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Micro- and Nanopatterned Silk Substrates for Antifouling Applications

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# Micro and nano-patterned silk substrates for antifouling applications

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**ABSTRACT**

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7 A major problem of current biomedical implants is due to bacterial colonization and subsequent biofilm  
8 formation, which seriously affects their functioning and can lead to serious post-surgical complications.  
9 Intensive efforts have been directed towards the development of novel technologies that can prevent  
10 bacterial colonization while requiring minimal antibiotics doses. To this end, biocompatible materials with  
11 intrinsic antifouling capabilities are in high demand. Silk fibroin, widely employed in biotechnology,  
12 represents an interesting candidate.  
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20 Here, we employ a soft-lithography approach to realize micro- and nano-structured silk fibroin substrates,  
21 with different geometries. We show that patterned silk film substrates support mammal cells (HEK-293)  
22 adhesion and proliferation, and at the same time they intrinsically display remarkable antifouling properties.  
23 We employ *Escherichia coli* as representative gram-negative bacteria and we observe a up to 66% decrease  
24 in the number of bacteria that adhere to patterned silk surfaces as compared to control, flat silk samples.  
25 The mechanism leading to the inhibition of biofilm formation critically depends on the microstructures  
26 geometry, involving both a steric and a hydrophobic effect. We also couple silk fibroin patterned films to a  
27 biocompatible, optically-responsive organic semiconductor, and we verify that the antifouling properties are  
28 very well preserved.  
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41 The technology described here is of interest for the next-generation of biomedical implants, involving the  
42 use of materials with enhanced antibacterial capability, easily processable, highly biocompatible and  
43 promptly available for coupling with photoimaging and photodetection techniques.  
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## 1. Introduction

In recent years, silk fibroin has been eliciting an ever increasing interest in the biotechnology field. Several applications have been reported, spanning from tissue engineering<sup>1,2</sup> to regenerative medicine,<sup>3-9</sup> from drug delivery<sup>10-12</sup> to bio-photonics,<sup>13-20</sup> from implantable devices<sup>21-25</sup> up to the last frontiers in bioengineering, the realization of bio-degradable, bio-resorbable and edible electronic devices.<sup>26-30</sup> The main reason behind such a success story relies on the peculiar advantages offered by silk, in terms of outstanding mechanical properties, chemical-physical versatility due to its polymorphic character, thermal and environmental stability, suitability to several processing techniques (easily allowing for covering dimensions from the nano- to the macro-scale), endless opportunities of functionalization with biomolecules and drugs, and excellent cytocompatibility. Overall, silk is widely recognized as a highly promising material platform, with global impact in the biomedical, biophotonics and bioelectronics fields.<sup>31</sup>

The development of silk-based materials for implantable biotechnological devices has fostered the investigation of the benefits of silk functionalization with antimicrobial agents.<sup>32-35</sup> In fact, the main, hardly addressable risk leading to failure of medical implants is represented by microbial contamination, which causes device-associated severe infections.<sup>36</sup> Bacteria preferentially adhere and proliferate on the implant surface, producing a biofilm, i.e. a layer of aggregated bacteria embedded into a matrix composed of extracellular polymeric substances such as proteins, DNA and polysaccharides.<sup>37</sup> The presence of the biofilm confers to the microorganisms a much higher resistance than the unattached bacteria, requiring 500–5000 times higher doses of antibiotics.<sup>37,38</sup> Some literature reports have addressed this problem by taking advantage of the chemical versatility of silk and by functionalizing the silk-based implant with antimicrobial agents. Unfortunately, this approach, albeit capable of substantially reducing bacterial contamination, has a serious drawback, since the extensive use of antibiotics inevitably leads to an enhanced bacterial resistance towards the common antibiotic agents, thus establishing a vicious circle.

A solution consists in the hindering of the formation of the biofilm (antifouling) by properly engineering implanted materials and devices.<sup>39</sup> It has been widely reported that microbial adhesion can be efficiently limited by modifying the material surface properties through the control of physical-chemical parameters

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3 such as steric hindrance, hydrophobicity, Van der Waals forces or electrostatic interactions.<sup>40–44</sup> Nature offers  
4 multiple examples of fouling control strategies employed by animal and plants, which involve different  
5 physical-chemical approaches. The latter are often associated to the presence of micro/nano-patterned  
6 surface topographies characterized by various diameters, widths, lengths, heights, and pitches.<sup>40,45–47</sup>  
7  
8 Laboratory research has been oriented to the mimic of these morphologies, finding that the tuning of the  
9 micro/nano-structure parameters may prevent surface colonization by specific microorganisms.<sup>45,46</sup> Many  
10 organic biomaterials have been tested, including, among others, poly(dimethylsiloxane), polyurethane,  
11 cellulose, poly(ethylene glycol) and thiophene-based conducting polymers.<sup>45,47,48</sup> Surprisingly, the endless  
12 possibilities offered by silk manufacturing and processing techniques just started to gain attention. In a very  
13 recent work by Chu and colleagues, the nanostructuring of silk substrates was first reported for orthopedic  
14 implants.<sup>49</sup>

15  
16 In this work, we report the preparation of micro and nano-patterned silk surface topographies and the  
17 quantification of the adhesion grade of GFP-encoded *Escherichia coli* (*E. coli*) bacteria by means of  
18 fluorescence microscopy imaging. We observe a reduction in bacterial adhesion on structured surfaces up  
19 to 66% as compared with unpatterned substrates. Importantly, this is not accompanied by alteration in the  
20 cytocompatibility of the patterned silk substrate, as evaluated by viability assays with mammal cell model  
21 cultures. In view of possible applications in the bio-photonics field, we also couple silk-based substrates to  
22 light-sensitive organic semiconducting polymer with distinct optoelectronics properties. We verify that the  
23 coating with the polymer does not hinder the antifouling effect shown by the patterned silk substrates. Our  
24 work offers interesting perspectives in the realization of novel silk-based biocompatible devices, endowed  
25 with both antifouling and photonics functionalities.

