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This is the final peer-reviewed author's accepted manuscript (postprint) of the following publication:

Published Version:

Cornigli, D., Reggiani, S., Gnudi, A., Gnani, E., Baccarani, G., Fabiani, D., et al. (2018). Electrical characterization of epoxy-based molding compounds for next generation HV ICs in presence of moisture. MICROELECTRONICS RELIABILITY, 88-90, 752-755 [10.1016/j.microrel.2018.07.030].

Availability: This version is available at: https://hdl.handle.net/11585/667585 since: 2020-11-26

Published:

DOI: http://doi.org/10.1016/j.microrel.2018.07.030

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D. Cornigli, S. Reggiani, A. Gnudi, E. Gnani, G. Baccarani, D. Fabiani, D. Varghese, E. Tuncer, S. Krishnan, L. Nguyen,

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Microelectronics Reliability, Volumes 88–90, 2018, Pages 752-755, ISSN 0026-2714

The final published version is available online at: https://doi.org/10.1016/j.microrel.2018.07.030

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Electrical characterization of epoxy-based molding compounds for next generation HV ICs in presence of moisture

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Abstract

The effects of moisture on the electrical properties of epoxy molding compounds containing high quantities of silica filler microparticles have been investigated by means of dielectric spectroscopy, steady-state conductivity and pulsed electro-acoustic space charge measurements. It has been shown that the presence of water at the filler/epoxy interfacial areas affects the low-frequency dielectric response of the materials and significantly increases their electrical conductivity, especially at higher temperatures. In addition, in presence of moisture, space charge measurements have shown the accumulation of heterocharges which significantly increase the electric field at the electrodes. These aspects are of primary importance, as the electrical properties of the encapsulation materials strongly influence the space charge distribution in high voltage devices.

1. Introduction

The high demands on weight, cost and size in new power electronic applications is leading to the need of plastic encapsulation materials with improved physical properties. Recently, epoxy-based polymers with a large fraction of insulating fillers have been proposed to decrease moisture permeability and suppress thermal expansion in order to avoid corrosion and exfoliation during high-temperature cycles [1]. In addition, the electrical conductivity is known to be source of charge spreading [2,3]. More specifically, high-voltage lateral devices, like, e.g., double-diffused DMOS (LDMOS) transistors, are based on the Reduced Surface Field (RESURF) principle [4] to significantly improve the breakdown voltage. But the performance of this class of devices is sensitive to the optimum charge balance in the drift region, which is in general at the silicon surface, and thus it is strongly impacted by mobile charges accumulated in the overlying isolation and mold-compound passivation on top [5]. For this reason, additional metal and/or poly-silicon field-plates and floating-rings are required to stabilize the surface electric field [6,7], even if they might limit the maximum intrinsic performance of the device. The understanding of mobile charge transport at the molding compound surface is thus of primary importance to improve the power device performance and reliability at the design stage. Moreover, lifetime tests are usually carried out with humidity as high as 85% RH. The latter condition would lead eventually to instability and failure of the device just due to the increased electrical transport of the packaging material.

In [8], molding compounds with different composition were investigated to assess the role of insulating fillers on the electrical conductivity, and charge spreading in the encapsulation was characterized on a dedicated IC test chip.

Due to the nature of polymers, water absorption is expected to occur in humid environments, inducing significant changes in the dielectric properties. Recently, polymers with nano-size fillers under moisture conditions has been investigated [9,10]. As nanoparticles have significantly different interface features, a specific characterization of micro-filled polymers is still needed.

In this work, the dielectric spectroscopy, DC conductivity and space-charge distribution of polymers with high filler contents as in commercial products have been analyzed under dry and humid conditions.

The reported data give useful information for assessing the reliability expectations on such composite systems.

