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This is the final peer-reviewed author's accepted manuscript (postprint) of the following publication:

*Published Version:*

Aşkın, A., Altınbaş, S., Giacinti Baschetti, M. (2025). Diffusion of organic solvents in thermoplastic elastomers: infinite dilution experiments. POLYMER BULLETIN, 82(3), 817-835 [10.1007/s00289-024-05545-x].

*Availability:*

This version is available at: <https://hdl.handle.net/11585/1007953> since: 2025-03-17

*Published:*

DOI: <http://doi.org/10.1007/s00289-024-05545-x>

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# **Diffusion of organic **solvents** in thermoplastic elastomers: infinite dilution experiments**

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## **Abstract**

Diffusion parameters, activity coefficients and Flory-Huggins interaction parameters of n-hexane, cyclohexane and benzene in several styrene-diene block copolymers (styrene-butadiene-styrene (SBS), styrene-ethylene/butylene-styrene (SEBS), and styrene-isoprene-styrene (SIS)) were obtained using the inverse gas chromatography (IGC) method with a packed column in the infinite dilution region and the range of 40-70 °C.

Diffusion coefficients based on the change in chromatographic peak width with carrier gas flow rate were calculated using the Van Deemter equation. The strong linear relationship indicated that using the Van Deemter equation was appropriate. The temperature increase caused the infinite dilution diffusion coefficient to increase, while the infinite dilution activity coefficient decreased; This proved the different behavior of the kinetics and thermodynamic parameters as a function of temperature. All solutes' infinite dilution diffusion coefficients decreased going from SBS to SEBS with the following order: SBS>SIS>SEBS. Activation energies were calculated from the Arrhenius equation, which provides the temperature dependence of diffusion coefficients.

**Keywords:** Activity Coefficient, Diffusion Coefficient, Inverse Gas Chromatography, Thermoplastic elastomers.

## Introduction

Knowing the equilibrium and mass transfer properties is of great concern in various manufacturing operations, such as drying, devolatilization, polymerization, and vacuum/gas stripping involving polymers [1-4]. Understanding the solvents' activity and diffusion coefficients in melts of polymers helps analyze basic processing stages including plasticization, bulk polymerization, and devolatilization.

Furthermore, due to the tightening environmental and health regulations governing the volatile residuals in polymer products, there is now more interest in the activity and diffusion coefficients at infinite dilution for volatile solvents in polymer melts [5].

Most of the reported diffusion and activity coefficients have been measured by conventional methods. They are based on the bulk equilibration and gravimetric sorption/desorption experiments and allow the determination of the coefficients over a relatively extended range of solvent concentration. Applying these techniques to polymer-solvent systems presents challenges, such as longer experiment durations when the solvent is present in minimal amounts or near or below the glass transition temperature, particularly when the diffusion value is low [6].

Inverse gas chromatography is another standard method that proved very suitable for describing the sorption kinetics and the thermodynamic behavior of gas polymer mixtures in the limit of infinite dilution [7]. It avoids the time limitation imposed by other methods. It provides much valuable information regarding molecule-molecule interactions, which are visible only in the dilute regimes and can only be assessed by different methods through regression from data in the finite solvent concentration region [8]. However, data from the literature appears to support the similarities between thermodynamic data from IGC and other static techniques such as gravimetric sorption methods [9, 10].

Understanding polymer transition phenomena is crucial for choosing materials for different applications and comprehending polymer features. The crystalline melting point and the glass-rubber transition are the two main transitions. Therefore, in polymer chemistry, determining the glass transition temperature ( $T_g$ ) and melting temperature ( $T_m$ ) is crucial both theoretically and practically. Many researchers have utilized the IGC technique in the literature to find the  $T_g$  value of homopolymers, copolymers, and polymer blends [4, 11-13] since it was first suggested by Guillet et al. [14] in 1969 to assign phase transitions in polymers. Plotting the logarithm of the specific retention volume against the reciprocal of

temperature allows phase transition to be identified using IGC data [15, 16]. Studies have also been conducted using temperature-dependent change curves of the dispersive component of the surface energy of n-alkanes adsorbed on polymers to predict temperature transitions in polymers [17-19].

The stationary phase and mobile phase compositions in the IGC technique are known. Analysis is used to determine how the two phases interact with each other. Compared to the static method, this technique is simpler, cost-effective, and time-efficient, and it only needs a small quantity of solvent and polymer. Additionally, the IGC has been suggested as a tool for researching solute diffusion into polymers [6, 20-22]. The slow solute diffusion in the polymeric stationary phase causes the chromatographic peak distortion and broadening. Consequently, the elution curve can be associated with the solvent's diffusion coefficient at infinite dilution in the polymer, provided the proper testing conditions are met. Many of these investigations have used packed columns [6, 23-25].

Because of their mechanical qualities and processability, styrene-diene block copolymers are employed in various industrial applications. These materials also form phase-segregated structures that contain rubbery diene and glassy styrene domains [26-28]. Therefore, these thermoplastic polymers have mechanical properties similar to crosslinked polymers at low temperatures. Still, they show high flexibility and good processability at high temperatures, at which polystyrene domains behave like rubber. In the literature, there are many studies in which the interaction parameters of some solvents with styrene-based block copolymers are examined by the IGC method [29-32] and shown to be consistent with those obtained with other techniques such as those based on intrinsic viscosity [33].