## 26 27 28 29 30 31 32 33 34 35 36 37 38 39 40 41 42 43 44 45 46 47 48 49 50 51 52 **2. Experimental**

### 53 54 **2.1 Materials and silk substrates fabrication**

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56 Regio-regular poly(3-hexylthiophene-2,5-diyl) (rr-P3HT) (purity 99.995%, molecular weight 15000 - 45000),  
57 o-dichlorobenzene, Dulbecco's Modified Eagles Medium (DMEM), Trypsin-EDTA, Penicillin, phosphate buffer  
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3 saline (PBS) tablets, streptomycin, fibronectin (from bovine plasma), 3-(4,5-dimethylthiazol-2-yl)-2,5-  
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5 diphenyltetrazolium bromide (MTT), LiBr, Ampicillin, EtOH were purchased from Sigma Aldrich. PDMS  
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7 elastomer (Sylgard 184) was purchased from Dow Corning. Fetal Bovine Serum (FBS) was purchased from  
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9 Euroclone. All chemicals were used without any further purification.

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12 For the preparation of silk fibroin films, *Bombyx mori* degummed silk fibers (2 g) were dissolved in a LiBr  
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14 solution (9.3 M, 10 mL) at 60° C for 90 min obtaining a 20% (w/v) solution. A dialysis tubing (molecular weight  
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16 cutoff of 12 000 Da) was used for dialyzing the solution in water for 48 h at room temperature to remove the  
17  
18 LiBr salt. The aggregates formed during dialysis were removed by centrifugation (4000 rpm, 25 min). After  
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20 these steps, the aqueous silk solution concentration was approximately 6% (w/v), as determined by a  
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22 gravimetric analysis performed on the dried sample. Silk solution was then casted on both flat and patterned  
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24 PDMS master mold and dried in fume hood under gentle aspiration for 24 h, following the procedure  
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26 described in ref <sup>50</sup>. The patterned PDMS films were obtained by replica molding of the polycarbonate layer  
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28 of a CD-ROM (to obtain the master for the nano stripes) and of honeycomb structured layers prepared by  
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30 breath figure templating technique (to obtain the masters for the micro wells). The polymeric materials used  
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32 and conditions of the breath figure process were defined in order to obtain two different kinds of  
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34 microporous templates, with larger (~5 μm) and smaller (1-2 μm) microcavities, as described in ref <sup>51</sup> and ref  
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36 <sup>52</sup>, respectively. Once solidified, the fibroin films were subjected to a water vapor annealing by placing them  
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38 in a desiccator together with a water reservoir for 12 hours. This process increases the crystallinity of the silk-  
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40 based material and makes it insoluble in water.<sup>53</sup> Then, the dry fibroin films were softened by 2 min  
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42 immersion in distilled water and carefully detached from the PDMS master.

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44 For the preparation of polymer-coated silk substrates, fibroin silk films were anchored on glass slide using  
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46 adhesive tape and air-dried for one night. rr-P3HT was dissolved in o-dichlorobenzene (5 g l<sup>-1</sup>) and stirred for  
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48 one night at 50°C. rr-P3HT solution was then deposited on dry silk substrates by spin coating (speed 1500  
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50 rpm, acceleration 1600 rpm s<sup>-1</sup>) and the obtained samples were left in vacuum for 40 minutes before use.  
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## 2.2 Substrates characterization

Scanning electron microscopy (SEM) micrographs were acquired using a TESCAN MIRA III scanning electron microscope (operating voltage 4kV, working distance 19 mm, stage tilt angle 30°). Prior to SEM images acquisition, silk substrates were attached to glass slides using adhesive tape, air-dried and covered by a thin gold layer (thickness 6 nm, 1.5 Cr adhesion layer) using a metal evaporator. The atomic force microscopy (AFM) images were acquired using NT-MDT NTEGRA apparatus in tapping mode under ambient conditions. For water contact angle measurements, silk-based samples were attached to glass slides using adhesive tape and air-dried overnight. Then, photographs of 5  $\mu$ l water droplets deposited on the different substrates were taken using an optical contact angle measuring and contour analysis system (DataPhysics OCA 15EC). The calculation of the static water contact angle was carried out using the SCA 20 software (DataPhysics), according to the sessile drop method.

## 2.3 Cells cultures preparation and viability assay

Silk substrates were sterilized by immersion in EtOH 70% for 3 hours and washed 3 times with mQ-H<sub>2</sub>O. Then, a layer of fibronectin (2  $\mu$ g ml<sup>-1</sup> in PBS buffer solution) was deposited on the samples surface and incubated for 1 hour at 37 °C, in order to promote cellular adhesion. Excess fibronectin was then removed by rinsing with PBS prior to cell plating. HEK-293 cells were cultured in cell culture flasks containing Dulbecco's modified Eagle's medium (DMEM) with 10% Fetal Bovine Serum (FBS), 100  $\mu$ g l<sup>-1</sup> Penicillin and 100  $\mu$ g ml<sup>-1</sup> Streptomycin. Culture flasks were maintained in a humidified incubator (Forma series II water jacketed CO<sub>2</sub> incubator, Thermofisher) at 37 °C with 5% CO<sub>2</sub>. When at confluence, HEK-293 cells were enzymatically dispersed using trypsin-EDTA and then plated on the different samples at a concentration of 20000 cells cm<sup>-2</sup> and maintained in the incubator at 37 °C with 5% CO<sub>2</sub>. Cells proliferation was assessed after 24, 48, 72 and 96 hours *in vitro* by performing the MTT assay. For each time point, the growing medium was replaced with RPMI medium without phenol red containing 0.5 mg ml<sup>-1</sup> of MTT and the samples were maintained in the dark at 37 °C for 3 h. Then, the medium was removed and the samples were air-dried at room temperature. Dry samples were subsequently immersed in 200  $\mu$ l of ethanol in order to dissolve the formazan salt produced by cells through reduction of MTT. The proliferation cell rate was calculated as the



