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# Highly-dispersed ultrafine Pt nanoparticles on microemulsion-mediated TiO<sub>2</sub> for production of hydrogen and valuable chemicals via oxidative photo-dehydrogenation of glycerol

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#### Abstract

The oxidative photo-dehydrogenation of glycerol to produce H<sub>2</sub> and other valuable chemicals was studied using different materials. In particular, Pt nanoparticles were deposited on microemulsion-synthesized TiO<sub>2</sub> via surface organometallic chemistry (SOMC) and compared with photocatalysts obtained using more conventional methods. Well-defined Pt(II) single-site titania-grafted were prepared reacting the surface hydroxyl groups of TiO<sub>2</sub> nano-oxides with the organometallic Pt(COD)Me<sub>2</sub> complex. Sample reduction under H<sub>2</sub> generated ultrafine Pt nanoparticles well-dispersed on titania surface. Its performance under simulated solar light showed superior activity when compared to analogous Pt-containing catalysts prepared by other methods. Improved dispersion of Pt metal on titania surface was among the primary reasons of a better overall activity, providing relatively high rates of hydrogen productivity. Moreover, an increase of glyceraldehyde productivity in liquid phase was observed with the increase of Pt dispersion, demonstrating that the metal dispersion can strongly affect the selectivity of chemicals produced in the reaction. Comparison with state of the art shows that the present material exhibits excellent performance for a combined positive effect of the high specific surface area of titania prepared by microemulsion, giving access to the increased

densities of active sites and the high dispersion of Pt nanoparticles given by the SOMC technique.

## 1. Introduction

The use of renewable energy, adoption of biomass as feedstock and new technologies for energy production are among the sustainability megatrends [1]. In this view, glycerol, a cheap and renewable biomass feedstock, co-product of triglyceride transesterification, which amounts to *ca*. 10 kg (with purity of 50-55%) per 100 kg of biodiesel produced [2], is considered a chemical platform molecule with high potential. Glycerol is used as substrate in thermo-, electro-, and photo-reforming processes involving hydrogenolysis, dehydration, esterification, etherification, oligomerization, polymerization, carbonylation, and oxidation reactions [3], and produces a variety of high added-value chemicals for the agrochemical and polymer industry, which can also be used as solvents, surfactants, adhesives, coatings and building blocks for other compounds [4]. The production of hydrogen from glycerol is among these high potential routes. It is expected for example, *inter alia*, to increase the profitability of biodiesel production [3,5].

Both steam reforming and aqueous phase reforming of glycerol lead to hydrogen production [6,7], but high energy consumption and operating costs are required due to the endothermic character of these processes. Heterogeneous photocatalysis can be thus regarded as an interesting alternative, energy saving way of glycerol valorization in the context of sustainability. The process of photoreforming of oxygenated substrates in anaerobic conditions to yield H<sub>2</sub> and CO<sub>2</sub> using suspension of semiconductor-based photocatalysts has been extensively studied [8,9]. Among the semiconductors thermodynamically capable of glycerol photoreforming, such as oxides [10-17], metallates [18,19], sulfides [20,21], different composites [15,22-29] etc., TiO<sub>2</sub> has drawn the greatest attention, because it is a cheap, nontoxic, abundant material with a good thermal and photo-stability. The addition of a metal in the form of nanoparticles helps to enhance hydrogen production as they serve as co-catalyst for proton reduction reaction. In the last ten years [30] several metals have been used as titania dopants for glycerol photoreforming (e.g. Pt [11,31-35], Au [31,34,35] Pd [31,35-37], Ni [38], Cu [39]). Among these, Pt nanoparticles (NPs) showed the best results because of a greater work function of this metal [35,40] and the possibility to synthesize small clusters of around 1 nm, which display a high dispersion [9]. Nevertheless, Pt dispersion can be affected by the synthetic route employed for both support and Pt NPs, and interaction between them.

Surface organometallic chemistry (SOMC) is an established technique allowing synthesizing a heterogeneous catalyst with well-defined active sites by reacting in a controlled fashion the organometallic and coordination complexes with the molecularly-identified reactive surface sites of the support, typically a solid oxide. This approach allows in the best cases a structureactivity relationship as in homogeneous catalysis [41] and can lead to performing catalysts [42-45]. Recently, metal and metal oxide nanoparticles prepared by using different SOMC approaches [46-48] have been applied in catalytic reactions, e.g. soot oxidation [49-51], hydrogenation of toluene [52], dehydrogenation of isobutene and CO oxidation [53], styrene hydrogenation [54], alkene hydrosilylation [55]. Beneficial effect of Rh clusters in methane hydrogenolysis was observed for pre-grafted  $Rh(\eta^3-C_3H_5)_3$  on TiO<sub>2</sub> and subsequently reduced in hydrogen compared to the sample prepared by incipient impregnation [46]. Nonetheless, in the field of photocatalysis the use of photo-active materials prepared by SOMC is rarely described. Preparation of single-atom cobalt centers on different supports and its testing in photocatalytic water splitting was reported by H. Ahn and co-workers [56]. Recently, G. Jeantelot et al. [57,58] reported superior activity of isolated Pt atoms on TiO<sub>2</sub> and ultrafine Pt on SrTiO<sub>3</sub> using Pt(COD)Me<sub>2</sub> organometallic precursor in photocatalytic water splitting with methanol as sacrificing agent compared to Pt-containing materials prepared by impregnation method.