Romdhane et al. [30], used the IGC method at infinite dilution to rank the solubility strengths of some solvents for two SBS triblock copolymers (with a styrene content of 31% and 17%), showing that aromatic hydrocarbons are potential solvents for both triblock copolymers. In contrast, hydrocarbons have a higher affinity for the material with higher diene content. Diez et al. [29], determined polymer-solvent interaction parameters of two SBS commercial rubbers with different structures (linear and radial), and similar styrene content by the IGC method, while Wu et al. [31], analyzed the compatibility of SEBS (styrene content of 30.3%) and SIS (styrene content of 15.6%) triblock copolymers with different organic solvents. Similar studies were also performed by Ovejero et al. [33] based on intrinsic viscosity measurements at 20 °C SEBS (styrene content of 32%).

Despite the number of works in the field, in the open literature, very little information about the diffusion coefficients in the infinite dilution region of organic solvent in these block copolymers can be found as most of the data refers to finite concentrations (through the gravimetric method) [27, 34-37].

In this concern, the present study focused on the mass transport properties of three common organic solvents, cyclohexane, n-hexane, and benzene, in three styrene-based block copolymers. The three polymers had similar styrene content, about 30%, and were linear (SIS and SEBS) or star-branched (SBS). The infinite dilution diffusion coefficients of the probes in styrene-diene block-copolymers were determined, at several different temperatures, by IGC based on the Van Deemter model of the chromatographic process [38].

## Experimental

### Materials and Methods

Three different monodisperse styrene-based block copolymers with the same glassy polymer content (SBS, SIS, and SEBS) were kindly supplied for testing by Versalis S.p.A. Their various properties are given in Table 1. The three penetrants, each representing a class of organics, were the reagent grade cyclohexane, purchased from Sigma-Aldrich, and n-hexane and benzene purchased from Merck.

Table 1. Characteristics of the block-copolymers

Block-copolymers	Styrene content (wt%)	M g/mol	Polydispersity	Diene (1,2)
SBS (radial)	34	234000	1.2	9%
SIS (linear)	26	131000	1.14	3%
SEBS (linear)	34	54000	1.02	25%

## IGC Measurements

Utilizing an autosampler and a thermal conductivity detector, the Agilent 6890 gas chromatograph was used to perform the chromatographic measurements. The carrier gas was high-purity helium. Air was employed as an unretained compound to calculate the dead volume. A stainless steel column, 2 m long and 5.35 mm i.d. was cleaned with methanol and acetone before packing. Block-copolymers were soaked using Al-Saigh and Munk's method [39] to coat Chromosorb W-AW-DMCS from Alltech (60-80 mesh) at an amount of about 12% (w/w). Using typical ashing processes [38], the total loadings of SBS, SIS, and SEBS on the support were 11.73 %, 11.77 %, and 11.73 % (w/w), respectively.

The column was packed and conditioned in an 80 °C helium stream for an entire night. Tests were then performed by injecting a known amount of the **solvent** to be tested in the column and by observing the elution peak. Symmetric peaks were measured for each **solvent** probe, and the peak maximum was determined by taking the retention duration. Elution times were independent of the amount of **solvent** injected, confirming that tests were carried out in the Henry's Law region. Three determinations were used to average the net retention volume required for computations.

## Theory

### Infinite Dilute Diffusion Coefficient

The following equation gives an infinite dilute diffusion coefficient following van Deemter's model [38].

$$D_{12}^{\infty} = \frac{8d_2^2}{\pi^2 C} \frac{k}{(1+k)^2} \quad (1)$$

where  $C$  is a parameter associated with column properties,  $d_2$  is the average thickness of the polymer on the support particles, and  $k$  is the so-called partition ratio. By using the following equations,  $d_2$  and  $k$  values can be determined;

$$d_2 = \frac{w_2 \rho_s d_s}{6 \rho_2 w_s} \quad (2)$$

$$k = \frac{t - t_o}{t} \quad (3)$$

where  $d_s$  is the average support particle diameter,  $t$  and  $t_o$  are the solvent and unadsorbed material retention times (methane, air, etc.),  $w_2$  is the weight of the polymer coating on the support material,  $w_s$  is the weight of the support material,  $\rho_2$  is the density of the polymer, and  $\rho_s$  is the density of the support material.

According to Plate Theory,  $C$  in Eq 1 is derived from the following equation:

$$H = A + \frac{B}{u} + Cu \quad (4)$$

where  $A$  is an eddy diffusion-related constant, and  $B$  is related to axial diffusion in the gas phase based on the structural elements or tortuosity of the column. At sufficiently high flow rates, the  $B/u$  term is small relative to  $A+Cu$  and can be neglected.  $C$  is related to the other parameters of the probe diffusion in the liquid phase,  $H$  is the plate height equivalent to a theoretical plate (HEPT), and  $u$  is the carrier gas's linear velocity.

From the experimentally eluted peak,  $H$  is computed as follows [38, 40].

$$H = \left( \frac{l}{5.54} \right) \left( \frac{t_{1/2}}{t} \right)^2 \quad (5)$$

where  $t$  is the solvent retention time determined at the peak's maximum,  $l$  is the column length, and  $t_{1/2}$  is the complete peak broadness at half maximum. The carrier gas's linear velocity ( $u$ ) can be computed using the following expression:

$$u = \frac{j F_o T_c}{\bar{a} T_o} \quad (6)$$

where  $T_c$  and  $T_o$  are respectively the temperatures of the column and room temperature;  $F_o$  is the carrier gas flow rate,  $\bar{a}$  is the gas phase volume per unit length, which is the ratio of the

retention volume of unadsorbed material to the column length, and  $j$  is the compressibility factor obtained from the following relationship:

$$j = \frac{3}{2} \left[ \frac{\left(\frac{P_i}{P_o}\right)^2 - 1}{\left(\frac{P_i}{P_o}\right)^3 - 1} \right] \quad (7)$$

Where  $P_i$  and  $P_o$  are the inlet and outlet pressures of the column, respectively, thus, the compressibility factor can be obtained immediately from the measured values of  $P_i$  and  $P_o$  at any temperature and carrier gas velocity.

