

# ADVANCED MATERIALS

## Supporting Information

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Repurposing Poly(3-hexylthiophene) as a Conductivity-Reducing Additive for Polyethylene-Based High-Voltage Insulation

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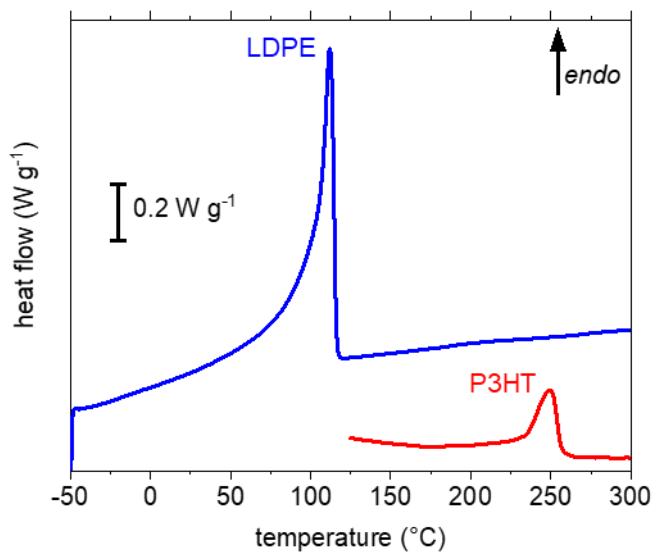
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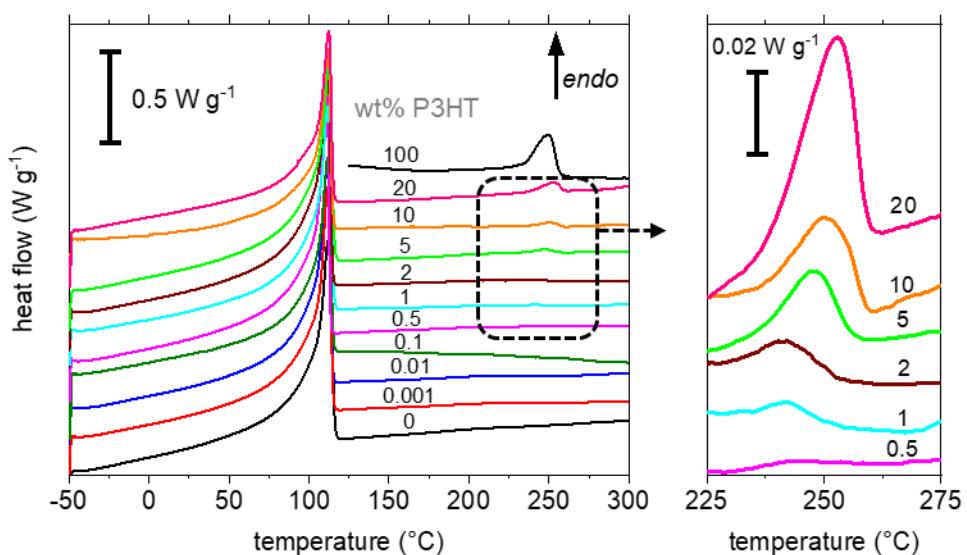
**Table S1.** Effect of different types of conductivity-reducing additives on the DC conductivity of LDPE expressed in terms of the DC conductivity  $\sigma_{DC}$  of the additive-containing resin relative to the DC conductivity of the neat LDPE resin  $\sigma_{DC}^{PE}$ , and as the efficiency  $\eta = (\sigma_{DC}^{PE}/\sigma_{DC})/\phi$ , where  $\phi$  is the additive content in wt%. We limit our survey to DC conductivity measurements where the electric field was applied for at least 1 h, although values approaching the steady-state are only obtained for much longer measurements.