In present work, titanium dioxide was synthesized in the form of nano-powder using a microemulsion-mediated system. The synthesis delivered high surface area anatase TiO<sub>2</sub> polymorph with a small amount of rutile. XRD and DRIFT analyses were used to select the appropriate temperature for dehydroxylation process, which allows generation of well-defined sites on the surface of TiO<sub>2</sub>, in principle hydroxyls, capable for grafting of Pt precursor complex. The reduction of grafted Pt(COD)Me<sub>2</sub> organometallic complex on the surface of synthesized titania resulted in well-dispersed ultrafine Pt nanoparticles and provided higher dispersion than classical incipient wetness impregnation and deposition precipitation methods. The photo-catalytic activity of these samples in the glycerol oxidative photo-dehydrogenation process towards hydrogen evolution as well as production of liquid phase products were tested and compared.

#### 2. Experimental

#### 2.1. Catalyst preparation.

TiO<sub>2</sub> nano-oxides were prepared by reverse microemulsion method based on the modification of procedure reported by M. Andersson et al. [59], by breaking the microemulsion through heating under reflux. The modifications were introduced also in composition of microemulsion as follows: titania precursor was dissolved in organic instead of aqueous phase, water-to-

surfactant molar ratio was equal to 21.29. In brief, the oil phase composed by cyclohexane (99%, Alfa Aesar), dispersant, Triton X-100 (Alfa Aesar), surfactant, and n-hexanol (99%, Alfa Aesar), co-surfactant, was mixed with 5 M aqueous solution of HNO<sub>3</sub> (65%, Sigma Aldrich), forming transparent system of microemulsion. Another oil phase was prepared adding titania precursor, titanium (IV) butoxide (97%, Sigma Aldrich). After stirring until being transparent, this solution was added to previously prepared microemulsion under vigorous stirring. The resulting reverse microemulsion system was stirred for 1 h at room temperature, and then heated up to 74°C under reflux and kept for 5 h under vigorous stirring. The resulting solid was separated by centrifugation and washed with ethanol (99.8%, Sigma Aldrich) 5 times. The obtained paste was dried overnight at 100°C and calcined at 400°C for 3 h with the ramp of  $2^{\circ}$ C/min. Resulting solid powder is denominated as **TiO<sub>2</sub>-m**.

Physisorbed layers of water were removed by treatment under dynamic vacuum as follow: 1 g of catalyst **TiO<sub>2</sub>-m** was placed either in a 30 cm glass reactor and heated up to the target temperature (200°C or 500°C) or in a quartz reactor to reach 700°C with the rate of 3°C/min for 12 h under dynamic vacuum (10-5 mbar). Samples were transferred in a glovebox, and stored under argon. This dehydroxylated samples are called **TiO<sub>2</sub>-m-T** (T = 200, 500 and 700, respectively).

Grafting of Pt(COD)Me<sub>2</sub> over TiO<sub>2</sub>-m-500 surface was performed as follow: the powder TiO<sub>2</sub>-(758.7 was placed in a double Schlenk flask in m-500 mg) glovebox. Dimethyl(cyclooctadiene)platinum (19.5 mg, 0.058 mmol, Alfa Aesar) was placed in the second reactor of the double Schlenk flask inside the glovebox. The amount of Pt(COD)Me<sub>2</sub> was chosen to achieve the nominal loading of 1.5% wt, a common value in the literature for photo-reforming essays. Degassed and dried pentane (5 ml) was added under argon atmosphere to the platinum complex and the solution was then transferred onto powder. The reaction was kept for 2 h under stirring at room temperature. The gaseous products of reaction as well as solvent were transferred into 6 L glass balloon for GC and GC-MS analysis. The resulting solid Pt-OM@TiO<sub>2</sub>-m-500 was dried under dynamic vacuum at room temperature overnight and stored in a glovebox.

Solid **Pt-OM@TiO<sub>2</sub>-m-500** (400.4 mg) was transferred to 500 ml glass reactor and exposed to H<sub>2</sub> (450 mbar) at 300°C (5°C/min) for 3 h to yield the Pt NP-loaded sampled obtained from the organometallic route, named **Pt/TiO<sub>2</sub>-m\_SOMC**.

