and ML5%. The 3D-stack images (Figure S6) of the scaffolds show that cells are similarly distributed across the three dimensions and are mostly alive in each sample. Furthermore, quantification of the LDH assay performed after 72 h confirmed that the mats are completely devoid of cytotoxicity even in the presence of GYY at 2 or 5%. In Figure 7b, the histogram reports the values of LDH release referred to cells grown on plastic as a control and shows that cytotoxicity remained below this reference value in each mat. To evaluate

the capacity of H₂S-releasing devices to protect cells against deadly stress, we performed a set of experiments by culturing h-MSCs in conditions of reduced serum (5%) for up to 72 h. Serum starvation is one way to establish environmental stress, which may lead to cell death by activating various pathways.^{53,54} When serum starvation was maintained for 72 h, it resulted in a significant increase of apoptosis in MSC grown on ML0 mats, where no H_2S is released, compared to the control sample represented by MSC grown on α -MEM 15% FBS (Figure 7c); increased apoptosis of MSC in the condition of serum starvation is in agreement with previous reports.55 Conversely, the increased level of apoptosis was prevented in mats containing 2 and 5% GYY, suggesting an active role of H₂S in preventing cell death. This result can be ascribed to the widely documented potential of H₂S-releasing biomaterials to provide a healing environment at sites of tissue damage.^{19,56,57}

CONCLUSIONS

Silk fibroin-based electrospun scaffolds for the controlled release of H₂S and suitable for tissue engineering applications were successfully realized. The scaffolds were characterized by a multilayer architecture, in which a PLGA nanofibrous layer containing GYY was sandwiched between two silk fibroin nanofibrous mats, previously treated in EtOH to favor the formation of the crystalline phase. Morphological analysis carried out on the PLGA layer documented a decrease in the fibers' mean diameter, with the increase of GYY concentration attributed to an increase in the polymeric solution's conductivity in the presence of this molecule. From thermal analysis, GYY contributed to a slight increase in the PLGA thermal stability without affecting its glass transition temperature. No relevant modifications of the ATR-FTIR spectrum of PLGA were observed in the presence of the H₂S donor due to its low concentration inside the electrospun nanofibers. SEM analysis of the silk fibroin electrospun mats confirmed the formation of regular and homogeneous nanofibers, whose morphology was not relevantly affected by the 15 min posttreatment in EtOH. The solid-state characterization of the silk fibroin electrospun mats performed through ATR-FTIR, WAXD, and DSC confirmed the formation of the β -sheet conformation in the SF secondary structure after the 15 min EtOH post-treatment, in addition to the α -helix/random coil. The investigation of the H₂S release kinetics from the multilayer scaffolds, performed in PB according to a properly designed procedure, documented a controlled delivery over 168 h and the possibility of modulating it by acting on GYY concentration inside the PLGA layer. The highest release registered for ML5% with respect to ML2% lies in the more evident "swelling" effect observed for the scaffolds with the increase of GYY concentration. In this context, future studies will be devoted to investigating possible strategies to prevent this partial loss of SF-mats fibrous morphology for long-time immersion in an aqueous solution, also with the aim to further optimize the kinetics of H₂S release.

When cells were added to the device, it became apparent that this multilayered scaffold supports cell colonization and spreading with no evident sign of toxicity linked to H_2S release. Moreover, in keeping with the cytoprotective role of H_2S , we found that H_2S -releasing devices were able to mitigate the cytotoxic effect induced by prolonged serum starvation as compared to control mats. The obtained *in vitro* results support the potential role of this scaffold in maintaining cell integrity and promoting tissue regeneration and pave the way

for the use of H_2S -releasing mats in procedures of GBR-based bone tissue regeneration and for future *in vivo* studies of tissue damage, also in different biological tissues.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acs.biomac.2c01383.