2. Materials and sample characteristics

The encapsulation materials used in this study are epoxy-based conventional mold compounds (EMCs) employed in semiconductor packaging. The samples are 50 mm x 50 mm squares with a thickness h = 0.75 mm. The EMCs are labelled MC0, MC1 and MC2, and contain different fractions of silica micro-fillers. The filler content was measured by thermogravimetric analysis to determine the weight percentage. MC2 has the lowest filler content, while MC1 and MC0 contain 20 % and 25 % more filler, respectively.

3. Humidity absorption

The moisture absorption of the EMCs has been investigated by exposing the samples to humid conditions in a climatic chamber (Genviro-060-C) with precise control of temperature and relative humidity. As a preliminary treatment, the samples have been dried at 125 °C for 24 hours and weighted with an analytical balance (Sartorius CP 124 S) with an accuracy of 0.0001 g to determine their dry weight M_0 . Subsequently, the samples have been placed under 85 °C / 85% RH and periodically weighted until equilibrium was reached following [11]. The relative water mass $M_{\%}$ is calculated as:

$$M_{\%} = 100 \text{ x } (M_t - M_0) / M_0 \tag{1}$$

where M_t is the weight of a wet specimen at time t. The absorption plot in Fig.1 shows curves similar to literature data [12,13]. By assuming that the weight gain of the EMCs initially follows the single-phase

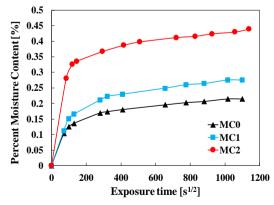


Fig. 1. Water uptake of the EMCs as functions of the square root of time.

Table 1	
Diffusivity and moisture equilibrium from Eq. 1.	

	MC0	MC1	MC2
D [mm ² /s]	5.1 x 10 ⁻⁶	4.0 x 10 ⁻⁶	5.4 x 10 ⁻⁶
<i>M</i> ∞ [%]	0.215	0.275	0.440

Fickian diffusion model, the diffusivity *D* has been calculated from the initial gradient of water uptake versus $t^{1/2}$ [11]:

$$D = \pi (h / 4 M_{\infty})^2 (M_2 - M_1)^2 / (\sqrt{t_2} - \sqrt{t_1})^2$$
(2)

where M_{∞} is the moisture content at equilibrium, h is the thickness of the samples and the last term corresponds to the slope of the moisture plot in the initial linear portion of the curve. The diffusivity and moisture equilibrium content extracted from the absorption plots are reported in Table 1. It can be noted that the diffusivity ranges between 4.0 x 10^{-6} mm²/s and $5.6 \times 10^{-6} \text{ mm}^2/\text{s}$ and seems to be independent from the filler fraction. In contrast, as the quantity of water absorbed by a polymeric material is related to the availability of free volume within the epoxy networks [14,15], significant variations have been measured in M_{∞} for the different EMCs. In Fig. 2, M_{∞} is plotted as a function of the filler fraction. The linear correlation between the two quantities is in agreement with similar measurements reported in literature [12,13]. confirming that a large fraction of insulating fillers limits the moisture permeability within the epoxy matrix. An opposite trend is obtained for nanocomposites [9,10]. . In order to optimize the filler content, additional experiments, such as long-term electro-thermal reliability tests, would be necessary.

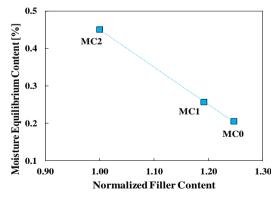


Fig. 2. Equilibrium moisture content at 85 °C/85 % RH as a function of the normalized filler content of silica with respect to the lowest one.

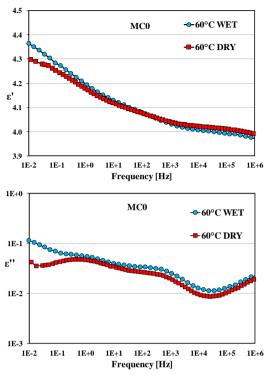


Fig. 3. (top) Permittivity and (bottom) loss factor of MC0 at 60 $^{\circ}$ C in (blue) wet and (red) dry conditions.