For comparison of preparation methods of Pt NPs and their effect on catalyst activity, two other samples were obtained by incipient wetness impregnation (IWI) and deposition-precipitation (DP) methods. For the IWI procedure, aqueous solution of  $H_2PtCl_6 \times H_2O$  (0.1868 M, 870 µl, 99.99%, Alfa Aesar, Premion®) was loaded on **TiO<sub>2</sub>-m** powder (2 g), followed by drying in oven at 100°C. Finally, the powder was dried at 120°C overnight, calcined at 350°C for 3h (ramp 10°C/min) in static air and sieved with 60-80 mesh. The powder was exposed to

 $H_2/N_2$  flow (100 ml/min, 10%) at 350°C for 3h (ramp 10°C/min) in static air, to yield material  $Pt/TiO_2-m_IWI$ . The DP synthesis was performed as follow: an aqueous solution of  $[Pt(NH_3)_4](NO_3)_2$  (0.001M, 0.117 ml, 99.9%, Alfa Aesar) was used at pH 8, adjusted by adding dropwise a 0.1 M NaOH solution. The resulting solution was added dropwise to the  $TiO_2-m$  suspension (1.5 g) in water (150 ml) under vigorous stirring at room temperature with a dropwise addition of 0.1 M NaOH to maintain pH 8. Once the entire platinum solution was transferred to the  $TiO_2-m$  suspension, the temperature was increased to 65°C, and the suspension was stirred for 2 h. The solid was then separated by centrifugation, washed several times with water, dried at 110°C overnight and calcined at 350°C for 3 h with a rate of 10°C/min) in static air and sieved with 60-80 mesh. The material (light gray powder, 1 g) was exposed to  $H_2/He$  flow (100 ml/min, 5%) at 350°C for 3h (ramp 10°C/min). The resulting catalyst was denoted  $Pt/TiO_2-m_DP$ . All the platinum-containing titania-based solids had a nominal loading in platinum of 1.5wt.%. The actual loading was determined by elemental analysis and is reported in Table 1.

# 2.2. Catalyst characterization.

The prepared materials were characterized in terms of their specific surface area, phase composition and light absorption properties, using nitrogen physisorption at the temperature of liquid nitrogen, X-ray diffraction (XRD), and diffuse reflectance UV-vis spectroscopy (DRS). **Nitrogen adsorption desorption isotherms** were measured on Belsorp-Max from BEL-JAPAN at 77 K after degassing the sample. No heat-pretreatment was performed on dehydroxylated samples (TiO<sub>2</sub>-m- 200, TiO<sub>2</sub>-m-500, TiO<sub>2</sub>-m-700, Pt-OM@TiO<sub>2</sub>-m-500) stored in the glove box. The samples prepared and stored in air (namely TiO<sub>2</sub>-m, Pt/TiO<sub>2</sub>-m\_DP, Pt/TiO<sub>2</sub>-m\_IWI) were pre-treated at 125°C for 2 h prior nitrogen adsorption. The specific surface area of the sample powder was calculated according to the BET model, BJH method was used to calculate the pore size.

**Powder X-ray diffraction analyses** of **TiO**<sub>2</sub>-**m**, **TiO**<sub>2</sub>-**m**-200, **TiO**<sub>2</sub>-**m**-500, **TiO**<sub>2</sub>-**m**-700 samples were performed using Bruker D8 Advance instrument (Cu, Kα radiation,  $\lambda = 1.5406$  Å, 40 kV, 4 MA). A thin smooth layer of the sample was deposited by evaporation from a suspension in ethanol on a non-crystalline substrate such as poly(methyl methacrylate). The XRD patterns were recorded from 5° to 70° at a step of 0.02° 20. The XRD of samples **Pt/TiO2-m\_A\_SOMC**, **Pt/TiO<sub>2</sub>\_IWI** and **Pt/TiO<sub>2</sub>\_DP** were carried out at room temperature with a Bragg/Brentano diffractometer (X'pertPro PANalytical) equipped with a fast X'Celerator detector, using a Cu anode as the X-ray source (Kα,  $\lambda = 1.5418$  Å). Diffractograms were recorded in the range of 10-80°20 with a step of 0.05°20. Particle size was determined by Scherrer equation over the 100% peak.

**Diffuse reflectance UV-vis spectroscopy** analyses were carried out in a Perkin Elmer Lambda 19 instrument equipped with integrating sphere in the range 280-800 nm. Diffuse Reflectance Infrared **Fourier Transform (DRIFT) spectra** were recorded on a Nicolet 6700-FT spectrometer using a cell equipped with CaF<sub>2</sub> window. DRIFT spectra were treated through Kubelka-Munk function (KM). **Elemental analyses** of C and Pt content were performed using CH-analyzer (Heraeus/Mannertz/Mikroanalytisches Labor Pascher) instrument and inductively coupled plasma (ICP) atomic emission spectrometer iCap 6500 (Thermo Fisher Scientific) at the Mikroanalytisches Labor Pascher in Remagen-Bandorf, Germany.

**Solid state NMR spectra** were collected on BRUKER AVANCE III 500 spectrometer operating at 125 MHz for 13C. The zirconia rotor of 4 mm was filled with the desired product inside glove box under argon and sealed with a kel-f stopper. It was then transferred into the probe Bruker CP 4 mm spectrometer allowing rotation of the rotor at a speed of 10 kHz. The time between two acquisitions was always optimized to allow complete relaxation of the protons.