### Activation Energy

The Arrhenius equation typically provides the diffusion coefficient's temperature dependency [25, 41, 42]:

$$D^\infty = D_0 e^{-\Delta E_D / RT} \quad (8)$$

where  $D_0$  is the preexponential factor of the diffusion process independent of temperature,  $T$  is the absolute temperature,  $R$  is the gas constant, and  $\Delta E_D$  is the activation energy required for a diffusing molecule in a particular polymer matrix to diffuse into a neighboring environment that is distinct from its current surroundings. The diffusion rate is constrained in diluted solutions because it takes energy for the diffusing species to leave its current surroundings and migrate into a nearby environment when it does not frequently come into contact with a polymer molecule.

### Infinite Dilute Activity Coefficient and Flory–Huggins Interaction Parameter

The generalized equations that relate the measured data to the solvent activity coefficients in the polymer can be found below [6, 41, 42]:

$$\Omega_1^\infty = \gamma_1^\infty \frac{M_2}{M_1} = \frac{273.15R}{v_g^0 M_1} \frac{1}{\phi_1^s P_1^s} \quad (9)$$

where  $M_2$  is the molar mass of the polymer on the support solid,  $M_1$  is the solvent's molar mass,  $P_1^s$  is the pure solvent's saturated vapor pressure,  $\phi_1^s$  is the solvents' saturated fugacity coefficients,  $R$  is the gas constant,  $\Omega_1^\infty$  and  $\gamma_1^\infty$  are the solvents' activity coefficients in mass fraction and mole fraction scale, respectively. The Antoine constants were used to determine the solvents' saturation vapor pressures ( $P_1^s$ ), and the Soave equation of state was used to determine the solvents' saturated fugacity coefficients ( $\phi_1^s$ ) [43]. The following equation can be used to determine the retention volume per gram of polymer phase ( $v_g^0$ ) in Eq. 9 based on the retention time at temperature.  $v_g^0$  is calculated at 273.15K at infinite dilution of the solute.

$$v_g^0 = \frac{jF}{w_2} \frac{273.15}{T_o} (t - t_o) \quad (10)$$

where  $F$  is the carrier gas flow rate,  $t$  is the solvent retention time,  $t_o$  is the hold-up time required for inert gas to pass through the column, and  $w_2$  is the polymer mass on the column-packed support material.

According to the Flory-Huggins approach, the interaction parameter for a specific solute-polymer pair is defined as [44-46]

$$\chi_{12}^\infty = \ln \frac{273.2Rv_2}{V_g^0 V_1 p_1^o} - \frac{p_1^o}{RT} (B_{11} - V_1) - 1 \quad (11)$$

where  $v_2$ ,  $V_1$ ,  $B_{11}$ , and  $p_1^o$  are the specific volume of the polymer, the sorbate's molar volume, the sorbate's second virial coefficient, and saturated vapor pressure, respectively.  $V_1$  and  $B_{11}$  values are calculated at column temperature as described by Voelkel and Fail [46].

## Results and Discussion

### Diffusion coefficient

The infinite dilution diffusion coefficients of cyclohexane, n-hexane, and benzene in styrene block copolymers at various temperatures were measured using inverse gas chromatography. Every solvent-polymer pair was measured at multiple carrier flow rates.

Plots of plate height ( $H$ ) vs linear velocity ( $u$ ) were generated for each solvent in several styrene block copolymers at various temperatures to demonstrate the applicability of the Van Deemter equation. As an example, graphs of plate height ( $H$ ) versus linear velocity ( $u$ ) for SIS-Cyclohexane are given in Figure 1. For all solvents in all block copolymers, the slope of the linear relationships between  $H$  and  $u$  was used to calculate the values of  $A$  and  $C$  at each temperature (40–70 °C). From the latter parameter in particular the calculation of values was possible using Equation 1-7. The results are given in Table 2 and Figure 2 as a function of the different solvents' kinetic diameter ( $\sigma$ ). From the plot, it is possible to notice that the diffusion kinetics of the various solvents are reduced in the following order n-hexane>benzene>cyclohexane which is the same as the kinetic diameter of the different penetrants, thus highlighting the direct relationship between these two parameters. The existence of a correlation between diffusivity and penetrant size is often used to describe the transport behavior of different penetrants in polymers qualitatively. Among the others, Teplyakov and Meares [47] also tried to build a more quantitative relationship and from an empirical approach they noticed that a straight line with a slope of  $k_1$  and an intercept of  $k_2$  is frequently produced by fitting the logarithm of the diffusion coefficients as a function of the square of the kinetic diameter as follows:

$$\ln D^\infty = -k_1\sigma^2 + k_2 \quad (12)$$

A greater  $k_1$  value generally denotes stronger diffusivity selectivity of the polymer films. In Figure 2, the diffusivity data measured in the present work for the three-block copolymers, were plotted against the solvent's kinetic diameters according to Eq.(12). The linearity was very good in the case of SIS with an  $R^2=0.998$ . At the same time, SEBS and SBS showed lower correlation coefficient values that always remained above 0.85, confirming the good correlation between the two quantities. The  $k_1$  values increased in the SBS <SIS <SEBS sequence, suggesting that the latter polymer shows a stronger dependence of diffusivity from the kinetic diameter.