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3 difference in absorbance at 560 nm and 690 nm. Statistical significance was determined by the one-way  
4 ANOVA Dunnett's post-hoc test.  
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## 10 2.4 Bacterial cultures and adhesion evaluation

11 The gene encoding eGFP was PCR amplified from a pDONR-P2R-P3 vector carrying the eGFP coding  
12 sequence, and cloned into a pET23a(+) by conventional methods using NdeI and XhoI restriction sites (Table  
13 PRIMER). The resulting construct (pET23a(+)-eGFP) was sequence verified and transformed into *E. coli*  
14 Rosetta (DE3) competent cells (Invitrogen).  
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Primer sequence (5'-3') used in this study.* <sup>a</sup>		
eGFP	Forward <sub>Nat</sub>	<u>ccccccat</u> atggtgagcaagggcgaggagc
	Reverse <sub>1</sub>	ggtctggctc <u>gaggtacagctc</u> gtccatgcc
*: the restriction sites for NdeI (forward) and XhoI (reverse) are underlined.		

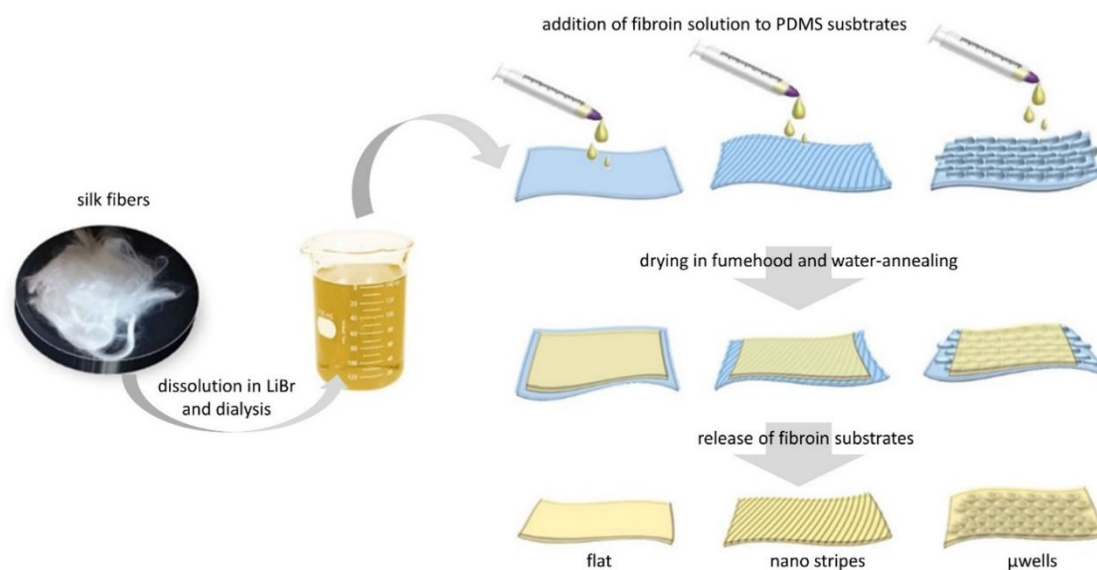
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34 Bacterial cells were cultured as previously described.<sup>54</sup> Briefly, a single colony of bacteria carrying the  
35 pET23a(+)-eGFP plasmid was inoculated in Luria Bertani (LB) broth in the presence of Ampicillin (50 µg/ml)  
36 and incubated overnight at 37°C until stationary phase was reached. The bacterial culture was then diluted  
37 to OD<sub>600</sub> = 0.1 (1x10<sup>7</sup> CFU ml<sup>-1</sup>) in LB media supplemented with the same antibiotics. Then, each substrate  
38 was incubated with 5 ml of bacterial suspension and kept in 12-well culture plates. Finally, isopropyl-1-thio-  
39 D-galactopyranoside (IPTG) was added to the bacterial culture at a final concentration of 1 mM, following a  
40 24-hours incubation at 25 C on a platform shaker. After 24 hours the samples were removed from the  
41 bacteria growing medium, washed 3 times with mQ-H<sub>2</sub>O and covered with a drop of water. Fluorescence  
42 images for the evaluation of the bacteria adhesion grade on the different substrates were acquired using an  
43 inverted fluorescence microscope (Nikon Eclipse Ti), by exciting with a 470 nm light source (Lumencor  
44 Spectra X) and using a standard FITC filters set. The average number of bacteria adhered on each substrate  
45 type was quantified using Imagej software. Mean values were averaged over n = 10 fields of view for each  
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3 sample type, and over  $n = 3$  statistically independent samples. For evaluating the bactericidal activity,  
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5 bacterial cells were cultured as described above. However, no IPTG was added into the bacterial solution.  
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7 The optical density of these solutions was measured at 600nm using a UV-Vis Spectrometer. Control  
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9 measurements on bacteria grown in the medium where no sample is present were carried out at the same  
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11 time.  
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### 16 **3. Results and Discussion**