**Transmission electron microscopy (TEM)** analysis of **Pt/TiO<sub>2</sub>-m\_SOMC** was performed using a JEOL 2100F electron microscope at the "Centre Technologique des Microstructures", Université Claude Bernard Lyon 1, Villeurbanne, France. The acceleration voltage was 200 kV. The samples were prepared by dispersing a drop of the ethanol suspension of a ground sample on a Cu grid covered by a carbon film. Whereas, TEM images of the samples **Pt/TiO<sub>2</sub>-m\_IWI** and **Pt/TiO<sub>2</sub>-m\_DP** were done at IMM\_CNR, Bologna, Italy. The analyses were carried out using TEM/STEM FEI TECNAI F20 instrument, equipped with a high-angle annular dark field (HAADF) detector at 200 keV. The corresponding samples were suspended in ethanol and treated in ultrasonic bath for 15 minutes. The suspension was deposited on a "quantifoil carbon film" Cu grid for TEM analysis, then dried at 100°C.

#### 2.3. Photocatalytic test and analytical methods.

Prepared materials were tested in oxidative photo-dehydrogenation of glycerol in a sealed topirradiated glass photo-reactor (**Figure S1**). Prior the reaction, 1 M, 21.5 ml aqueous solution of glycerol with 0.5 g/L of catalyst were stirred for 20 min in the dark, while purging continuously N<sub>2</sub>. The amount of catalyst was optimized following H. Kisch [60]. The sealed reactor with inner diameter of 4.6 cm was irradiated for 6 h using the solar simulator that consists of 300 W Xe lamp (irradiance of 100 mW/cm<sup>2</sup> was regularly measured prior each test by radiometer HD2102.2 DELTA OHM equipped with two probes of 315-400 nm and 400-1050 nm). Samples were collected at the end of the reaction. After reaction, the samples were qualitatively analyzed by Electrospray Ionization Mass-spectrometry (ESI-MS) and Nuclear Magnetic Resonance (NMR) techniques for identification of the products. Waters micromass ZQ 4000 instrument with quadrupole mass analyzer was used to perform ESI-MS analysis. The aqueous product solution was diluted in methanol and injected without further derivatization. Positive ions conditions were 3.53 kV for the probe and 10 V for the cone, while for negative ions the probe voltage was 2.54 kV and 40 V for the cone with a 20 µL/min flux. 600 MHz <sup>1</sup>H NMR analysis was performed using Varian Inova (600 MHz for 1H) equipped with an indirect triple resonance probe. The data treatment was done using a presaturation sequence (PRESAT, with power=4dB and presaturation delay 2s), while analyzing aqueous solutions (the residual  $H_2O$  signal  $\delta$  4.79 ppm). After the reaction, the liquid phase was filtered, diluted 100 times (from 1 M to 0.01 M) and analyzed in an Agilent HPLC over Rezex ROA Organic Acid column (0.0025M H<sub>2</sub>SO<sub>4</sub> eluent, oven temperature 30°C and 0.6 mL/min flux) with DAD and RID detectors. Aqueous solutions of commercial standards were used to calibrate the products by external method. Hydrogen was analyzed in off-line Agilent Technologies GC equipped with CP Molesieve 5A UM 25m x 0.53mm x 50µm column and TCD detector. The calibration was performed using constant volume (1 ml) of gas-tight syringe with different molar fractions of  $H_2/N_2$ . The other gas phase products were analyzed off-line by Autosystem XL GC with CARBOXENE 1000 60-80 mesh column and flame ionization detector, equipped with methanizer. The calibration was performed injecting different volumes of the commercial mixture of gases with constant molar fraction. The conversion of glycerol and the selectivity (S) of products were calculated as follows:

$$Conversion(\%) = \frac{\sum n(product), mol \cdot SF}{n(glycerol)_{in}, mol} \times 100\%$$

$$S\% = \frac{n(product), mol \cdot SF}{\sum n(product), mol \cdot SF} \times 100\%$$

where  $n(glycerol)_{in}$  is the initial moles of glycerol, *SF* is the stoichiometry factor, which is equal to 1 for glyceraldehyde, 2/3 for glycolaldehyde and 1/4 for H<sub>2</sub>; n is moles of corresponding product.

The solution obtained after the tests were analysed with an Agilent Technologies 4210 MP-AES Inductively Coupled Plasma (ICP) instrument. The analyses showed the absence of Pt in solution, assessing the absence of leaching of the active phase.

# 3. Results and discussions.

#### 3.1. TiO<sub>2</sub> catalysts characterization.

The titania obtained by microemulsion, **TiO<sub>2</sub>-m**, was mainly characterized as anatase crystalline polymorph with a small amount of rutile as can be seen from XRD patterns shown on **Figure S2** and **Table S1**. The crystallite size of **TiO<sub>2</sub>-m** was found to be 8 nm with respect

to anatase (101) reflection. This is an advantage of microemulsion method, which allows to obtain small crystallites with high surface area together with a control over the composition and morphology of the system compared to other techniques [61-63].

Thermal pre-treatment was carried out under vacuum at different temperatures in order to monitor the transformation of titania, and the presence and amount of isolated hydroxyl groups on the surface since they are considered as primary grafting sites for metal precursor.