Characterization of electrospun PLGA/GYY4137 scaffolds; optimization of the ethanol treatment on SF-mats and characterization; preparation of the calibration curve for sulfide-dibimane (SDB) quantification; degradation of GYY and H_2S release; and cell viability evaluation (PDF)

AUTHOR INFORMATION

Corresponding Author

Maria Letizia Focarete – Department of Chemistry "Giacomo Ciamician" and INSTM UdR of Bologna, University of Bologna, 40126 Bologna, Italy; Health Sciences & Technologies (HST) CIRI, University of Bologna, 40064 Ozzano Emilia Bologna, Italy; • orcid.org/0000-0002-0458-7836; Email: marialetizia.focarete@unibo.it

Authors

- Anna Liguori Department of Chemistry "Giacomo Ciamician" and INSTM UdR of Bologna, University of Bologna, 40126 Bologna, Italy
- Elisabetta Petri Department of Chemistry "Giacomo Ciamician" and INSTM UdR of Bologna, University of Bologna, 40126 Bologna, Italy
- Chiara Gualandi Department of Chemistry "Giacomo Ciamician" and INSTM UdR of Bologna, University of Bologna, 40126 Bologna, Italy; Interdepartmental Center for Industrial Research on Advanced Applications in Mechanical Engineering and Materials Technology, CIRI-MAM, University of Bologna, 40136 Bologna, Italy; Health Sciences & Technologies (HST) CIRI, University of Bologna, 40064 Ozzano Emilia Bologna, Italy; @ orcid.org/0000-0002-2020-1892
- Luisa S. Dolci Department of Chemistry "Giacomo Ciamician" and INSTM UdR of Bologna, University of Bologna, 40126 Bologna, Italy
- Valentina Marassi Department of Chemistry "Giacomo Ciamician" and INSTM UdR of Bologna, University of Bologna, 40126 Bologna, Italy; byFlow srl, Bologna 40129, Italy
- Mauro Petretta RegenHu Company, CH-1690 Villaz-St-Pierre, Switzerland; orcid.org/0000-0002-8803-7013
- Andrea Zattoni Department of Chemistry "Giacomo Ciamician" and INSTM UdR of Bologna, University of Bologna, 40126 Bologna, Italy; byFlow srl, Bologna 40129, Italy
- Barbara Roda Department of Chemistry "Giacomo Ciamician" and INSTM UdR of Bologna, University of Bologna, 40126 Bologna, Italy; byFlow srl, Bologna 40129, Italy
- Brunella Grigolo RAMSES Laboratory, IRCCS Istituto Ortopedico Rizzoli, 40136 Bologna, Italy
- Eleonora Olivotto RAMSES Laboratory, IRCCS Istituto Ortopedico Rizzoli, 40136 Bologna, Italy

Francesco Grassi – RAMSES Laboratory, IRCCS Istituto Ortopedico Rizzoli, 40136 Bologna, Italy; © orcid.org/ 0000-0002-5127-9561

Complete contact information is available at: https://pubs.acs.org/10.1021/acs.biomac.2c01383

Author Contributions

^VA.L. and E.P. contributed equally to this work. The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

Funding

The research and publication of the article were partly funded by Ministero della Salute, Italy (Fondi Cinque per Mille to the IRCCS Istituto Ortopedico Rizzoli), and by the National Recovery and Resilience Plan (NRRP), Mission 04 Component 2 Investment 1.5 – NextGenerationEU, Call for tender n. 3277 dated 30/12/2021, Award Number: 0001052 dated 23/ 06/2022.

Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

The Italian Ministry of University and Research (MIUR) is acknowledged.

REFERENCES

(1) Abe, K.; Kimura, H. The possible role of hydrogen sulfide as an endogenous neuromodulator. *J. Neurosci.* **1996**, *16*, 1066–1071.

(2) Yang, R.; Liu, Y.; Shi, S. Hydrogen Sulfide Regulates Homeostasis of Mesenchymal Stem Cells and Regulatory T Cells. J. Dent. Res. 2016, 95, 1445–1451.

(3) Whiteman, M.; Li, L.; Rose, P.; Tan, C. H.; Parkinson, D. B.; Moore, P. K. The effect of hydrogen sulfide donors on lipopolysaccharide-induced formation of inflammatory mediators in macrophages. *Antioxid. Redox Signaling* **2010**, *12*, 1147–1154.

(4) Zhou, J.; Cao, C.; Ma, X.; Hu, L.; Chen, L.; Wang, C. In vitro and in vivo degradation behavior of aqueous-derived electrospun silk fibroin scaffolds. *Polym. Degrad. Stab.* **2010**, *95*, 1679–1685.

(5) Wang, M. J.; Cai, W. J.; Li, N.; Ding, Y. J.; Chen, Y.; Zhu, Y. C. The hydrogen sulfide donor NaHS promotes angiogenesis in a rat model of hind limb ischemia. *Antioxid. Redox Signaling* **2010**, *12*, 1065–1077.