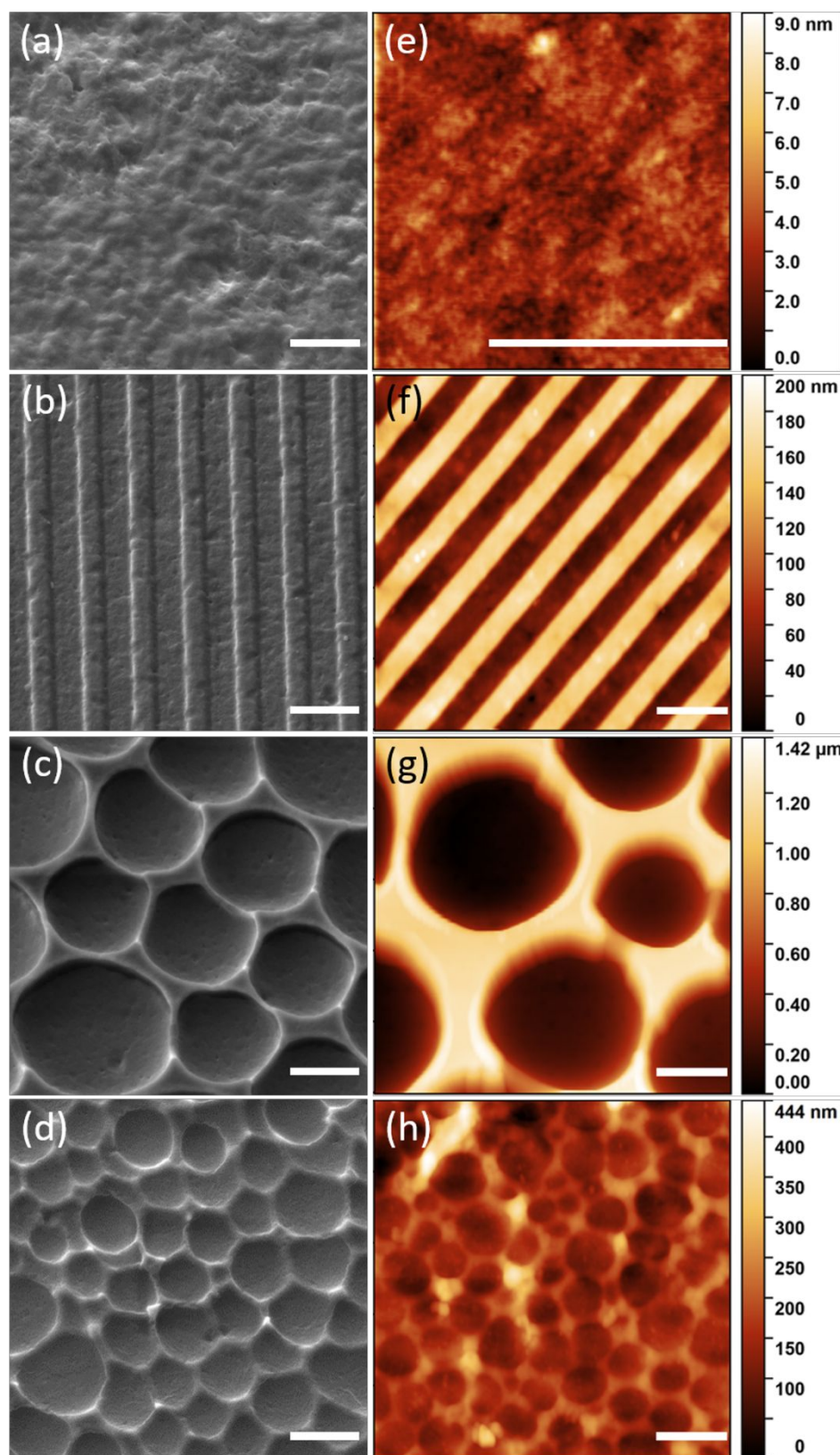
#### 17 **3.1. Fabrication of micro-patterned silk substrates**

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19 We fabricate different micro- and nano-patterned silk fibroin substrates by employing a soft-lithography  
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21 approach.<sup>50</sup> The procedure, sketched in **Figure 1**, starts with the extraction of the silk fibroin protein from  
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23 the *Bombyx mori* degummed silk fibers through dissolution in LiBr solution and subsequent dialysis against  
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25 ultrapure water. The silk fibroin solution is then drop casted on top of micro- and nano-patterned PDMS  
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27 molds. Once solidified, the fibroin films are subjected to water vapor annealing for increasing the crystallinity  
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29 of the silk-based material, thus making them water insoluble and producing free-standing silk fibroin  
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31 patterned films.<sup>53</sup> Four types of surface morphology are obtained by replicating different PDMS molds: flat,  
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33 nano stripes, small  $\mu$ wells and big  $\mu$ wells. To obtain the PDMS mold for nano-striped silk, we replicated the  
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35 grooved polycarbonate layer of a CD-ROM, while for fabricating the micro-domes mold (negative morphology  
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37 for  $\mu$ wells) we employed the nature-inspired templating process known as breath figure formation.<sup>55</sup>  
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39 Therefore, not only the preparation of silk substrates but the whole patterning process is achieved by  
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41 affordable and simple techniques based on self-assembly. For the flat controls, a flat PDMS layer is used.  
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**Figure 1. Silk substrates fabrication.** A soft lithography and self-assembly approach allows for fast, scalable and highly repeatable fabrication of patterned silk substrates, with different design. In this case, nano stripes and micro wells with different diameters have been realized.