After pretreatment at 200°C ( $TiO_2$ -m-200) XRD analyses (Figure S2) showed phase composition and crystallite size similar to the starting material,  $TiO_2$ -m. At 500°C the phase composition remained unchanged, although a slight growth of particles was observed in  $TiO_2$ -m-500 due to sintering. Following the 700°C thermal treatment, the particles size and rutile fraction increased drastically in  $TiO_2$ -m-700 as shown in Figure S2 (d), resulting in shrinking of surface area, while the pore size increased (Table S1). This is in agreement with the previous literature [64,65] where small anatase particles tended to agglomerate in temperature range above 700°C due to sintering. This led to the formation of big crystals of rutile, changing the porosity from inter-crystallite pores to inter-agglomerate ones.

DRIFT spectra reported in **Figure S3** provide information about adsorbed species and the presence of hydroxyl groups. The spectra show numerous bands above 3000 cm<sup>-1</sup> corresponding mainly to physisorbed water, chemisorbed water and surface titanols, Ti-OH [66-69]. While the distinct assignment of each band is beyond the scope of this work, it can be seen that: (*i*) physisorbed water, characterized by very broad bands below 3300 cm<sup>-1</sup> can be substantially reduced through thermal pre-treatment, as observed starting at 200°C; at the same time (ii) most of the strongly chemisorbed water, as testified *inter alia*, by the broad band around 3400 cm<sup>-1</sup> requires thermal treatment beyond 500°C to be substantially removed from the surface.

The peaks of a weak intensity that appeared around 2220 cm<sup>-1</sup> especially after dehydroxylation at 500°C (**Figure S3**) could be assigned to N-containing species [70-72] such as isocyanate groups (-NCO) or cyanate (-CN), whose presence is justified taking into account that in the microemulsion-mediated synthesis of **TiO<sub>2</sub>-m** aqueous solution of nitric acid was used as hydrolyzing agent.

Strongly physisorbed CO<sub>2</sub> in the range of 2460 and 2250 cm<sup>-1</sup> (**Figure S4**) indicates the presence of basic sites on **TiO<sub>2</sub>-m** and **TiO<sub>2</sub>-m-T**. Interestingly, K. Yamakawa and co-workers [73] reported different frequencies attributed to CO<sub>2</sub> adsorbed at different sites of TiO<sub>2</sub>. For instance, the peak at 2340 cm<sup>-1</sup> was assigned to defective sites related to oxygen vacancies (V<sub>o</sub>s), which in fact play an important role in photocatalysis, affecting the energy bandgap of the semiconductor. To note, after dehydroxylation process, samples changed color to grayish-blue depending on the temperature, indicating the formation of oxygen deficient sites. On the other hand, the peak at 2350 cm<sup>-1</sup> was associated with CO<sub>2</sub> adsorbed at the fivefold Ti<sup>4+</sup> sites

of anatase [73]. After dehydroxylation at 700°C, a peak at 2361 cm<sup>-1</sup> appeared due to stretching of  $CO_2$  adsorbed at  $Ti_{5c}$  on rutile (110) [74]. The presence of rutile is in agreement with the results given by XRD analysis.

In summary, dehydroxylation process removed most physisorbed water from  $TiO_2$ -m, giving access to non-mobile grafting sites (chemisorbed water and/or surface titanols) on the surface of titania by SOMC technique [43]. For grafting process,  $TiO_2$ -m-500 sample was selected as a compromise between desired crystalline polymorph, namely anatase, and the presence of sizable amount of surface reactive site for further functionalization.

## 3.2. Pt/TiO<sub>2</sub> catalysts preparation and characterization.

To examine whether a deposition method can affect the size and dispersion of Pt nanoparticles and, therefore, photocatalytic activity, three different synthetic techniques were used, viz. (i) incipient wetness impregnation, (ii) deposition-precipitation and (iii) grafting of Pt complex via surface organometallic chemistry (SOMC) route. Nominal loading of Pt over titania for all prepared sample was 1.5%wt. Detailed characterizations are summarized in **Table 1**.

**Table 1.** Data on specific surface area ( $S_{BET}$ ), crystalline size of titania ( $d_{TiO2}$ ), mean size of Pt ( $d_{Pt}$ ), dispersion of Pt (FE), energy bandgap ( $E_g$ ), elemental analysis represented in units of mass percent (%wt.).

Sample name	S <sub>BET</sub> (m²/g)	d <sub>⊤iO2</sub> (nm)	d <sub>Pt</sub> a (nm)	FE <sup>b</sup>	Element	%wt.	E <sub>g</sub> c (eV)	
Pt-OM@TiO <sub>2</sub> -m-500	121	-	-	-	С	0.97		
	121				Pt	1.22		
Pt/TiO <sub>2</sub> -m-SOMC	100	9	1.01	0.87	-	-	3.12	
Pt/TiO <sub>2</sub> -m_DP	128	7	0.93	0.31	Pt	1.22	3.11	
Pt/TiO₂-m_IWI	121	8	1.87	0.11	Pt	1.40	3.11	

<sup>a</sup> Particle size of platinum determined from TEM analysis for reduced samples.

<sup>b</sup> Pt dispersion calculated following A. Borodziński et al. [75] (see SI for more information).