(6) Grassi, F.; Tyagi, A.; Calvert, J. W.; Gambari, L.; Walker, L. D.; Yu, M.; Robinson, J.; Li, J. Y.; Lisignoli, G.; Vaccaro, C.; Adams, J.; Pacifici, R. Hydrogen Sulfide is a Novel Regulator of Bone Formation Implicated in the Bone Loss Induced by Estrogen Deficiency. *J. Bone Miner. Res.* **2016**, *31*, 949–963.

(7) Liu, H.; Weng, L.; Yang, C. A review on nanomaterial-based electrochemical sensors for H2O2, H2S and NO inside cells or released by cells. *Microchim. Acta* **2017**, *184*, 1267–1283.

(8) Xu, T.; Scafa, N.; Xu, L. P.; Zhou, S.; Al-Ghanem, K. A.; Mahboob, S.; Fugetsu, B.; Zhang, X. Electrochemical hydrogen sulfide biosensors. *Analyst* **2016**, *141*, 1185.

(9) Hughes, M. N.; Centelles, M. N.; Moore, K. P. Making and working with hydrogen sulfide: The chemistry and generation of hydrogen sulfide in vitro and its measurement in vivo: A review. *Free Radical Biol. Med.* **2009**, *47*, 1346–1353.

(10) Powell, C. R.; Dillon, K. M.; Matson, J.B. A Review of Hydrogen Sulfide (H2S) Donors: Chemistry and Potential Therapeutic Applications. *Biochem. Pharmacol.* **2018**, *149*, 110–123.

(11) Kolluru, G. K.; Shen, X.; Bir, S. C.; Kevil, C. G. Hydrogen sulfide chemical biology: Pathophysiological roles and detection. *Nitric oxide* **2013**, *35*, 5–20.

(12) Shen, X.; Chakraborty, S.; Dugas, T. R.; Kevil, C.G. Hydrogen sulfide measurement using sulfide dibimane: critical evaluation with electrospray ion trap mass spectrometry. *Nitric oxide* **2014**, *41*, 97–104.

(13) Shen, X.; Kolluru, G. K.; Yuan, S.; Kevil, C. G. Measurement of H2S in vivo and in vitro by the monobromobimane method. *Methods Enzymol.* **2015**, *554*, 31–45.

(14) Togawa, T.; Ogawa, M.; Nawata, M.; Ogasawara, Y.; Kawanabe, K.; Tanabe, S. High performance liquid chromatographic determination of bound sulfide and sulfite and thiosulfate at their low levels in human serum by pre-column fluorescence derivatization with monobromobimane. *Chem. Pharm. Bull.* **1992**, *40*, 3000–3004.

(15) Wintner, E. A.; Deckwerth, T. L.; Langston, W.; Bengtsson, A.; Leviten, D.; Hill, P.; Insko, M. A.; Dumpit, R.; Vanden Ekart, E.; Toombs, C. F.; Szabo, C. A monobromobimane-based assay to measure the pharmacokinetic profile of reactive sulphide species in blood. *Br. J. Pharmacol.* **2010**, *160*, 941–957.

(16) Mauretti, A.; Neri, A.; Kossover, O.; Seliktar, D.; Di Nardo, P.; Melino, S. Design of a Novel Composite H 2 S-Releasing Hydrogel for Cardiac Tissue Repair. *Macromol. Biosci.* **2016**, *16*, 847–858.

(17) Wu, J.; Chen, A.; Zhou, Y.; Zheng, S.; Yang, Y.; An, Y.; Xu, K.; He, H.; Kang, J.; Luckanagul, J. A.; Xian, M.; Xiao, J.; Wang, Q. Novel H2S-Releasing hydrogel for wound repair via in situ polarization of M2 Macrophages. *Biomaterials* **2019**, *222*, No. 119398.

(18) Wu, J.; Li, Y.; He, C.; Kang, J.; Ye, J.; Xiao, Z.; Zhu, J.; Chen, A.; Feng, S.; Li, X.; Xiao, J.; Xian, M.; Wang, Q. Novel H2S Releasing Nanofibrous Coating for In Vivo Dermal Wound Regeneration. ACS Appl. Mater. Interfaces **2016**, *8*, 27474–27481.