However, as the minimum water uptake is shown by MC0, which would assure the best reliability response against corrosion at the IC bond pads [16], the following investigations on charge transport effects have been carried out on this kind of material

4. Dielectric spectroscopy

Dielectric analysis (DEA) was carried out on a sample of MC0 both in dry and wet (0.21%) conditions. Circular gold electrodes with 40 mm diameter were sputtered on both sides of the sample and the analysis was carried out by using the Novocontrol Concept 80 alpha analyser. A small AC signal is used (V_{RMS} = 1 V). The permittivity ε ' and loss factor ε '' have been measured for frequencies f from 10^{-2} to 10^{6} Hz at 60 °C. In EMCs, the dielectric spectra generally show the effect of epoxy/filler interfacial polarization (ϵ ' at low *f*), the material conductivity (ϵ '' at low f), and the dipolar relaxation times (ε '' peaks) [2,17]. Moisture influences the dielectric properties of EMCs in different ways [9,18]. Water molecules in dispersed states can cause a dipolar response, while water clusters or droplets may cause interfacial polarization. As shown in Fig. 3, the wet sample

Table 2	
Conductivity measurements of MC0.	

Water cont	tent σ at 25 °C [S/cn	n] σ at 60 °C [S/cm]
0 %	7.0 x 10 ⁻¹⁸	1.4 x 10 ⁻¹⁷
≈ 0.1 %	1.9 x 10 ⁻¹⁷	3.5 x 10 ⁻¹⁷
≈ 0.2 %	3.2 x 10 ⁻¹⁷	2.5 x 10 ⁻¹⁶

exhibits an increase of both ε ' and ε '' at low f, while no significant differences are observed for f > 1 Hz. This suggests that the presence of water mostly increases the epoxy/filler interfacial polarization and the conductivity without changing the dipolar response.

5. DC current measurements

In order to study the relationship between the conductivity and the moisture content, the steady-state current densities have been measured. To this purpose, a current has been induced across the samples by applying a DC step voltage and it has been monitored with a Keithley-6514 electrometer for 1000 seconds into a thermostatic oven. The measurement time is long enough to separate polarization effects and carrier conduction, but short enough to avoid any significant moisture desorption during the measurement. To this purpose, the analysis has been limited to a maximum temperature of 60 °C, which has been checked to give rise to a desorption lower than 15% of the initial water content, while a larger reduction is expected to affect the validity of the analysis. The conductivity σ has been extracted from the steady-state current as:

$$\sigma = I h / (A V) \tag{3}$$

where *I* is the current, *V* is the applied voltage across the sample, *h* is the thickness of the sample, *A* is the area of the electrodes. All the measurements have been obtained by applying a step voltage of 3 kV, corresponding to an electric field of about 4 kV/mm and the experimental setup allows to measure a minimum conductivity about 10⁻¹⁹ S/cm. The analysis has been carried out on a sample of MC0 with different moisture contents at the temperatures of 25 °C and 60 °C and the results are reported in Table 2. It can be noted that the conductivity increases with the moisture content both at 25 °C and 60 °C. This effect is particularly enhanced at 60 °C, where the conductivity near moisture equilibrium content is about 18 times its dry counterpart. These considerations seem to support

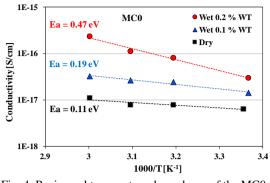


Fig. 4. Reciprocal temperature dependence of the MC0 conductivity at different humidity contents. The corresponding activation energy, E_a , has been extracted with eq. (4) and significantly increases when the material is in saturation conditions.