<sup>a</sup>The matrix is crosslinked LDPE (XLPE); <sup>b</sup>Values of the reported charging current in A.

	type of additive	$\phi$ (wt%)	$\sigma_{DC}^{PE}$ (S m <sup>-1</sup> )	$\sigma_{DC}$ (S m <sup>-1</sup> )	$\sigma_{DC}^{PE}/\sigma_{DC}$	$\eta$ (wt% <sup>-1</sup> )	Ref.
inorganic nanoparticles	ZnO	3	$3 \times 10^{-14}$	$1 \times 10^{-16}$	300	100	[1]
	MgO	3	$3 \times 10^{-14}$	$5.5 \times 10^{-16}$	54	18	[2]
	Al <sub>2</sub> O <sub>3</sub>	3	$3 \times 10^{-14}$	$1 \times 10^{-15}$	30	10	[3]
	SiO <sub>2</sub>	2	<sup>b</sup> $6.5 \times 10^{-10}$	<sup>b</sup> $8 \times 10^{-11}$	8	4	[4]
polyolefins	HDPE <sup>a</sup>	1	$1 \times 10^{-14}$	$8 \times 10^{-16}$	13	13	[5]
hybrid	HDPE + Al <sub>2</sub> O <sub>3</sub>	4 + 3	$1 \times 10^{-14}$	$3 \times 10^{-17}$	333	48	[6]
carbon allotropes	graphene	0.1	$1 \times 10^{-14}$	$1.1 \times 10^{-15}$	9	90	[7]
	graphene	0.1	$2.5 \times 10^{-13}$	$1.2 \times 10^{-13}$	2	20	[7]
	graphene oxide <sup>a</sup>	0.01	$5.5 \times 10^{-14}$	$8 \times 10^{-15}$	7	700	[8]
voltage stabilizers	anthracene	0.5	$2 \times 10^{-11}$	$4 \times 10^{-12}$	5	10	[9]
	3- aminobenzoic acid <sup>a</sup>	1	$9 \times 10^{-13}$	$5 \times 10^{-14}$	18	18	[10]
	4,4'-dihydroxy benzophenone	0.5	$7 \times 10^{-15}$	$1.5 \times 10^{-15}$	5	10	[11]
conjugated polymers	P3HT	0.0005	$7 \times 10^{-14}$	$2.6 \times 10^{-14}$	3	6'000	this work