(19) Feng, S.; Zhao, Y.; Xian, M.; Wang, Q. Biological thiolstriggered hydrogen sulfide releasing microfibers for tissue engineering applications. *Acta Biomater.* **2015**, *27*, 205–213.

(20) Zhang, H.; Hao, L. Z.; Pan, J. A.; Gao, Q.; Zhang, J. F.; Kankal, R. K.; Wang, S. B.; Chen, A. Z.; Zhang, H. L. Microfluidic fabrication of inhalable large porous microspheres loaded with H2S-releasing aspirin derivative for pulmonary arterial hypertension therapy. *J. Controlled Release* **2021**, 329, 286–298.

(21) Patil, A.; Singh, S.; Opere, C.; Dash, A. Sustained-Release Delivery System of a Slow Hydrogen Sulfide Donor, GYY4137, for Potential Application in Glaucoma. *AAPS PharmSciTech* **2017**, *18*, 2291–2302.

(22) Raggio, R.; Bonani, W.; Callone, E.; Dirè, S.; Gambari, L.; Grassi, F.; Motta, A. Silk Fibroin Porous Scaffolds Loaded with a Slow-Releasing Hydrogen Sulfide Agent (GYY4137) for Applications of Tissue Engineering. *ACS Biomater. Sci. Eng.* **2018**, *4*, 2956–2966. (23) Rose, P.; Dymock, B. W.; Moore, P. K. GYY4137, a Novel Water-Soluble, H2S-Releasing Molecule. *Methods Enzymol.* **2015**, 554, 143–167.

(24) Li, L.; Whiteman, M.; Guan, Y. Y.; Neo, K. L.; Cheng, Y.; Lee, S. W.; Zhao, Y.; Baskar, R.; Tan, C. H.; Moore, P. K. Characterization of a Novel, Water-Soluble Hydrogen Sulfide–Releasing Molecule (GYY4137) New Insights Into the Biology of Hydrogen Sulfide. *Circulation* **2008**, *117*, 2351–2360.

(25) Alexander, B. E.; Coles, S. J.; Fox, B. C.; Khan, T. F.; Maliszewski, J.; Perry, A.; Pitak, M. B.; Whiteman, M.; Wood, M. E. Investigating the generation of hydrogen sulfide from the phosphonamidodithioate slow-release donor GYY4137. *MedChem*-*Commun.* **2015**, *6*, 1649.

(26) Kundu, B.; Rajkhow, R.; Kundu, S. C.; Wang, X. Silk fibroin biomaterials for tissue regenerations. *Adv. Drug Delivery Rev.* **2013**, *65*, 457–470.

(27) Varkey, A.; Venugopal, E.; Sugumaran, P.; Janarthanan, G.; Pillai, M. M.; Rajendran, S.; Bhattacharyya, A. Impact of silk fibroinbased scaffold structures on human osteoblast MG63 cell attachment and proliferation. *Int. J. Nanomed.* **2015**, *10*, 43–51.

(28) Drury, J. L.; Mooney, D. J. Hydrogels for tissue engineering: scaffold design variables and applications. *Biomaterials* **2003**, *24*, 4337–4351.

(29) Lavik, E.; Langer, R. Tissue engineering: current state and perspectives. *Appl. Microbiol. Biotechnol.* 2004, 65, 1–8.

(30) Du, G. Y.; He, S. W.; Sun, C. X.; Mi, L. D. Bone Morphogenic Protein-2 (rhBMP2)-Loaded Silk Fibroin Scaffolds to Enhance the Osteoinductivity in Bone Tissue Engineering. *Nanoscale Res. Lett.* **2017**, *12*, 573.

(31) Rockwood, D. N.; Preda, R. C.; Yücel, T.; Wang, X.; Lovett, M. L.; Kaplan, D. L. Materials fabrication from Bombyx mori silk fibroin. *Nat. Protoc.* **2011**, *6*, 1612–1631.

(32) Zhu, J.; Zhang, Y.; Shao, H.; Hu, X. Electrospinning and rheology of regenerated Bombyx mori silk fibroin aqueous solutions: The effects of pH and concentration. *Polymer* **2008**, *49*, 2880–2885.