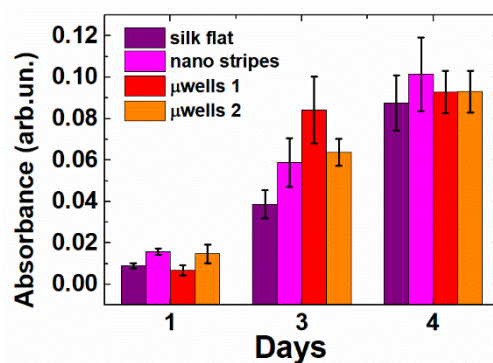
**Figure 2** shows the surface topography of the planar, unpatterned silk substrates (**Figure 2a**) and of micro/nano-structured samples with three different geometries (**Figure 2b-2d**), as obtained by acquiring SEM micrographs in top-view configuration. By employing different PDMS molds, three topographies are obtained: (1) a nano stripes-patterned silk fibroin surface, composed by grooves of  $\sim 800$  nm width and spacing of the same dimension (nano stripes, **Figure 2b**); (2) silk fibroin micro wells of different diameters, ranging from 3.5 to 5  $\mu\text{m}$  ( $\mu\text{wells}$  1, **Figure 2c**); (3) silk fibroin micro wells with a shape similar to  $\mu\text{wells}$  1, but lower diameter, in the range 1-2  $\mu\text{m}$  ( $\mu\text{wells}$  2, **Figure 2d**). Silk substrates morphology is also characterized by AFM (**Figure 2e-2h**). The surface roughness of the flat silk control substrates (**Figure 2e**) is in line with the existing literature (root mean square roughness (RMS) value  $\pm$  standard error of the mean (s.e.m.),  $0.86 \pm 0.03$  nm).<sup>56</sup> The average height of the micro-structured features amounts at  $0.124 \pm 0.005$   $\mu\text{m}$ ,  $1.06 \pm 0.03$   $\mu\text{m}$  and  $0.122 \pm 0.007$   $\mu\text{m}$  (mean  $\pm$  s.e.m.) for the nano stripes,  $\mu\text{wells}$  1 and  $\mu\text{wells}$  2 cases respectively (**Figures 2f-2h**).



**Figure 2. Topography of flat and micro/nano-patterned silk substrates.** (a)-(d): SEM top-view images of silk flat (a), nano stripes (b),  $\mu$ wells 1 (c) and  $\mu$ wells 2 (d). Scale bars, 2  $\mu$ m. (e)-(h): AFM topography images of silk flat (e), nano stripes (f),  $\mu$ wells 1 (g) and  $\mu$ wells 2 (h) samples. Scale bars, 2  $\mu$ m.

### 3.2. HEK-293 cells cultures on micro/nano-patterned silk substrates

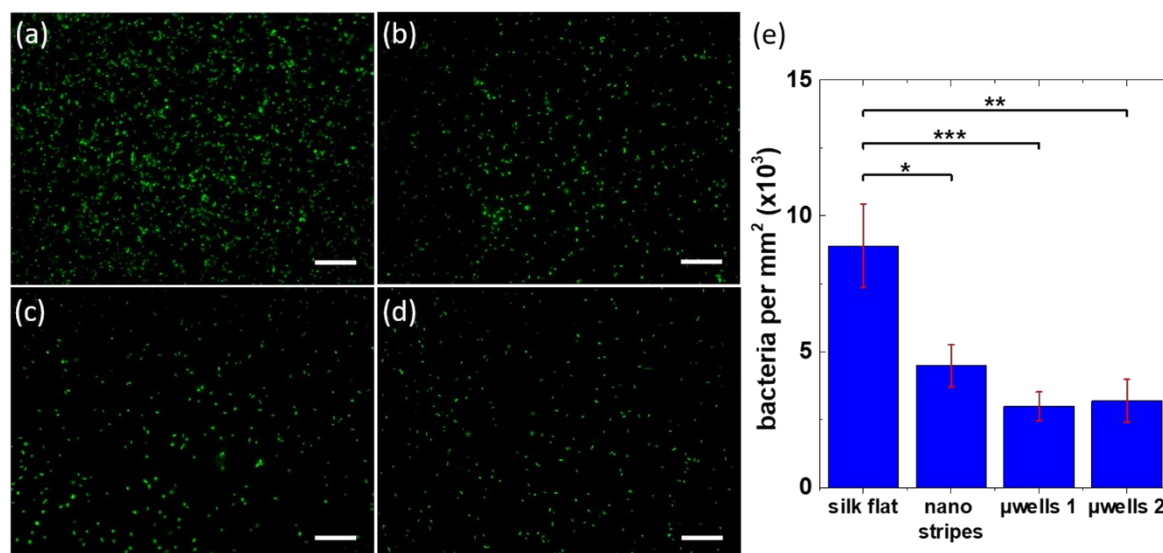
Once extracted from *Bombyx mori* silk following an established purification route,<sup>57</sup> silk fibroin shows optimal biocompatibility, as documented in several *in-vivo* and *in-vitro* studies.<sup>58–60</sup> However, much less is known about the cytocompatibility of patterned silk fibroin substrates. Thus, by culturing HEK-293 cell models on top of the different micro- and nano- patterned silk fibroin-based substrates, we directly investigate whether structured surface topographies sizably affect the viability of living cells. Cell proliferation is monitored by performing the MTT assay (**Figure 3**). The method is based on the use of the water-soluble tetrazolium salt (MTT), which is metabolized/reduced by living cells into a water-insoluble formazan product. As the optical absorption of formazan, which has a characteristic purple color, is proportional to the number of living cells, it can be directly related to the cell viability and capability to proliferate. Formazan absorption in the MTT-treated cell cultures on the different morphologies is evaluated after 1, 3 and 4 days *in vitro* (DIV), showing no statistically significant differences between micro/nano-patterned and planar silk topographies cases (**Figure 3**). This clearly indicates that the micro and nanostructures do not substantially affect the viability and the proliferation of HEK-293 cells, despite the fact that cells may be subjected to a higher surface pressure due to the reduced contact area. Instead, conditions favorable to cell proliferation are established, as also reported by recent literature and attributed to the higher surface energy and subsequent superhydrophilicity typical of the silk substrate.<sup>49</sup> In our case, there are no significant differences between flat and patterned silk substrates, but this observation may critically depend on the considered cell model, and it should not be considered of general validity. Interestingly, an enhancement in the proliferation of cells that were seeded on nanopatterned silk was recently observed in the case of human adipose mesenchymal stem cells, which usually feature exquisite sensitivity to the underlying substrate topography.<sup>61</sup>