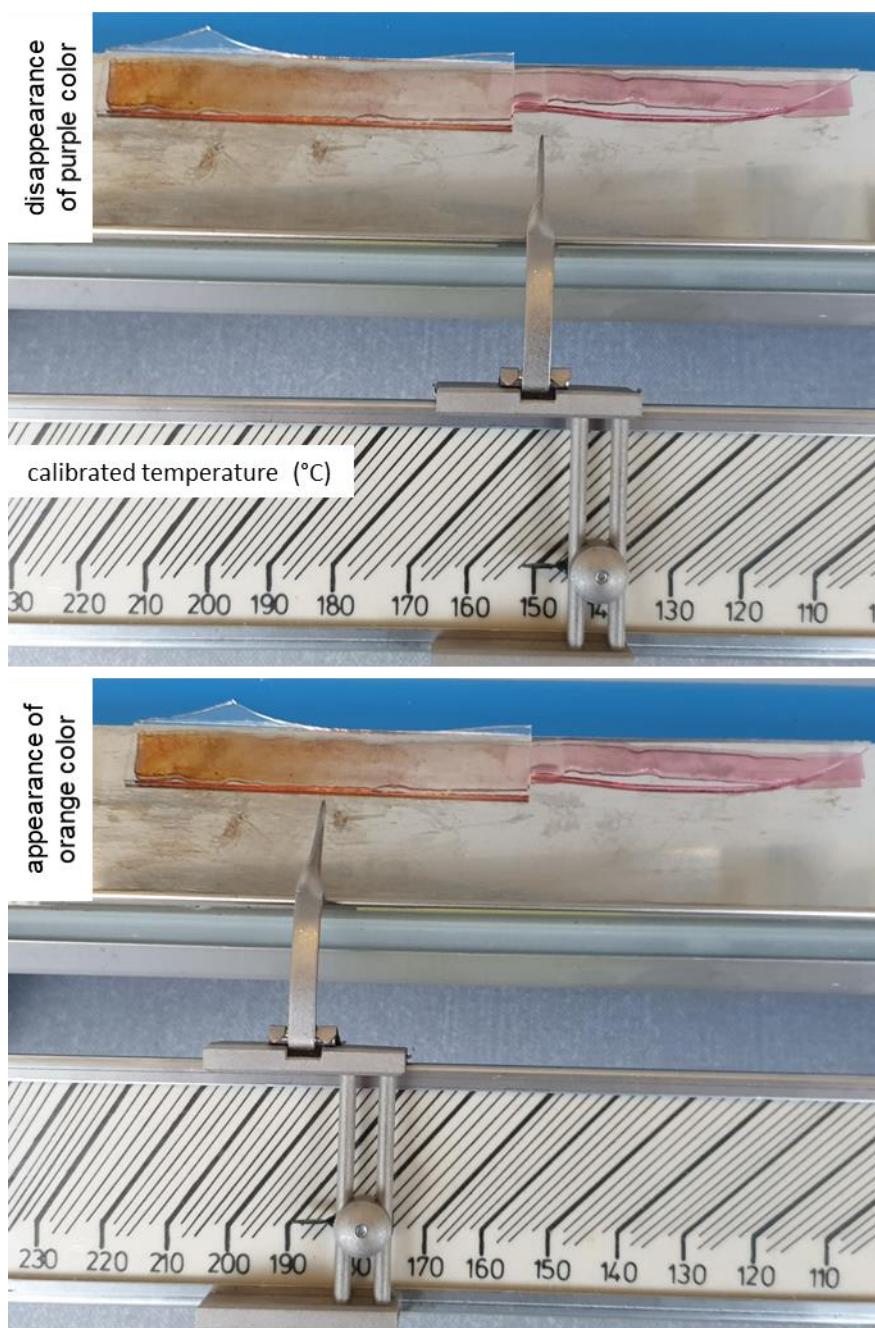


**Figure S1.** Second heating DSC thermograms of neat LDPE and P3HT.



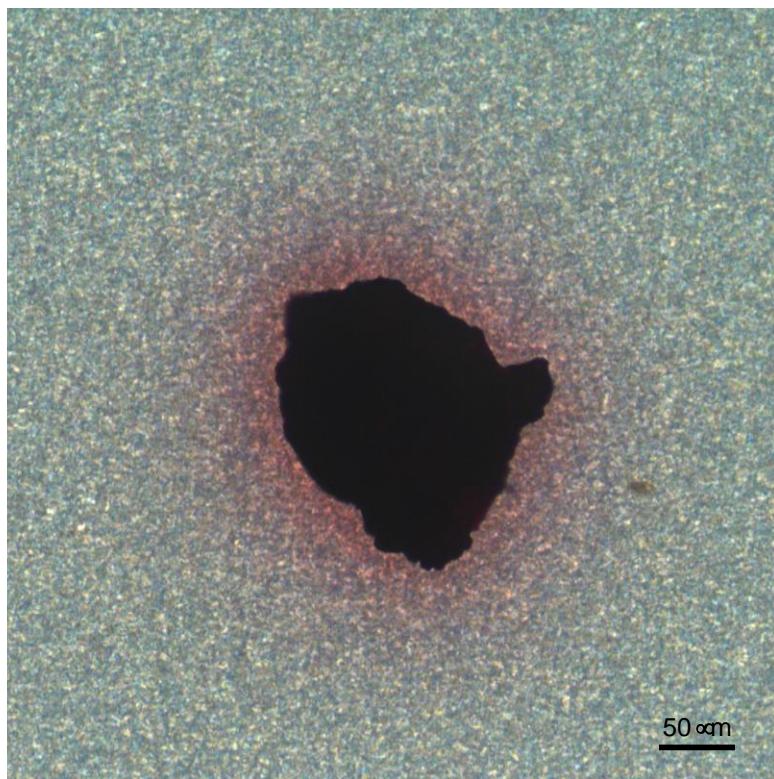
**Figure S2.** Second heating DSC thermograms of P3HT:LDPE blends.

$c_{P3HT}$  0.01 wt.%

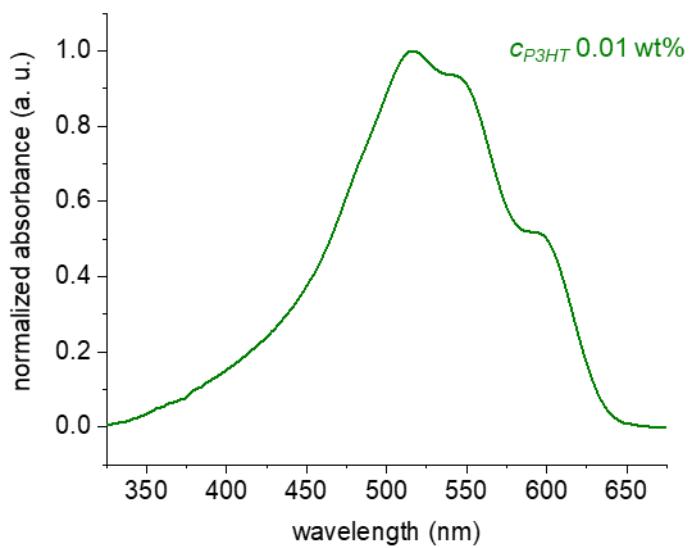


**Figure S3.** Visual detection with a Kofler bench of melting of P3HT in a P3HT:LDPE blend

$c_{P3HT}$  = 0.01 wt%; sample thickness ~ 0.3 mm.