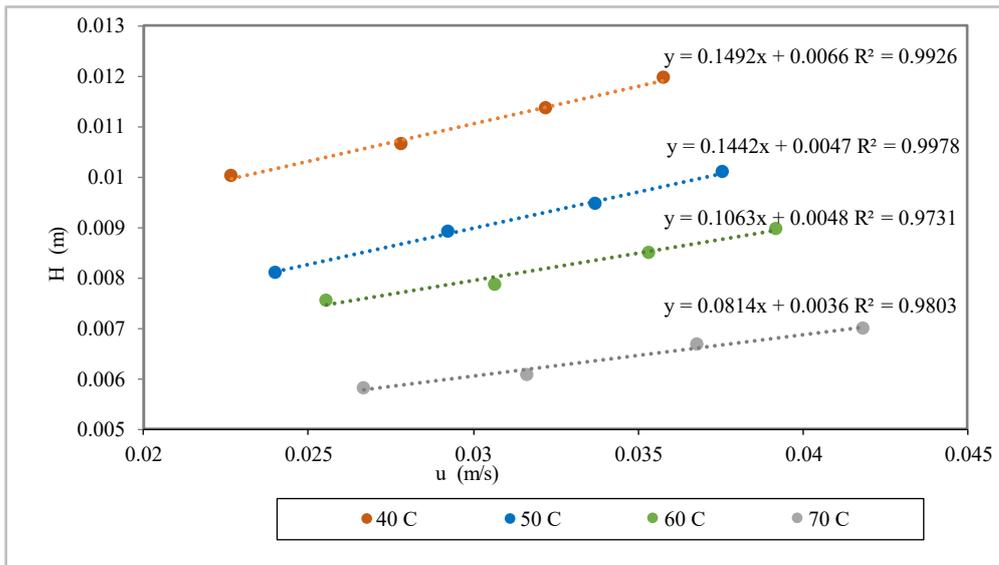


Figure 1. The plots of plate height (H) against linear velocity (u) for SIS-Cyclohexane

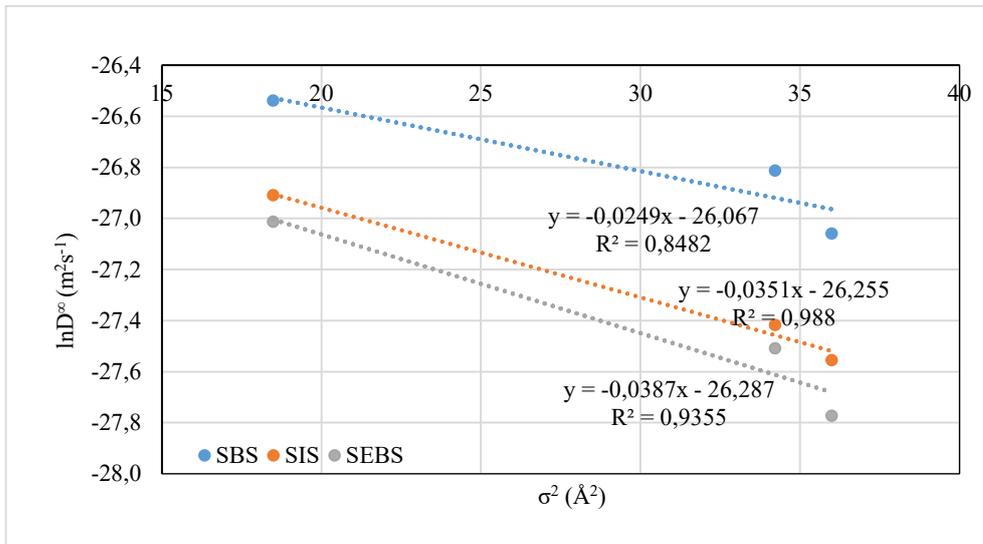


Figure 2. The logarithm of Infinite dilute diffusion coefficients ( $D^\infty$ ) at 40°C as a function of the kinetic diameter of the solvents squared ( $\sigma^*$ ). Kinetic diameters are taken from [www.mdpi.com/2073-4344/5/4/2147/s1](http://www.mdpi.com/2073-4344/5/4/2147/s1) [48]. ( $\sigma_{\text{benzene}}=5.85\text{\AA}$ ;  $\sigma_{\text{n-hexane}}=4.3\text{\AA}$ ;  $\sigma_{\text{cyclohexane}}=6.0\text{\AA}$ )

Table 2. Infinite dilute diffusion coefficients ( $D^\infty$ ), diffusion constants ( $D_0$ ), and activation energies for diffusion ( $\Delta E_D$ ) of the solvents

SOLVENT	T (°C)	$D^\infty \cdot 10^{12}$ (m <sup>2</sup> s <sup>-1</sup> )			$D_0 \cdot 10^8$ (m <sup>2</sup> s <sup>-1</sup> )			$-\Delta E_D$ (kJ.mole <sup>-1</sup> )		
		SBS	SIS	SEBS	SBS	SIS	SEBS	SBS	SIS	SEBS
Cyclohexane	40	1.77±0.01	1.08±0.00	0.87±0.01	14.01±0.7	39.5±1.1	77.70±2.2	29.17±1.9	33.5±2.9	35.94±3.9
	50	2.67±0.02	1.39±0.00	1.01±0.01						
	60	3.71±0.02	2.24±0.03	1.92±0.02						
	70	4.70±0.01	3.22±0.09	2.69±0.02						
<i>n</i> -Hexane	40	2.98±0.02	2.06±0.02	1.85±0.02	2.70±1.0	3.84±1.3	7.20±0.8	23.61±2.6	25.71±3.5	27.63±2.2
	50	4.36±0.01	2.44±0.05	2.33±0.02						
	60	5.62±0.00	3.82±0.07	3.31±0.02						
	70	6.58±0.00	4.63±0.05	4.62±0.01						
Benzene	40	2.26±0.02	1.24±0.02	1.13±0.04	7.90±0.9	6.72±0.8	16.80±1.3	27.22±2.3	28.40±0.5	30.98±3.7
	50	3.15±0.09	1.71±0.02	1.60±0.03						
	60	4.61±0.01	2.39±0.05	2.62±0.08						
	70	5.49±0.01	3.19±0.10	3.04±0.06						