the comments on the dielectric responses in Section 4 and are in agreement with the results reported in the literature [9,10,19], where the low-frequency permittivity increases and the measured higher conductivity in wet nanocomposites has been attributed to the presence of water at the filler/epoxy interfacial areas. Within this framework, it is clear that the filler content and its size would affect the way in which water molecules are distributed inside the material. This, in turn, influences in a non-trivial way the conduction mechanisms, and hence the electrical properties of the material. A detailed analysis of these aspects would require a combination of numerical methods and analytical models [8,20]

In Fig. 4, the conductivities for different humidity conditions are reported as functions of the reciprocal temperature. It is possible to extract the activation energies corresponding to different humidity contents

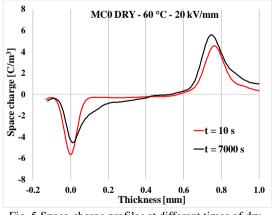


Fig. 5 Space-charge profiles at different times of dry MC0 at 60 °C, with a poling field of 20 kV/mm.

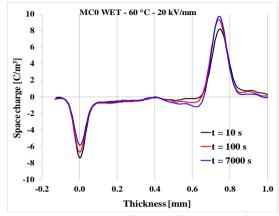


Fig. 6. Space charge profiles at different times of wet MC0 at 60 °C, with a poling field of 20 kV/mm.

by using the formula:

$$\sigma = \sigma_0 \exp(-E_a / k T)....(4)$$

where σ_0 is the pre-exponential term, E_a is the activation energy in eV, k is the Boltzmann constant and T is the temperature in Kelvin. It should be noted that the activation energy of σ is strongly influenced by the water content. In particular, it increases non linearly with the water content, passing from 0.19 eV to 0.47 eV as the relative water mass increases from 0.1 % to 0.2 %. This behavior is in contrast with what observed in literature for similar materials carried out at temperatures higher than the glass-transition one [18,21]. In the present work the analysis is carried out at temperatures significantly lower than the glass transition, which might cause a relevant difference in the involved physical effects and in their thermal dependencies.

6. Pulsed electro-acoustic analysis

The Pulsed Electro-Acoustic (PEA) analysis method is used to measure space charge distributions in solid insulating materials, and is based on the acoustic propagation through the material [22]. The samples are 50 mm x 50 mm squares with a thickness h=0.75 mm, without metallization. The space charge distribution of a sample of MC0 has been measured at 60°C for 7000 s in presence of a poling field of 20 kV/mm. Again, the wet sample refers to a moisture content of about 0.21 % wt, i.e., close to moisture equilibrium content. Space charge profiles at 10 s and 7000 s of polarization on dry and wet specimens are shown in Figures 5 and 6, respectively. It can be noticed that a significant amount of negative homocharge is accumulated close to the cathode of the dry specimen, due to charge injection from the electrode, as shown in Fig. 5. In the case of the wet specimen, the negative space charge turns into heterocharge accumulation close to the anode, as shown in Fig. 6. This behaviour could be associated with the increased conductivity of the wet specimen, which favours the transport of injected charge from the cathode to the anode and/or enhances the conductivity of ionic species [23]. This heterocharge build-up can enhance the local field at the electrode/insulation interface (see the increase of the anode peak at 7000 s), thus constituting a possible critical issue for the electrical insulation [24].

7. Conclusions

Moisture in highly-filled EMCs causes a significant increase in the conductivity, of about a factor 18 at 60 °C, with 0.2% water uptake, reached at 85C/85% in 178 hours. However, a water uptake as high as 0.19%, which is expected to increase the conductivity of a similar amount, is measured in less than 48 hours. Moreover, charge injection is improved in the presence of high fields, leading to charge spreading. As already observed in previous investigations, a modest increase of EMC conductivity leads to a significant increase of the surface potential on top of active regions of semiconductor devices. Thus, it is recommended that the electrical properties of EMCs under moisture condition should be evaluated to mitigate any unexpected degradation in device performance.

Acknowledgments

This work was supported by Texas Instruments Inc., Dallas, TX.

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