**Figure 3. HEK-293 cells viability.** MTT viability assay on HEK293 cells plated on top of the different silk substrates. Data are reported as mean  $\pm$  s.e.m. A one-way ANOVA followed by post-hoc Dunnett's correction analysis reveals that none of the patterned silk substrates present any statistically significant difference in comparison to the planar silk samples.

### 3.3. Evaluation of *Escherichia coli* adhesion on pristine and structured silk substrates

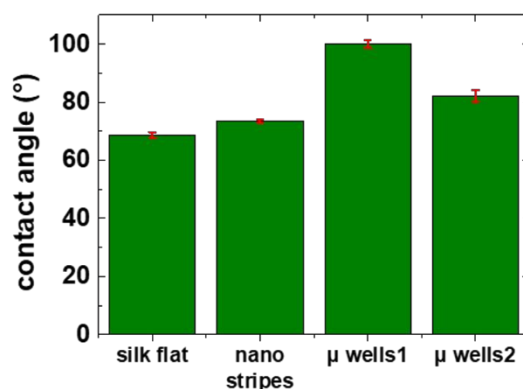
The antifouling capability of the different micro/nano-patterned silk samples is evaluated by employing the *E. coli* bacterial model, genetically modified to express green fluorescent protein (GFP), which is used as a visual marker. The substrates are incubated with the GFP-expressing *E. coli* for 24h, washed with ultrapure water for removing all unattached bacteria and covered with an ultrapure water drop. The number of adherent bacteria is then quantified by acquiring GFP fluorescence emission (**Figure 4a-4d**) using a fluorescence microscope. Measurements were acquired after 24 hours, when biofilm formation is known to be irreversible and fully completed.<sup>62</sup> All the silk samples display a lower number of adhered bacteria in comparison to the flat silk control. Importantly, the micro- and nano-patterned silk samples present a significant reduction in the number of *E. coli* attached to their surfaces in comparison with the flat silk morphology, amounting at 49 % ( $p < 0.05$ ), 66 % ( $p < 0.001$ ) and 64 % ( $p < 0.01$ ) for the nano stripes,  $\mu$ wells 1 and  $\mu$ wells 2 cases respectively (**Figure 4e**).



**Figure 4. Evaluation of bacterial adhesion on patterned silk.** Fluorescent microscopy images of GFP expressing bacteria cultured on flat (a), nano-striped (b),  $\mu$ wells 1-patterned (c),  $\mu$ wells 2-patterned (d) silk substrates, after 24h. Scale bars, 50  $\mu$ m. (e) Quantitative analysis of bacteria observed in a-d. \* $p < 0.05$ , \*\* $p < 0.01$ , \*\*\* $p < 0.001$ .

This result can be attributed to a topography-driven antifouling mechanism. In previous reports, two main effects were found to have a central role in the inhibition of bacterial adhesion: (1) a steric effect due to the presence of a regular micro-patterned morphology characterized by a features size lower than bacterial dimension,<sup>39,40,63</sup> and (2) the hydrophobic effect induced by micro-structured surfaces.<sup>39,41,64,65</sup> In both cases the driving force of the antifouling process relies on the reduction of the surface area to which the organisms can adhere. The reduced bacterial adhesion achieved with silk nano stripes and  $\mu$ wells 2 can be explained by considering the former mechanism, since the *E. coli* average length ( $\sim 2 \mu\text{m}$ ) and diameter ( $\sim 1 \mu\text{m}$ )<sup>66,67</sup> are higher than the width of the nano stripes silk grooves ( $\sim 800 \text{ nm}$ ), as well as the  $\mu$ wells 2 mean diameter (between 1 and 2  $\mu\text{m}$ ). On the contrary, this argumentation is not valid for explaining the bacterial adhesion reduction induced by  $\mu$ wells 1, because the diameter of the wells that compose this architecture (in the range between 3.5 and 5  $\mu\text{m}$ ) is about twice as large as the *E. coli* average dimension. In order to clarify the origin of the antifouling effect observed, experiments for the evaluation of the hydrophobicity of the samples are carried out by measuring the static contact angle of the considered surfaces (**Figure 5** and **Figure S1**).