In general, the infinite dilution diffusion coefficients of the solvents are reduced with the order of SBS> SIS> SEBS, with SBS showing higher diffusivity for the other materials which are instead rather close to each other. The results may suggest that the presence of pendant groups in the rubbery domains reduces the ability of the penetrant molecules to travel within the polymeric matrix, however, it is likely that the structure of the block copolymers, that is the distribution of the rubbery and glassy domains, plays a more important role in determining the effective diffusivity of the materials. It is well known that at the temperature inspected the styrene domains are well below  $T_g$  thus behaving as a glassy impermeable phase. Consequently, the transport mainly happens in the dienic phase, where the molecules are free to move. However, they are still strongly affected by the glassy phase, forcing them to follow a tortuous path to diffuse into the polymers. For example, data obtained here are at least 1 order of magnitude lower than those obtained in pure polybutadiene ( $5.7 \cdot 10^{-7}$  and  $1.1 \cdot 10^{-6}$   $\text{cm}^2/\text{s}$  at  $25^\circ\text{C}$  for n-hexane and cyclohexane respectively) [49] showing the non-negligible effect of the glassy phase on the overall properties of the materials.

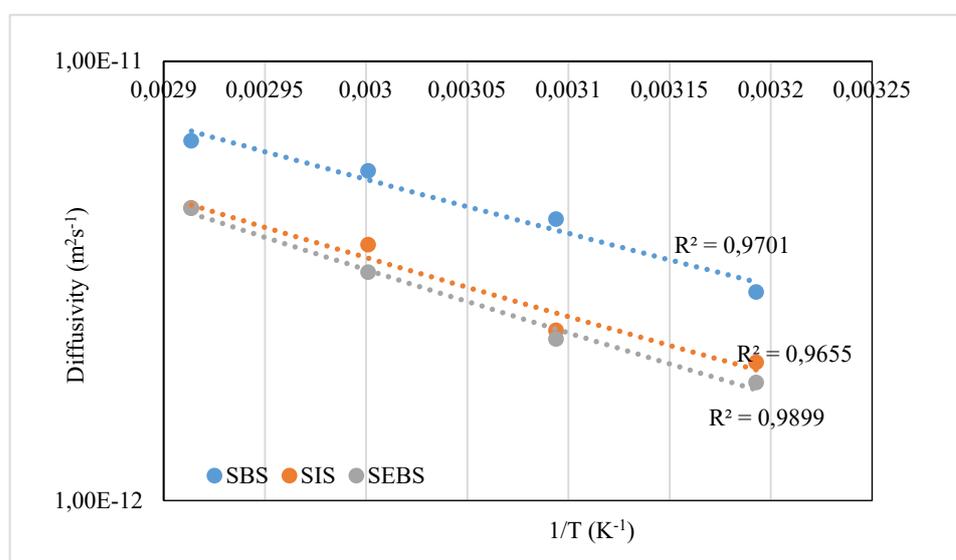
Romdhane and Danner [3] demonstrated the validity of the IGC method by measuring the diffusivity and solubility of toluene and benzene in polystyrene above the polymer's glass transition temperature. The diffusion coefficient of benzene in polystyrene varies between  $12.6 \cdot 10^{-9}$ -  $117 \cdot 10^{-9}$   $\text{cm}^2 \cdot \text{s}^{-1}$  in the range of 130 and  $160^\circ\text{C}$  operating temperatures.

There is no study in the literature where the diffusion parameters for concerned copolymers are determined by the IGC method. However, gravimetric studies (at finite concentration regions) have been conducted to examine the mass transport properties of cyclohexane, toluene, chloroform, n-nonan [27, 34, 36, 37, 50].