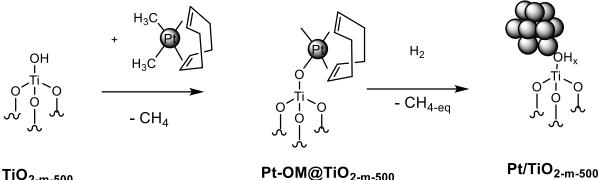
<sup>c</sup> Energy bandgap of reduced samples estimated from Kubelka-Munk spectra.

Grafting of Pt(COD)Me<sub>2</sub> over **TiO<sub>2</sub>-m-500** surface by SOMC technique was performed in a double Schlenk flask under argon atmosphere at room temperature.

The deposition of Pt precursor on solid **Pt-OM@TiO<sub>2</sub>-m-500** was monitored by DRIFT, solid state NMR techniques as well as elemental analysis.

DRIFT spectra on Figure S5 show disappearance of surface titanols and appearance of peaks around 3018 cm<sup>-1</sup> and 2950-2850 cm<sup>-1</sup> that correspond to alkenyl and alkyl C-H stretching of cyclooctadiene and/or methyl of the Pt precursor.

The GC and GC-MS analyses of the gaseous products released during the grafting of Pt(COD)Me<sub>2</sub> over **TiO<sub>2</sub>-m-500** showed exclusively CH<sub>4</sub>, which is in good agreement with reported literature [47,52,57], supporting the formation of monopodal [-O-Pt(COD)Me] and/or bipodal [-O-Pt(COD)-O-] species as shown in Scheme 1.



TiO<sub>2-m-500</sub>

Scheme 1: Overview of the main synthetic route to catalytically perfoming titania supported highly-dispersed platinum nanoparticles (Pt/TiO<sub>2</sub>-m-500): Starting dehydroxylated (T<sub>dehydrox</sub>= 500°C) microemulsion titnaia (TiO<sub>2</sub>-m-500, strongly chemsisorbed water omitted for simplicity); grafting of the organometallic precursor PtMe<sub>2</sub>(COD) to yield as major surface species the monoatomic grafted Platinum(II) complex Pt-OM@TiO<sub>2</sub>-m-500; reduction under dihydrogen.

Elemental analysis (Table 1) showed 1.22%wt. of platinum content for Pt-OM@TiO<sub>2</sub>-m-500, demonstrating that substantial amount (around 90%) of the organometallic complex dissolved in solution has been grafted. Solid-state <sup>13</sup>C{<sup>1</sup>H} CP-MAS NMR spectrum of Pt-OM@TiO<sub>2</sub>-m-500 is shown on Figure S6, revealing the resonance signals which are in agreement with previously reported works [47,52,57]. The ratio of C to Pt was found from elemental analysis to be 13:1, which is higher than the calculated one (9:1) for  $\eta^1$ -coordinated species [-O-Pt(COD)Me], suggesting that some pentane solvent was present in the solid.

Hydrogenation of material Pt-OM@TiO2-m-500 to generate Pt nanoparticles led to a rapid change of color to dark grey. Methane, ethane and propane were detected by GC as the outlet gases after hydrogenation. In addition, DRIFT spectra showed disappearance of C-H stretching of alkenyl and alkyl groups (Figure S5c), indicating that cyclooctadiene (COD) and  $CH_3$  were eliminated as hydrogenated volatile byproducts. Meanwhile, the recovery of surface hydroxyls was observed by IR, in line with the expected cleavage of Pt-O bond and formation

of Pt nanoparticles (**Scheme 1**). Therefore, Pt(COD)Me<sub>2</sub> can be considered as suitable thermolytic molecular precursor, easily leaving its organic moiety under heat treatment, for preparation of narrowly distributed Pt nanoparticles whose density is mainly determined at the grafting process by the density of hydroxyl groups [76].

TEM analysis of reduced samples prepared by the three different methods were carried out. TEM analysis revealed Pt NPs obtained by grafting technique of around 1 nm with a notably narrower size distribution compared to samples prepared by deposition-precipitation and incipient wetness impregnation methods (**Figure 1** and **Table 1**). While the depositionprecipitation resulted in similar amount, distribution and size of Pt to the organometallic route, the incipient wetness impregnation showed poor dispersion and formation of large clusters on the surface of titania (**Figure 1**).

Crystalline structure and optical properties of Pt-containing catalysts were analysed by XRD and DRS, respectively (**Figures S7, S8**). The comparison of the results obtained from XRD, DRS and nitrogen physisorption analysis and collected in **Table 1** did not show substantial difference.

**Table 1** reports the estimated dispersion, calculated following A. Borodziński et al. [75] (see SI for more information), and shows the best dispersion with the organometallic route as reported also in literature [47,77]. The reasons of this improved Pt dispersion is believed to be connected to the well-defined surface chemistry between reacting metal precursor and support, avoiding the formation of unreactive species coming from dissolution/precipitation steps when metal salts and metal oxide support are reacting in water [76]. Indeed, grafting the organometallic precursor on controlled surface titanol sites produced mostly uniform isolated Pt species, avoiding their agglomeration during preparation and also reducing their tendency to sinter when being already in the form of Pt nanoparticles anchored on the TiO<sub>2</sub> support. In general, the SOMC technique was demonstrated to be able to provide Pt supported catalysts with small active phase, narrow metal size distribution and higher dispersion than samples obtained by standard techniques.