(33) Kim, S. H.; Nam, Y. S.; Lee, T. S.; Park, W. H. Silk Fibroin Nanofiber. Electrospinning, Properties, and Structure. *Polym. J.* 2003, 35, 185–190.

(34) Meechaisue, C.; Wutticharoenmongkol, P.; Waraput, R.; Huangjing, T.; Ketbumrung, N.; Pavasant, P.; Supaphol, P. Preparation of electrospun silk fibroin fiber mats as bone scaffolds: a preliminary study. *Biomed. Mater.* **2007**, *2*, 181–188.

(35) Amiraliyan, N.; Nouri, M.; Kish, M. H. Electrospinning of Silk Nanofibers. I. An Investigation of Nanofiber Morphology and Process Optimization Using Response Surface Methodology. *Fibers Polym.* **2009**, *10*, 167–176.

(36) Alessandrino, A.; Marelli, B.; Arosio, C.; Fare, S.; Tanzi, M. C.; Freddi, G. Electrospun Silk Fibroin Mats for Tissue Engineering. *Eng. Life Sci.* **2008**, *8*, 219–225.

(37) Elgali, I.; Omar, O.; Dahlin, C.; Thomsen, P. Guided bone regeneration: materials and biological mechanisms revisited. *Eur. J. Oral Sci.* **2017**, *125*, 315–337.

(38) Liang, D.; Hsiao, B. S.; Chu, B. Functional electrospun nanofibrous scaffolds for biomedical applications. *Adv. Drug Delivery Rev.* 2007, *59*, 1392–1412.

(39) Katti, D. S.; Robinson, K. W.; Ko, F. K.; Laurencin, C. T. Bioresorbable nanofiber-based systems for wound healing and drug delivery: optimization of fabrication parameters. *J. Biomed. Mater. Res., Part B* **2004**, *70B*, 286–296.

(40) Martins, C.; Sousa, F.; Araújo, F.; Sarmento, B. Functionalizing PLGA and PLGA Derivatives for Drug Delivery and Tissue Regeneration Applications. *Adv. Healthcare Mater.* **2018**, *7*, No. 1701035.

(41) Shahverdi, S.; Hajimiri, M.; Esfandiari, M. A.; Larijani, B.; Atyabi, F.; Rajabiani, A.; Dehpour, A. R.; Gharehaghaji, A. A.; Dinarvand, R. Fabrication and structure analysis of poly(lactide-coglycolic acid)/silk fibroin hybrid scaffold for wound dressing applications. *Int. J. Pharm.* **2014**, *473*, 345–355.

(42) Yao, J.; Wang, Y.; Ma, W.; Dong, W.; Zhang, M.; Sun, D. Dual-Drug-Loaded Silk Fibroin/PLGA Scaffolds for Potential Bone Regeneration Applications. J. Nanomater. **2019**, 2019, No. 8050413.

(43) Roda, B.; Zhang, N.; Gambari, L.; Grigolo, B.; Eller-Vainicher, C.; Gennari, L.; Zappi, A.; Giordani, S.; Marassi, V.; Zattoni, A.; Reschiglian, P.; Grassi, F. Optimization of a Monobromobimane (MBB) Derivatization and RP-HPLC-FLD Detection Method for Sulfur Species Measurement in Human Serum after Sulfur Inhalation Treatment. *Antioxidants* **2022**, *11*, 939.

(44) Gambari, L.; Lisignoli, G.; Gabusi, E.; Manferdini, C.; Paolella, F.; Piacentini, A.; Grassi, F. Distinctive expression pattern of cystathionine-beta-synthase and cystathionine-gamma-lyase identifies mesenchymal stromal cells transition to mineralizing osteoblasts. *J. Cell. Physiol.* **2017**, *232*, 3574–3585.

(45) Kang, J.; Neill, D.; Xian, M. Phosphonothioate-Based Hydrogen Sulfide Releasing Reagents: Chemistry and Biological Applications. *Front. Pharmacol.* **2017**, *8*, 457.

(46) Fouad, H.; Elsarnagawy, T.; Almajhdi, F.N.; Khalil, K. A. Preparation and In Vitro Thermo-Mechanical Characterization of Electrospun PLGA Nanofibers for Soft and Hard Tissue Replacement. *Int. J. Electrochem. Sci.* **2013**, *8*, 2293–2304.