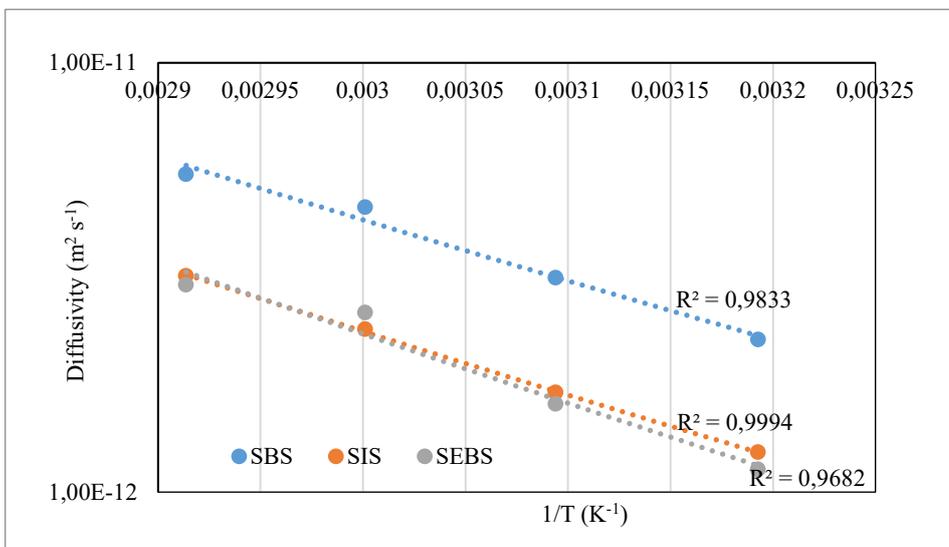
In an experimental study by Chiang et al. based on measuring the weight change over time for the adsorption of cyclohexane by SBS (styrene content 27.7% by weight), the approximate value of the equilibrium diffusion coefficient was found to be  $3.2 \cdot 10^{-7}$   $\text{cm}^2/\text{s}$  [27] According to Bacchelli et al.'s study [50], diffusion increased with penetrant (cyclohexane) concentration for all copolymers, but it increased more for SBS than for SIS and SEBS. In their experiments with the pressure decay technique at  $60^\circ\text{C}$  up to 0.8 activity (0.015-0.2  $\text{g}/\text{g}_{\text{pol}}$  penetrant concentration range), the diffusion coefficient of cyclohexane in copolymers decreased in the order SBS>SEBS>SIS at high penetrant values. Diffusion coefficients in SIS and SEBS approached each other at low penetrant concentrations.

In all three copolymers, the diffusion coefficients of all solvents increased with temperature. The graphs plotted according to the results of  $\ln D^\infty$  versus are presented in Figure 2. a, b, c for the three-block copolymers in n-hexane, benzene and cyclohexane respectively and results are consistent with the Arrhenius equation (Eq. 8). From the fitted straight lines, the activation energies of the three solvents are determined (in all lines,  $R^2$  values greater than 0.95 were obtained) and shown in Table 2.

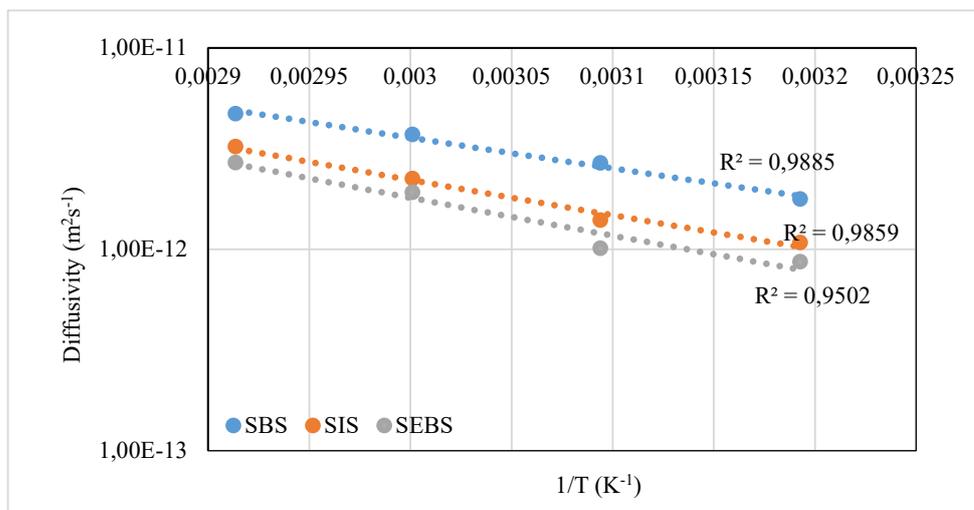
It can be seen from Table 2 that the diffusion activation energy is positive, as expected for such type of thermally activated process, with values ranging from 23 to 35 kJ/mol in line with the values found for the same solvent in other polymeric materials [25, 51, 52]. The activation energies increase in the SBS <SIS <SEBS order for all solvents. n-Hexane, then, shows lower activation energy than Benzene and Cyclohexane showing once again a clear effect of the kinetic diameter in determining the diffusion behavior of the different molecules. In infinite dilution conditions indeed, the effect related to swelling of the polymer phase or penetrant-penetrant interaction can be neglected and the activation energy is strictly related to the required energy for the penetrant molecules to jump from one polymer cavity to the next one. This latter energy is of course higher when bigger molecules are involved.



(a)



(b)



(c)

Figure 3.  $\ln D^\infty$  versus  $1/T$  (a) n- hexane; (b) benzene and (c) cyclohexane

## Activity coefficient

The activity coefficients ( $\Omega_1^\infty$ ) in the polymer phase were estimated using Equations 9 and 10 and presented in Table 3. At constant temperature, the  $\Omega_1^\infty$  values of solvents increase in the following order: benzene<cyclohexane<n-hexane for SBS and SIS, and cyclohexane<benzene<n-hexane for SEBS.

Benguergoura et al. [53] used capillary column inverse gas chromatography to determine the thermodynamic properties of poly(styrene-co-butadiene) rubber-various solvent systems. They compared their results with those of packed column studies in the literature. They determined that polar solvents had higher  $\Omega_1^\infty$  values than nonpolar solvents. Furthermore, they noted that the parameter is largely temperature-dependent in non-polar and weakly polar solvents. When the  $\Omega_1^\infty$  values of the solvent-SBS rubber pairs used in this study are compared with the values found in their study, it is seen that the coefficients are in a similar order (benzene<cyclohexane<n-hexane).

Table 4 gives infinite dilute Flory-Huggins interaction parameters obtained using Equation 11. The uncertainty of the data in Tables 3 and 4 is between  $\pm 1\%$  and  $\pm 5\%$ . Uncertainties were estimated as stated by Wang et al [42].