#### 3.3. Photocatalytic test.

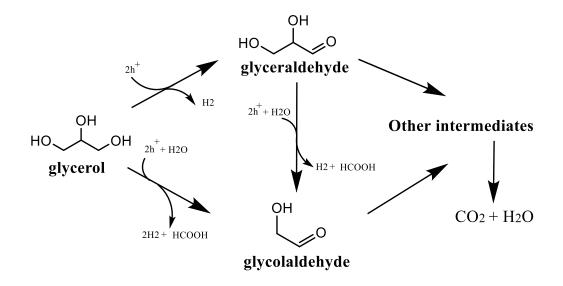
According to previous investigations [78-81], anaerobic photoreforming of glycerol mainly results in hydrogen production and in the formation of glyceraldehyde, dihydroxyacetone, formaldehyde, glycolaldehyde, hydroxyacetone and acetaldehyde as the principal species in liquid phase.

**Figure 2** shows hydrogen and liquid phase molecules production rates for pristine **TiO<sub>2</sub>-m** and supported Pt NPs by different methods. Pt addition provided a significant enhancement of the rate of hydrogen production compared to the bare support. This enhancement is associated to the migration of photo-excited electrons from the conduction

band of TiO<sub>2</sub> onto the supported metal nanoparticles, participating in proton reduction, and the ensuing rate decrease of electron-hole recombination phenomena [18,34,81,83].

The use of Pt co-catalyst permitted formation of detectable amount of liquid phase products such as glyceraldehyde and glycolaldehyde, which were the main compounds observed in these conditions. It is worth mentioning that Pt/TiO<sub>2</sub>-m\_SOMC exhibited slightly different rates ratio of liquid phase products. Figure 2 shows that the difference in the rates of glycolaldehyde and glyceraldehyde formation for Pt/TiO<sub>2</sub>-m\_SOMC sample is not as evident as for Pt/TiO<sub>2</sub>-m\_IWI and Pt/TiO<sub>2</sub>-m\_DP. Changes in acidity have been reported to be among the causes, because they lead to a different cooperative effect of Pt and support. At the same time, literature reports that solid oxides outgassed in different conditions do not change the acidity properties, so this parameter should not be the leading reason for our observed data [84]. As discussed below, the better dispersion might be the main reason (see Fig 4).

Formic acid in liquid phase and  $CO_2$ , CO and  $CH_4$  in gas phase were also detected in trace amount. No activity was observed when the reactions were performed in dark or when only visible light (>420 nm) was used. Although the process of light-driven conversion of glycerol is complex, meaning the production of multiple intermediates and different rates of their formation, a common feature for all the mechanisms is the formation of glyceraldehyde and glycolaldehyde among the primary products [79,78,85-88], which is compatible with observed data (**Scheme 2**).



**Scheme 2.** Suggested mechanism of glycerol oxidative photo-dehydrogenation. Adapted from [78,79,84-87].

The formation of glyceraldehyde as one of the primary products is believed to take place via indirect mechanism described by K. Sanwald and co-workers using Rh NPs supported on commercial titania P25 [78]. In principle, the photo-generated holes are trapped by lattice oxygen sites in TiO<sub>2</sub>, abstracting the H-atom from C-H bond with subsequent formation of carbonyl group. One molecule of hydrogen is produced concurrently at a cathodic site. Further oxidation of glyceraldehyde can yield glycolaldehyde and formic acid. On the other hand, glycolaldehyde can be formed also via direct C-C cleavage of glycerol as was reported by C. Minero [87], starting from a hole transfer to a chemisorbed glycerol on coordinatively unsaturated Ti(IV)-OH site to generate an alkoxide radical followed by  $\beta$ -C–Cscission. Thus, hydrogen and glyceraldehyde formation are favored in the presence of both Pt and TiO<sub>2</sub>, while glycolaldehyde production can occur on sole titania. For this reason a higher rate of hydrogen production and glyceraldehyde formation are observed at higher Pt dispersions. Further oxidation of glyceraldehyde to glycolaldehyde probably provides a concurrent increase in the rate formation of the latter.

As seen in **Figure 2**, higher  $H_2$  production rates were obtained with the sample prepared by SOMC. In order to verify the absence of leaching of Pt, ICP analysis was carried out on the reaction solution after centrifugation to remove the solid catalyst and the analysis confirmed the absence of Pt in the reaction mixture.

A correlation between hydrogen productivity and Pt dispersion for tested materials is shown in **Figure 3**, suggesting that improved hydrogen production rate could be correlated to better homogeneously dispersed Pt NPs on titania surface. Analysis of selectivity of liquid phase products showed a decrease of glycolaldehyde and an increase of glyceraldehyde selectivity together with glycerol conversion as the Pt dispersion improves (**Figure 4**). Comparing the data from XRD, N<sub>2</sub> physosorption, TEM and DRS one can suggest that conversion of glycerol and glyceraldehyde selectivity increase by the type of preparation method from IWI < DP < SOMC as the dispersion of Pt NPs over **TiO<sub>2</sub>-m** improves, confirming the importance of deposition method on the metal dispersion on the surface of titania. The stability of the catalyst was confirmed by performing consecutive tests using **Pt/TiO<sub>2</sub>-m**\_**SOMC** (**Figure S9**). The catalyst proved to be active in the reuse tests showing hydrogen productivities comparable to the first one.