**Figure S4.** Polarized optical micrograph of a P3HT grain placed on top of a 30  $\mu\text{m}$  thick sheet of polyethylene, sandwiched between glass slides and heated for 2 min at 250  $^{\circ}\text{C}$ .



**Figure S5.** UV-Vis absorbance spectrum of a melt-pressed plaque of a blend with  $c_{\text{P}3\text{HT}} = 0.01 \text{ wt\%}$ ; sample thickness  $\sim 1 \text{ mm}$ .

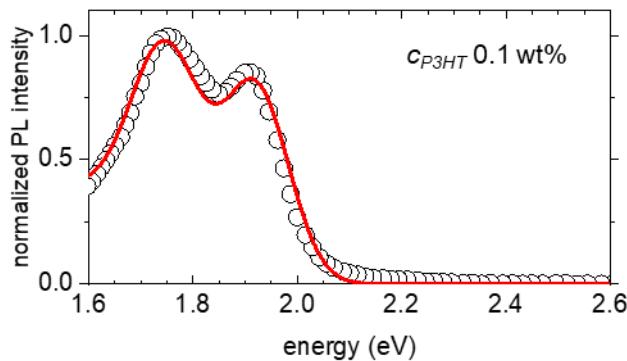
According to standard Franck-Condon standard progression, the relative intensity of the vibronic replica is given by<sup>[12]</sup>:

$$I_{0 \rightarrow m} \propto (\hbar\omega)^3 n_f^3 \frac{S^m e^{-S}}{m!} \quad (\text{Eqn. 1})$$

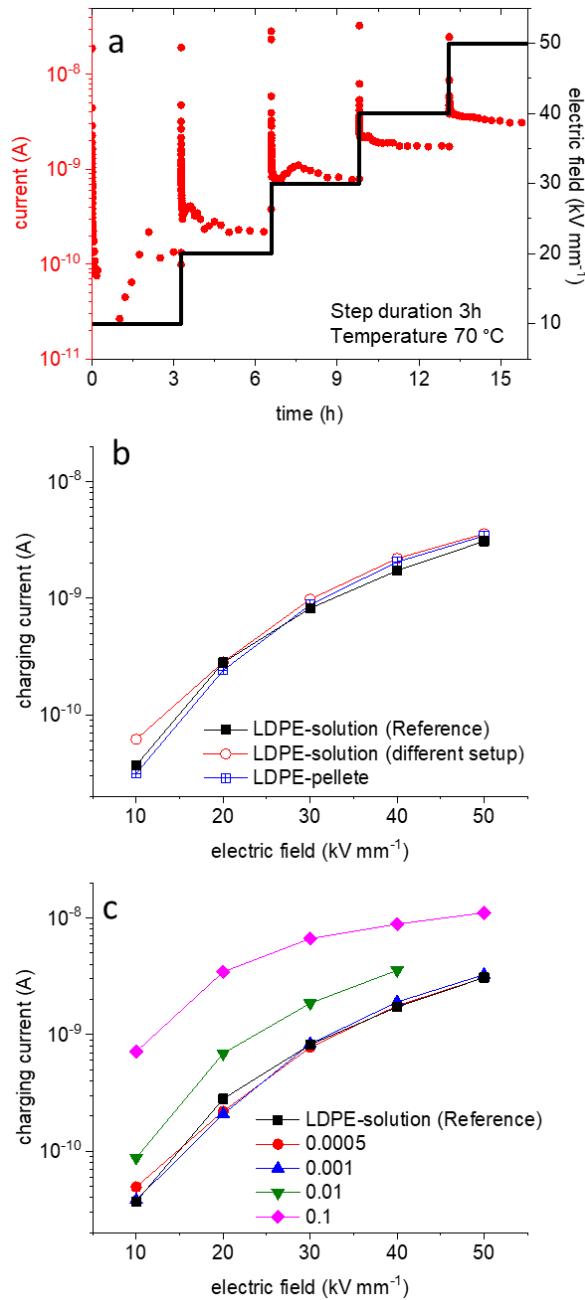
where  $n_f$  is the refractive index at the given photon energy of  $\hbar\omega$  at optical frequency of  $\omega$ ,  $m$  is the Franck-Condon vibronic index, and  $S$  is the Huang-Rhys factor. PL spectrum can be modeled as a modified Franck-Condon progression with a variable 0–0 amplitude<sup>[12]</sup>:

$$I(\omega) \propto (\hbar\omega)^3 n_f^3 e^{-S} \times [(\alpha \Gamma(\hbar\omega - E_0) + \sum_{m=1} S^m \frac{1}{m!} (\hbar\omega - (E_0 - mE_P)))] \quad (\text{Eqn. 2})$$

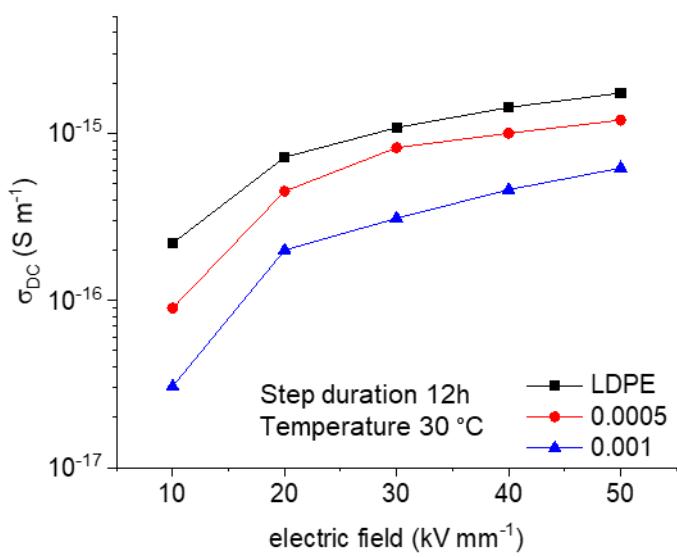
where  $E_0$  is the 0–0 transition energy,  $E_P$  is the phonon energy of the C=C symmetric stretch,  $\Gamma$  is the line-shape function (simplified to be purely Gaussian with a constant width), and  $\alpha$  is a constant, known as relative intensity of 0–0 band. In above fitting, the parameters of  $S$ ,  $E_P$  are kept constant and they are respectively, 1 and 0.18 eV. The fitted  $\alpha$  and  $E_0$  are obtained respectively,  $0.52 \pm 0.01$  and  $1.916 \pm 0.002$ , for the PL spectra of P3HT:LDPE blends with  $c_{P3HT} = 0.1$  wt%.