In the three copolymers, the Flory-Huggins parameters of n-hexane increased with temperature, while cyclohexane and benzene decreased with temperature. This parameter measures the free energy of the interaction between the components and the tested polymers. The dependence of the calculated parameters on the technologically specific temperature can be used to predict how the interactions between the compounds in the polymer-solvent systems will change with temperature. The decrease in the Flory-Huggins parameter with increasing temperature indicates that the interaction between the investigated polymers and compounds is increased when temperature is raised.

Table 3. Infinite Dilute Activity Coefficients of Solvents in Block-copolymers at Different Temperatures.

T (°C)	$\Omega_1^\infty$								
	Cyclohexane			<i>n</i> -Hexane			Benzene		
	SBS	SIS	SEBS	SBS	SIS	SEBS	SBS	SIS	SEBS
40	6.114	6.285	5.666	8.730	8.546	7.581	4.850	5.666	6.028
50	6.059	6.221	5.594	9.094	8.856	7.724	4.789	5.423	5.827
60	6.018	6.169	5.458	9.285	9.140	7.865	4.748	5.154	5.666
70	5.961	6.131	5.335	9.569	9.319	8.036	4.684	4.944	5.513

Table 4. Infinite dilute Flory-Huggins Interaction Parameters

T (°C)	$\chi_{12}^\infty$								
	Cyclohexane			<i>n</i> -Hexane			Benzene		
	SBS	SIS	SEBS	SBS	SIS	SEBS	SBS	SIS	SEBS
40	0.625	0.661	0.545	0.792	0.782	0.649	0.494	0.661	0.710
50	0.601	0.639	0.519	0.818	0.803	0.653	0.470	0.606	0.665
60	0.581	0.617	0.482	0.825	0.820	0.657	0.451	0.544	0.625
70	0.567	0.597	0.448	0.840	0.825	0.663	0.433	0.491	0.587

According to data in the literature, the temperature dependence of the Flory-Huggins parameter may vary depending on the type of solvent. For example, Romdhane et al. [30] in their study where they determined the Flory-Huggins parameters of SBS triblock copolymers using various solvents, the Flory-Huggins parameter of cyclohexane decreased with temperature, while that of chloroform increased.

Diez et al. [29] evaluated the effect of carbon number on Flory Huggins interaction parameters for two commercial SBS rubbers and different solvent pairs using the IGC method. They found that aromatic compounds had the lowest value and paraffinic compounds had the highest value. They attributed this result to the olefinic and aromatic character of SBS. They also stated that n-alkanes were the worst solvents for both rubbers due to the high values of the Flory-Huggins parameter and that the effect of temperature on the parameter was negligible in most of the polymer-solvent pairs examined. Similarly, it is observed that n-hexane has the highest value for SBS and benzene has the lowest value in this study conducted at identical temperatures.

The compatibility between polymers and solvents is examined using the polymer-solvent interaction parameter. The more compatible the polymer and solvent are, the lower its value. Benguergoura et al. [53] examined the Flory Huggins interaction parameter change depending on the molecule's molar mass and carbon number for poly(styrene-co-butadiene) rubber-various polymer solvent systems. They concluded that the parameter increases as the molar mass of the molecule increases and that aromatic and cyclic solvents containing the same number of carbons are more compatible with polymers. Similarly, in Table 4, it can be seen from the parameters of SBS that cyclohexane and benzene are more compatible with polymer than hexane.

The Flory-Huggins theory criterion of  $\chi_{12}^{\infty} < 0.5$  states that a polymer and solvent must be perfectly miscible throughout the composition range. In addition, for  $\Omega_1^{\infty}$ , the following rule has been built for the polymer-solvent systems [54]:  $\Omega_1^{\infty} < 5$ : good solvents;  $\Omega_1^{\infty} > 5$ : moderate solvents or poor solvents. The solubility of each probe for a given polymer was consistent based on the two theories mentioned above and comparisons with the values of  $\Omega_1^{\infty}$  and  $\chi_{12}^{\infty}$ .

Evaluating the values of  $\Omega_1^\infty$  together, it was found that the solvency capacity of the solvents increased in the sequence: n-hexane < cyclohexane < benzene for SBS and SIS; n-hexane  $\approx$  benzene < cyclohexane for SEBS.

Compared to these limit values, benzene is a good solvent for SBS at all operating temperatures and for SIS at 70°C. For SEBS on the other hand, cyclohexane can be defined as a good solvent at 60 and 70°C. In all other conditions tested these materials can be considered poor solvents for the considered copolymer, similar to n-hexane which showed limited affinity with all the materials investigated.

The same results were estimated for these solvents in the study by Jeong and Lee [55] in which a solubility test investigated the dissolution behavior of SBS, intrinsic viscosity measurements, and group additive test.

The comparison of the values found and the selected values presented in the literature by different authors are given in Table 5 for the three-block copolymers. It can be said that the small differences between the values and the ones in the literature are due to the molecular weight and styrene ratios of rubbers. The polymer's molecular weight and polydispersity can affect polymer dissolution in solvents. Polydisperse samples dissolved approximately twice as quickly as monodisperse samples with the same molecular weight [56]. On the other hand, Ovejera et al. [57] demonstrated that polymer-solvent interaction parameters are not affected by rubber linear and radial structures for SBS.

The SIS data in the table shows how the styrene content affects the parameter values. The values of copolymers with similar styrene content appear to be closer. The effect of molecular weight is less apparent. The SEBS data in the table shows this situation. In Table 5 it can be said that the small difference between the IGC and intrinsic viscosity measurement results of the parameters for SBS and SEBS is related to the polymer concentration rather than temperature. While viscosity measurements are performed at low polymer concentrations, IGC measurements are performed in the high range of polymer levels [32].