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3 Generally, the limit between hydrophilicity and hydrophobicity is placed at a contact angle value of  $90^\circ$ .<sup>68</sup>  
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5 Planar silk is hydrophilic, presenting a water contact angle value of  $69^\circ \pm 1^\circ$ . Silk nano stripes and  $\mu$ wells 2  
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7 samples show increased water contact angle values ( $73^\circ \pm 0.5^\circ$  and  $82^\circ \pm 2^\circ$  respectively), but they still fall in  
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9 the hydrophilic regime, confirming that the observed decrease of *E. coli* adhesion is due to a steric effect  
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11 exerted by the surface topography and it is not strictly related to the surface wettability.  $\mu$ wells 1 substrates  
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13 display instead a hydrophobic behavior, with a water contact angle of  $100^\circ \pm 1^\circ$ . This result is in line with the  
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15 existing literature, where similar honeycomb surface topographies have been shown to influence the  
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17 wettability of polymer films.<sup>64,69,70</sup> In particular, this behavior has been explained by the formation of air  
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19 pockets between the surface of the substrate and the water droplets, when the latter are much larger than  
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21 the dimension of the structures present on the surface of the substrate.<sup>69</sup> This phenomenon could explain  
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23 the  $\mu$ wells 1 antifouling properties, since it directly determines the amount of surface that is available for  
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25 bacterial attachment. Our interpretation of the results for  $\mu$ wells 1 is also in line with the work by Manabe *et*  
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27 *al.*, showing that when the size of the pores lies between 3.5 and 11  $\mu\text{m}$  and the contact angle of the culture  
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29 medium is high, bacteria experience a limited contact with the surface.<sup>64</sup>  
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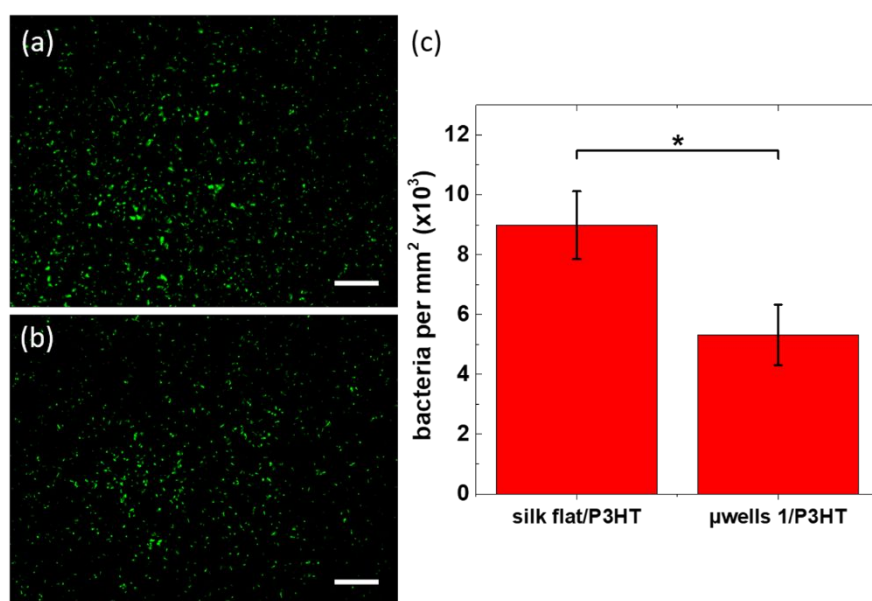
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49 **Figure 5. Water contact angle measurement.** Static contact angles established by water drops over planar  
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51 and micro/nano-patterned silk morphologies.  
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### 3.4. Evaluation of *E. Coli* adhesion on rr-P3HT-covered silk substrates

Among the numerous biotechnology applications of silk-based biomaterials, optics and biophotonics are emerging as some of the most attractive ones. Various types of implantable optical devices, including optical waveguides,<sup>14,71,72</sup> optical fibers<sup>73</sup> and microprism arrays,<sup>74,75</sup> based on biocompatible silk have been described in the literature.<sup>13,76</sup> This is due to the silk's good optical transparency, which ensures a high refractive index and low optical losses, even when used within *in vivo* systems. Silk has also been recently used as a long-term biocompatible substrate in the fabrication of functionally autonomous photovoltaic visual prostheses.<sup>60</sup> In this case, the excellent mechanical properties of silk were usefully coupled to the light responsivity and charge generation properties of a thiophene-based conjugated polymer, regio-regular poly(3-hexylthiophene-2,5-diyl) (rr-P3HT). It has been shown that a silk/polymer device is highly tolerated within the subretinal space of rats and partially recovers visual acuity in blind retinas.<sup>60,77,78</sup> These seminal works demonstrated that silk-based biomaterials can be coupled to optically-active polymers, thus providing a vast array of opportunities for the development of light-based diagnostic and therapeutic tools. In view of its huge *in vivo* application potential, we set out to investigate whether the excellent antifouling efficacy shown by micro-structured silk substrates is preserved even in the presence of a thin film of a semiconducting polymer deposited on top of the surface. We focused our attention on the substrate that gave the best antifouling result,  $\mu$ wells 1, and on the planar silk morphology as a control. The deposition of the layer of the semiconducting polymer is carried out by spin-coating, by selecting the proper rr-P3HT concentration and spin parameters that do not lead to a passivation of the micro-structured topography. Polymer thin film thickness is in the order of 30 nm, showing an optical absorption of  $\sim 0.3$ , which ensures good charge generation efficiency and photovoltaic properties also in a physiological-like environment.<sup>79</sup> SEM images of the rr-P3HT-covered/uncovered samples confirm that the honeycomb microstructure is not altered upon polymer deposition (**Figure S2**). *E. coli* adhesion is evaluated in rr-P3HT-covered silk samples (**Figure 6**). We notice that the average density of bacteria adherent to polymer-coated silk flat samples is similar to the one recorded in uncoated samples (**Figure 4**), thus allowing us to exclude a concomitant antifouling action due to the polymer thin film alone. We also notice that the hydrophobicity value typical of rr-P3HT polymer thin