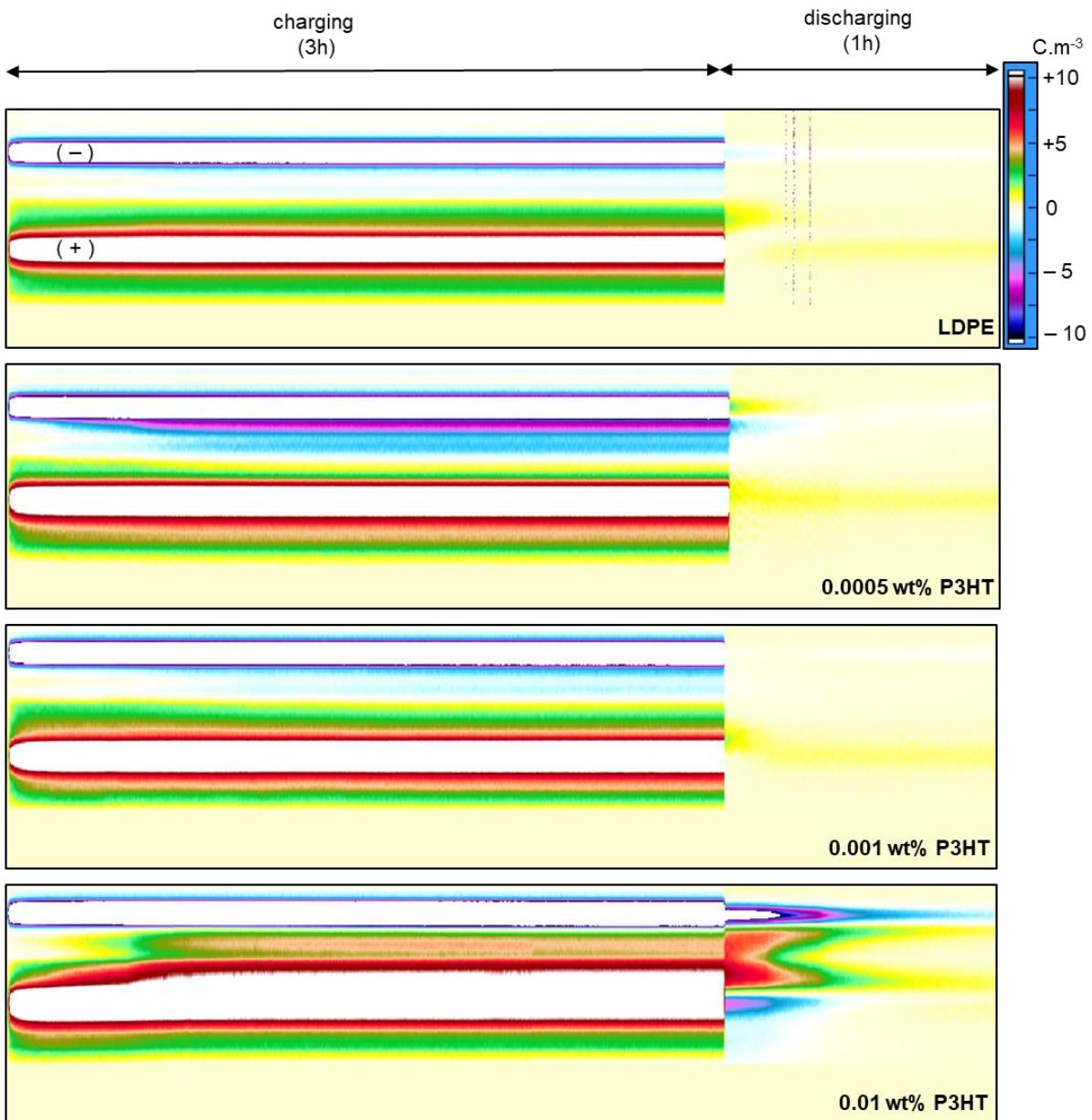
**Figure S6.** PL spectrum of a melt-pressed plaque of a blend with  $c_{P3HT} = 0.1$  wt% as well as a fit using the modified Franck-Condon model proposed by Spano et al.<sup>[12-15]</sup> (red solid line); sample thickness  $\sim 0.1$  mm.



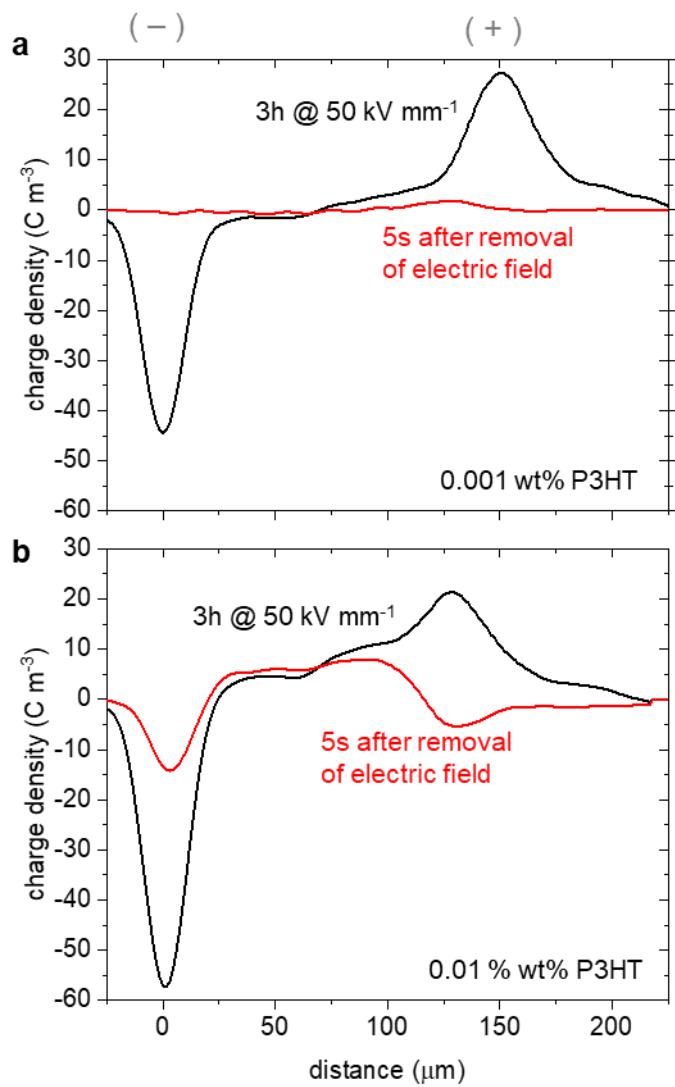
**Figure S7.** High-voltage charging current of P3HT:LDPE blends at 70 °C. (a) the applied step-wise electric field and resulting current as a function of time, (b) charging current at the end of each 3 h step as a function of the applied electric field for samples measured with two different setups (with the same arrangements and electrode dimensions) as well as different processing protocols (melt-pressing of precipitated material vs. as-received pellets), (c) quasi-steady-state charging current at the end of each 3 h step as a function of the applied electric field for blends with  $c_{P3HT} = 0.0005\text{-}0.1$  wt%.