Table 5. Comparison of the present and the literature data

	$\chi_{12}^{\infty}$			$\Omega_1^{\infty}$			Reference
	Cyclohexane	n-Hexane	Benzene	Cyclohexane	n-Hexane	Benzene	
<b>SBS</b>	0.60	0.82	0.47	6.06	9.09	4.79	In this study
By IGC with a styrene (radial) content of 34% by weight $M_w$ : 234,000 g.mol <sup>-1</sup> , T=50 °C							
	0.56	0.79	0.40	5.88	8.79	4.46	[30]
By IGC with a styrene (linear) content of 31% by weight $M_w$ : 110,800 g.mol <sup>-1</sup> , T=55 °C							
	0.43	—	0.39	—	—	—	[57]
By intrinsic viscosity, with a styrene (radial) content of 30% by weight $M_w$ =237,000 g.mol <sup>-1</sup> , T=30 °C							
	0.33	0.69	0.34	4.40	7.43	3.93	[29]
By IGC, with a styrene (radial), content of 30% by weight $M_w$ =237,000 g.mol <sup>-1</sup> , T=50 °C							
	0.71	0.94	0.53	6.94	10.32	5.10	[29]
By IGC, with a styrene (linear) content of 31% by weight $M_w$ =113,000 g.mol <sup>-1</sup> , T=50 °C							
<b>SIS</b>	0.64	0.80	0.61	6.22	8.86	5.42	In this study
By IGC with a styrene (linear) content of 26% by weight $M_w$ : 131,000 g.mol <sup>-1</sup> , T=50 °C							
	0.48	0.46	—	5.52	6.57	—	[31]
By IGC, with a styrene (linear) content of 15.6 % by weight $M_w$ =130,960 g.mol <sup>-1</sup> , T=50 °C							
<b>SEBS</b>	0.52	0.65	0.63	5.59	7.72	5.83	In this study
By IGC with a styrene (linear) content of 34% by weight $M_w$ : 54,000 g.mol <sup>-1</sup> , T=50 °C							
	0.47	0.63	0.58	5.57	7.76	5.47	[32]
By IGC, with a styrene content of 32% by weight $M_w$ : 90,000 g.mol <sup>-1</sup> , T=50 °C							
	—	—	—	4.54	7.09	5.98	[32]
By UNIFAC-FV group contribution method [58]							
	0.45	0.61	0.52	—	—	—	[32]
By intrinsic viscosity, with a styrene content of 32% by weight, $M_w$ =86,238 g.mol <sup>-1</sup> , T=20 °C							
	0.44	0.38	—	5.25	5.99	—	[31]
By IGC with a styrene (linear) content of 30.3 % by weight, $M_w$ =71,880 g.mol <sup>-1</sup> , T=50 °C							

Therefore, the general affinity for the three-block copolymer seems to decrease in the following order benzene > cyclohexane > n-hexane, in line with previous studies on the subject as visible in Table 2. This behavior can be related to the different interactions with the styrenic and dienic domains in the materials. For example, studying the solubility of various solvents in SBS triblock copolymers with different styrene content; showed that Flory-Huggins parameters of some aromatic (such as benzene, toluene ethyl-benzene, p-xylene) and aliphatic (such as n-hexane, n-heptane) solvents decreases increasing the butadiene content of the materials suggesting higher affinity with this domain inside the system.

Considering the segregated structure of the considered block copolymer it can be interesting to analyze the results concerning the behavior in the corresponding homopolymers: polystyrene and polybutadiene. In this concern, the literature has reported that aliphatic hydrocarbons have higher solubility against polybutadiene than polystyrene [59, 60]. The affinity for polystyrene instead, decreased in the aromatic > alicyclic > aliphatic order, as reported by Tmsek et al. [61], which also observed that benzene was the best dissolving agent among the solvents studied. The affinity of the different solvents for the various materials therefore seems to be somewhat dominated by their affinity for the styrenic phase despite this having a lower concentration than the dienic one in the considered block copolymers.

## Conclusions

In the present study, mass transport properties of n-hexane, cyclohexane, and benzene in three block copolymers based on styrene (SIS and SEBS SBS) were examined at column temperatures between 40 and 70 °C by infinite dilution IGC experiments.

The diffusion coefficient was calculated using the data from the chromatography peak and the retention period using the Van Deemter model of the chromatographic process. The infinite dilution diffusion coefficients of the solvents are reduced with the order of SBS> SIS> SEBS. In all three copolymers, the diffusion coefficients of all solvents increase with temperature. The logarithm of the diffusion coefficient is linearly related to the square of the kinetic diameter. Diffusion activation energies are positive and increase in the SBS <SIS <SEBS order for all solvents.

Additionally,  $\chi_{12}^{\infty}$  and interaction parameters are used to analyze polymer-solvent compatibility. Assessing the values of these parameters together, it was found that the solving

capacity of solvents increased in the following order: n-hexane <cyclohexane< benzene for SBS and SIS; n-hexane  $\approx$  benzene < cyclohexane for SEBS. The interaction parameters obtained were compared with the literature.

### **Acknowledgments**

The authors are grateful to the Scientific and Technological Research Council of Turkey for the financial support of this study under a research project (GR/110M007) and to the Versalis S.p.A for the supply of the polymers. Ayşegül Aşkın thanks The Scientific and Technological Research Council of Turkey for Postdoctoral Research Scholarship.

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