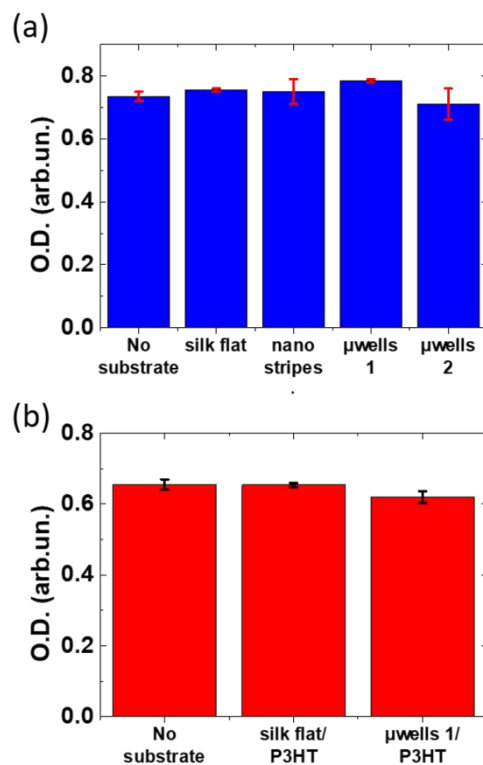
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3 films in dry conditions (about 100°) is substantially decreased when the polymer is directly exposed to an  
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5 aqueous environment, due to water permeation and ion doping, dropping up to 60°-70°. <sup>80</sup> Thus, the polymer  
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7 itself is not expected to sustain or to boost an antifouling effect. Importantly, however, the  $\mu$ wells 1/P3HT  
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9 sample displays a reduction of 41 % ( $p < 0.05$ ) in the adhesion of bacteria relative to the flat control, thus  
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11 demonstrating that the topography-driven antifouling effect of structured silk is substantially preserved also  
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13 in the presence of a thin film of a conjugated polymer deposited on top of its upper, bacteria-exposed surface.  
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38 **Figure 6. Evaluation of bacteria adhesion.** Fluorescent microscopy images of GFP expressing bacteria  
39 cultured on silk flat/P3HT (a) and  $\mu$ wells 1/P3HT (b), after 24h. Scale bar 50  $\mu$ m. (c) Quantitative analysis of  
40 bacteria observed in a,b. \* $p < 0.05$ .  
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46 In order to corroborate our findings, we carried out experiments aimed at the quantification of the  
47 bacterial growth inside the culture medium in the presence of all the tested substrates, thus providing an  
48 insight into the existence of possible sample-induced bactericidal effects. *E. coli* cultures are prepared in the  
49 same way as in the antifouling experiments case, and the bacterial growth is evaluated by measuring the  
50 culture medium optical absorption at 600 nm after 24h (**Figure 7**). The results show that there are no  
51 significant differences between the case where no substrate is present and the one where bacteria are  
52 incubated with both P3HT-uncovered (**Figure 7a**) and P3HT-covered (**Figure 7b**) silk samples. On the basis of  
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these findings we can conclude that the reduction of the surface-attached bacteria is only due to an unfavorable bacterial adhesion induced by the micro/nano-patterned topographies and we can exclude a bactericidal effect exerted by the micro/nano-patterned silk-based morphologies.



**Figure 7. Evaluation of the antibacterial activity of the micro/nano-patterned silk-based substrates.** Bacterial growth in solution after 24h incubation with P3HT-uncovered (a) and P3HT-covered (b) silk substrates in comparison to the case where no substrate is present inside the growth medium. Bacterial growth is evaluated as the optical density (O.D.) at 600 nm.

## Conclusions

A fast, facile and easily scalable processing method based on soft lithography is applied to realize nano- and micro-structured silk samples, with different geometries. We show that patterned silk substrates display excellent antifouling properties, leading in all cases to a substantial reduction of bacterial adhesion. At the same time, we show that the well-known cytocompatibility properties, which have been widely assessed in multiple studies for silk flat substrates, are fully preserved also in the case of silk microstructures. The antifouling effect is explained on the basis of a complex interplay between geometrical constraints, as the

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3 bacteria membrane has a lower surface area available for adhesion, and the enhancement of hydrophobicity  
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5 induced by microstructuring. In particular, we identify an optimal combination of the above mentioned  
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7 effects and we demonstrate that suitable silk patterning leads to a more than 65% decrease in bacteria  
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9 adhesion, as compared to flat silk substrates. As an important step towards the realization of fully-  
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11 biocompatible and functional devices for photo-medicine, we demonstrate that the antifouling properties of  
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13 patterned silk microstructures are fully preserved also in combination with a light-active conjugated polymer.  
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15 This paves the way towards the realization of a novel generation of bio-polymer implantable devices,  
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17 endowed with mechanical properties similar to those of biological tissues, and able to both optically  
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19 modulate/optically sense the activity of cells and to substantially inhibit biofilm formation.  
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## ASSOCIATED CONTENT

### Supporting Information

The following file is available free of charge. Representative photographs of water droplets in contact with the silk-based substrates. SEM top-view images of the bare and P3HT-covered silk substrates.

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## AUTHORS CONTRIBUTION

G.T., M.P., E.P., F.G. and M.R.A. planned the research. F.G. prepared the silk substrates. G.T. carried out SEM, AFM with help from F.G., fluorescence microscopy, cells viability assay with help from F.L., and contact angle measurements. S.D. prepared and characterized bacteria cultures. C.B. and F.L. helped with cell cultures preparation. G.T. carried out data analysis. G.T. and M.R.A. interpreted the data and wrote the main manuscript, with contribution from all authors. M.R.A. supervised the work. All authors approved the final manuscript.

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