Comparison of literature reports on hydrogen production using oxygenates by means of heterogeneous photocatalysis is complicated task because of the differences in light sources, geometry of photoreactors and reaction conditions used by different research groups. Recently, protocols and standardization to report the results in the field of heterogeneous photocatalysis have been proposed [89,90]. In an attempt to compare obtained results with the state of the art, different literature reports on hydrogen production from glycerol using platinum supported titania are summarized in **Table 3**. Comparing similar light source, glycerol and catalyst loading, the material prepared in this work outperformed the systems reported up to date [32,33,91]. On one side, this could be explained by a higher loading of Pt nanoparticles,

which play role as active sites capable of proton reduction. Nevertheless, the results of  $H_2$  production obtained for different Pt loading did not increase proportionally to the metal content [35,88], and in fact, in the range of 1-2%wt. of Pt the rate of  $H_2$  production was found to be comparable [11]. With the further increase of amount of metal co-catalyst the hydrogen production may decrease due to following reasons: decreased light absorption by titania surface (shading effect), decreased surface area of the metal co-catalyst due to the particles agglomeration, reduced metal co-catalyst dispersion on the surface of TiO<sub>2</sub> [9]. Thereby, improved photocatalytic activity in this work could be mainly explained by a high specific surface area of titania prepared by microemulsion, giving access to the increased densities of active sites and the high dispersion of Pt given by the SOMC technique. In fact, if the amount of glycerol and Pt added is taken into account in the hydrogen rate equation (Rate  $H_2$  in mmol/g<sub>gly</sub>/g<sub>Pt</sub>/h), the catalyst developed in this work highly outperforms other Pt/TiO<sub>2</sub> tested in similar conditions, evidencing that the high activity is not solely due to the Pt loading.

**Table 3.** Comparison of photocatalytic hydrogen production obtained in this work with results reported in the literature at similar conditions (Pt/TiO<sub>2</sub>, Xe lamp).

Type of titania	S <sub>TiO2</sub> , g/m²	Pt NPs <sup>a</sup>	wt% Pt	d <sub>Pt</sub> , nm	Catalyst loading, g/L	Glycerol loading, %vol.	Rate H <sub>2</sub> , mmol/g <sub>c</sub> <sub>at</sub> /h	Rate H <sub>2</sub> , mmol/ g <sub>gly</sub> /g <sub>Pt</sub> / h	Source	Ref
	49	IWI, H <sub>2</sub>	1	1.8	0.6	3.2	3.30	55	300 W	[33,92]
P25 (Evonik		PD		2.2			2.04	34		
Degussa)		In situ		2.0			4.28	71	Xe	
		PD		2.0			4.20	71		
TiO <sub>2</sub>	71	In situ	0.2	1-3	0.6	7	1.3	92	150 W	[90]
(hydrothermal)		PD							Xe	
P25 (Evonik	41	IWI, H <sub>2</sub>	0.5	1.2	1.3	8.1	2.7	18	300 W	[32,87]
Degussa)									Xe	
TiO2-m	164	DP, H <sub>2</sub>	1.5	0.9	0.5	7.2	3.68	125	300 W	This
		IWI, H <sub>2</sub>		1.9			3.06	104		
		SOMC,		1.0			4.93	167	Xe	work
		$H_2$								

<sup>a</sup> Method of Pt NPs preparation: IWI,  $H_2$  – impregnation of platinum salt with subsequent reduction in hydrogen; PD and *in situ* PD – photodeposition and *in situ* reduction prepared in different conditions; DP,  $H_2$  – deposition-precipitation with subsequent reduction in hydrogen; SOMC,  $H_2$  – grafting of SOMC complex with subsequent reduction in hydrogen.

#### 4. Conclusions

Organometallic complex Pt(COD)Me<sub>2</sub> was grafted on the surface of **TiO<sub>2</sub>-m** prepared from reverse microemulsion system. The prepared material was then reduced in hydrogen forming well-defined Pt nanoparticles with a small 1 nm size, and a narrow size distribution. The presence Pt NPs on the surface of titania efficiently reduced electron-hole recombination and enhanced the hydrogen evolution as well as liquid phase products formation, namely glyceraldehyde and glycolaldehyde. The prepared sample outperformed in the reaction of glycerol oxidative photo-dehydrogenation the samples with Pt deposited by incipient wetness impregnation and deposition-precipitation owing to a better dispersion of Pt nanoparticles. Besides, it has been observed that the dispersion can affect the selectivity of liquid phase products together with hydrogen production rate. With increase of Pt dispersion H<sub>2</sub> productivity raised, selectivity of glycolaldehyde decreased, while glyceraldehyde selectivity increased. Comparison of the obtained results on hydrogen productivity with the state of the art revealed that high surface area of support obtained by microemulsion synthesis and highly dispersed ultrafine Pt NPs given by SOMC method improved the photocatalytic performance, providing hydrogen productivities higher than previously reported works.

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