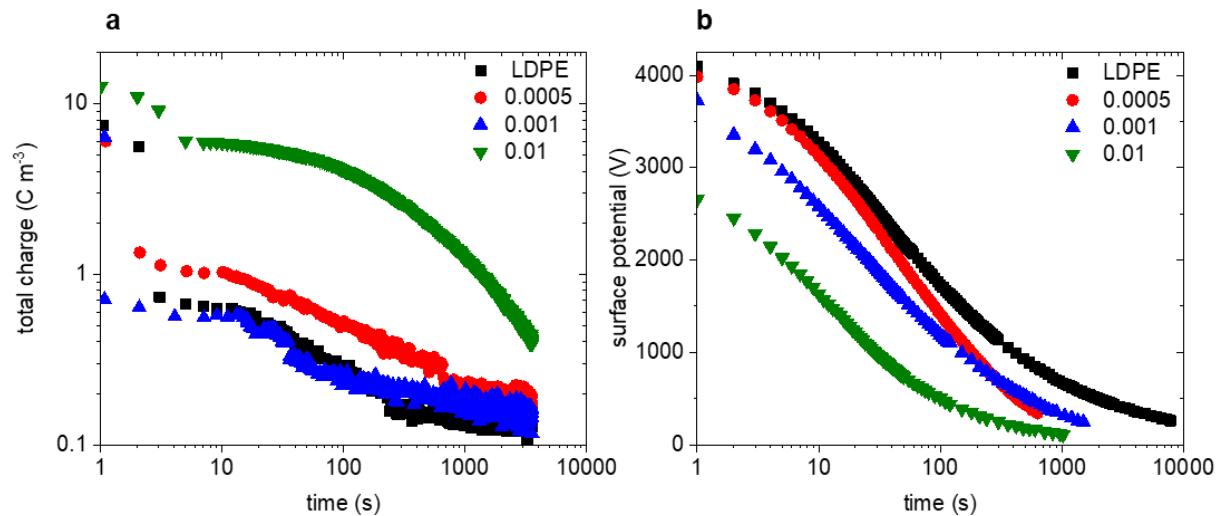
**Figure S8.** High-voltage DC conductivities of P3HT:LDPE blends at 30 °C.



**Figure S9.** Space charge distribution in melt-pressed plaques of LDPE and P3HT:LDPE blends at 70 °C; PEA space charge distribution of film samples during charging (3 h, applied electric field of  $50 \text{ kV mm}^{-1}$ ) and depolarization (1 h, removal of the electric field). The position of electrodes is shown as (-) and (+) for LDPE.



**Figure S10.** Space charge distribution of P3HT:LDPE blends with (a)  $c_{P3HT} = 0.001$  wt% and (b)  $c_{P3HT} = 0.01$  wt.% at  $70^\circ\text{C}$ .



**Figure S11.** (a) PEA total charge (not net-charge) decay during 1 h discharging, (d) isothermal surface potential decay (ISPD) after charging of 0.1 mm thick melt-pressed plaques at 8 kV.

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