(47) Min, B. M.; Lee, G.; Kim, S. H.; Nam, Y. S.; Lee, T. S.; Park, W. H. Electrospinning of silk fibroin nanofibers and its effect on the adhesion and spreading of normal human keratinocytes and fibroblasts in vitro. *Biomaterials* **2004**, *25*, 1289–1297.

(48) Li, M.; Tao, W.; Kuga, S.; Nishiyama, Y. Controlling Molecular Conformation of Regenerated Wild Silk Fibroin by Aqueous Ethanol Treatment. *Polym. Adv. Technol.* **2003**, *14*, 694–698.

(49) Kamalha, E.; Zheng, Y.; Zeng, Y. Analysis of the secondary crystalline structure of regenerated Bombyx mori fibroin. *Res. Rev. BioSci.* **2013**, *7*, 76–83.

(50) Chalmers, J. M.; Griffiths, P. R.Instrumentation for FT-IR imaging. In *Handbook of Vibrational Spectroscopy*; Wiley Chichester, 2002; Vol. 2, pp 1386–1404.

(51) Um, I. C.; Kweon, H. Y.; Park, Y. H.; Hudson, S. Structural characteristics and properties of the renenerated silk fibroin prepared from formic acid. *Int. J. Biol. Macromol.* **2001**, *29*, 91–97.

(52) Asakura, T.; Kuzuhara, A.; Tabeta, R.; Saito, H. Conformation Characterization of Bombyx mori Silk Fibroin in the Solid State by High-Frequency 13C Cross Polarization-Magic Angel Spinning NMR, X-ray Diffraction, and Infrared Spectroscopy. *Macromolecules* **1985**, *18*, 1841–1845.

(53) Braun, F.; Bertin-Ciftci, J.; Gallouet, A. S.; Millour, J.; Juin, P. Serum-nutrient starvation induces cell death mediated by Bax and Puma that is counteracted by p21 and unmasked by Bcl-x(L) inhibition. *PLoS One* **2011**, *6*, No. e23577.

(54) Huang, Y.; Fu, Z.; Dong, W.; Zhang, Z.; Mu, J.; Zhang, J. Serum starvation-induces down-regulation of Bcl-2/Bax confers apoptosis in tongue coating-related cells in vitro. *Mol. Med. Rep.* **2018**, *17*, 5057–5064.

(55) Zhu, W.; Chen, J.; Cong, X.; Hu, S.; Chen, X. Hypoxia and Serum Deprivation-Induced Apoptosis in Mesenchymal Stem Cells. *Stem Cells* **2006**, *24*, 416.

(56) Cacciotti, I.; Ciocci, M.; Di Giovanni, E.; Nanni, F.; Melino, S. Hydrogen Sulfide-Releasing Fibrous Membranes: Potential Patches for Stimulating Human Stem Cells Proliferation and Viability under Oxidative Stress. *Int. J. Mol. Sci.* **2018**, *19*, 2368.

(57) Zhao, X.; Liu, L.; An, T.; Xian, M.; Luckanagul, J. A.; Su, Z.; Lin, Y.; Wang, Q. A hydrogen sulfide-releasing alginate dressing for effective wound healing. *Acta Biomater.* **2020**, *104*, 85–94.

Recommended by ACS

Is 3D Printing Promising for Osteochondral Tissue Regeneration?

Duygu Ege and Vasif Hasirci	
MARCH 21, 2023	
ACS APPLIED BIO MATERIALS	READ 🗹

Tissue-Adhesive Decellularized Extracellular Matrix Patches Reinforced by a Supramolecular Gelator to Repair Abdominal Wall Defects

Akihiro Nishiguchi, Tetsushi Taguchi, *et al.* MARCH 07, 2023 BIOMACROMOLECULES

Albumin-Coated Polycaprolactone (PCL)–Decellularized Extracellular Matrix (dECM) Scaffold for Bone Regeneration

Radoslaw Junka, Xiaojun Yu, et al. NOVEMBER 14, 2022 ACS APPLIED BIO MATERIALS

Absorbable Electrospun Poly-4-hydroxybutyrate Scaffolds as a Potential Solution for Pelvic Organ Prolapse Surgery

Kim Verhorstert, Zeliha Guler, *et al.* OCTOBER 31, 2022 ACS APPLIED BIO MATERIALS

READ 🗹

READ 🗹

Get More Suggestions >